



# SYNCHROTRON-RELATED METHODS FOR INVESTIGATION OF NANOSTRUCTURES

**SOME**

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*A. Minkevich, E. Fohtung, ANKA synchrotron source, KIT Karlsruhe*

*S. Bernstorff, ELETTRA Trieste*



## OUTLINE:

1. What can be done by x-ray scattering?
2. Standard applications – rather boring
3. Iso-strain method & anomalous diffraction
4. DAFS
5. In-situ scattering
6. Nanobeams & coherent scattering
7. Summary

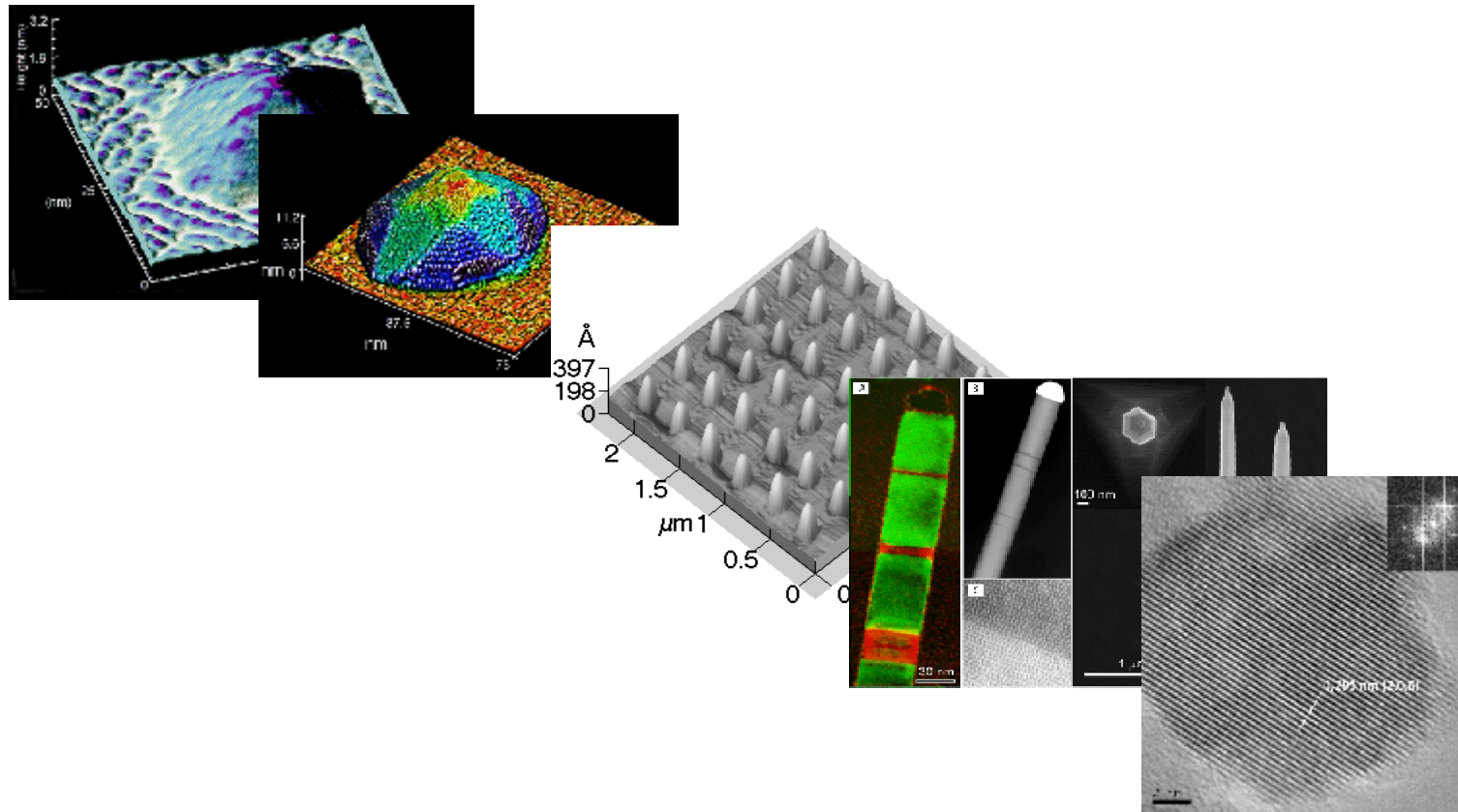


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# Various types of semiconductor nanostructures





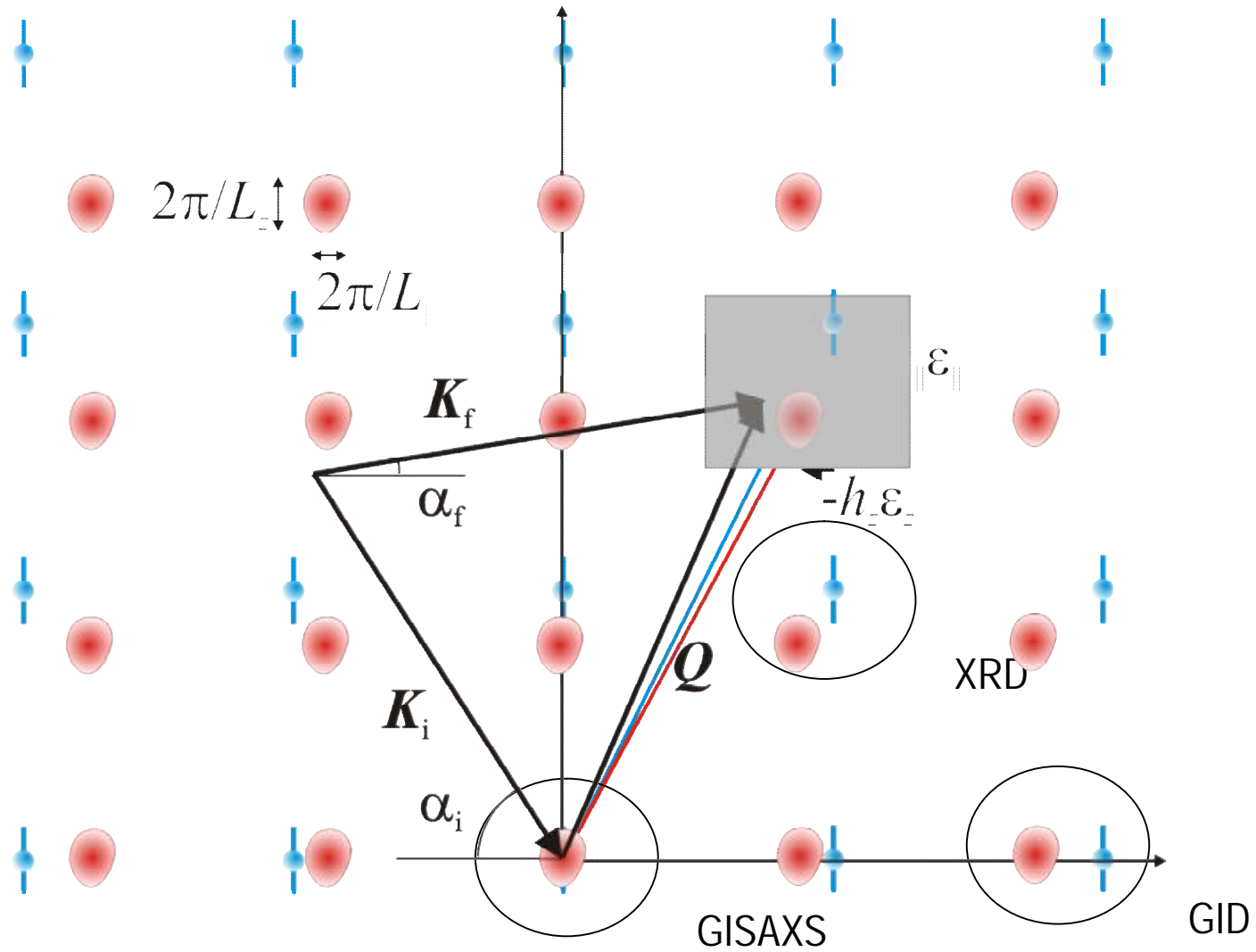
**Structure of self-assembled nanostructures:** position, sizes, chemical composition, atomic ordering

**Methods local in real space:** TEM, AFM, STM – investigation of few objects only

**Methods local in reciprocal space:** x-ray scattering assuming far-field limit – averaging over the irradiated volume (many objects). A direct interpretation of the data is impossible (the phase problem)

**New methods:** anomalous scattering, DAFS, coherent scattering – data can be interpreted almost directly without any data modeling

It is not possible to achieve both real-space and reciprocal-space resolution simultaneously (uncertainty principle) – **resolution problem**





Main problem:

very small volume of the nanoobjects  $\longrightarrow$  extremely weak intensity

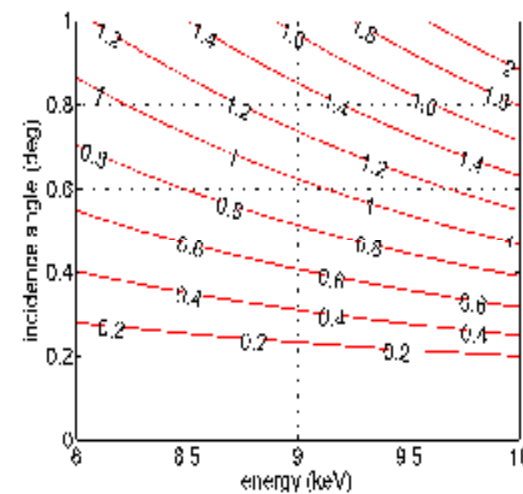
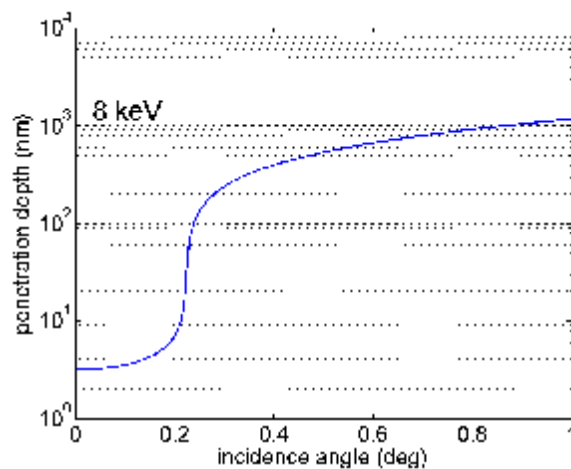
Possible solutions:

grazing-incidence geometry  $\longrightarrow$  reduction of the substrate signal

anomalous scattering  $\longrightarrow$  increase of the signal from the objects

Penetration depth of the primary radiation can be reduced down to few nm for incidence angles  $\alpha_i$  close to the critical angle  $\alpha_c$  of total external reflection. Analogously, the escape depth can also be tuned by changing the take-off angle  $\alpha_f$ .

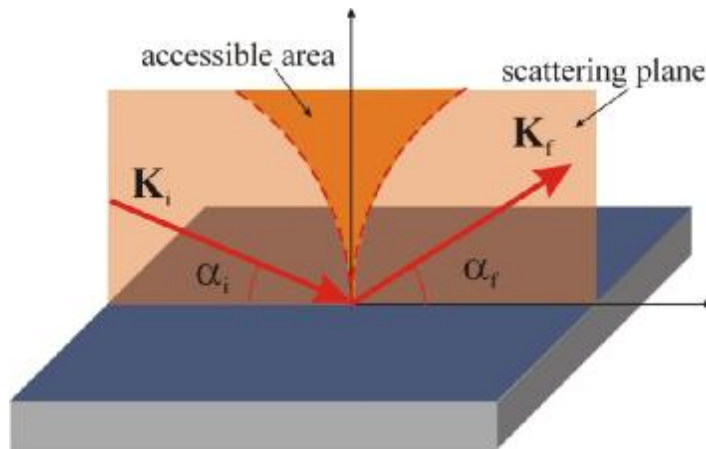
penetration depth of x-rays in Si



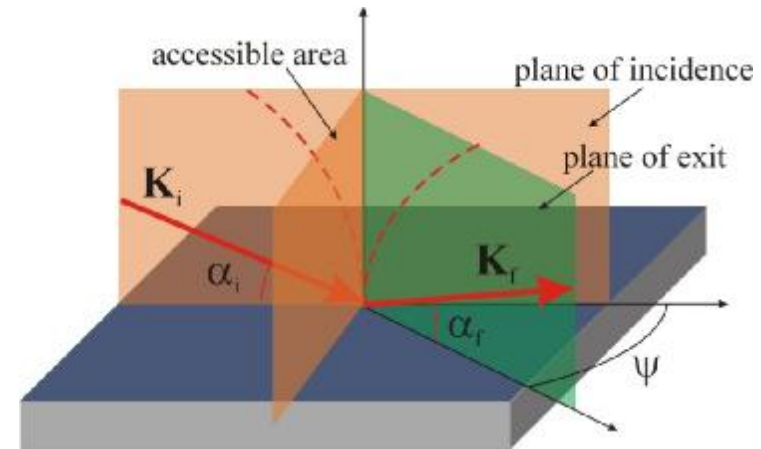
penetration depth in  $\mu\text{m}$



coplanar arrangement – x-ray reflection (XRR) and x-ray diffraction (XRD)



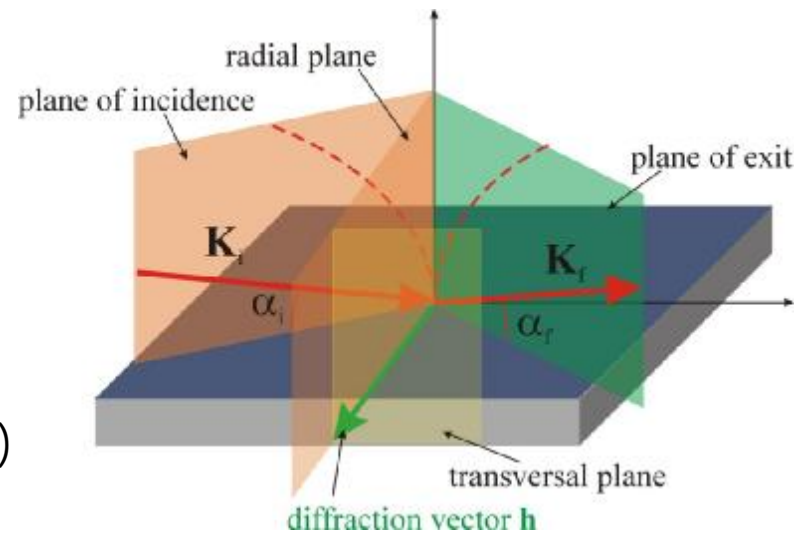
Non-coplanar grazing-incidence small-angle x-ray scattering (GISAXS)



scattering vector:  $\mathbf{Q} = \mathbf{K}_f - \mathbf{K}_i$

reduced scattering vector:  $\mathbf{q} = \mathbf{Q} - \mathbf{h}$

Grazing-incidence diffraction (GID)



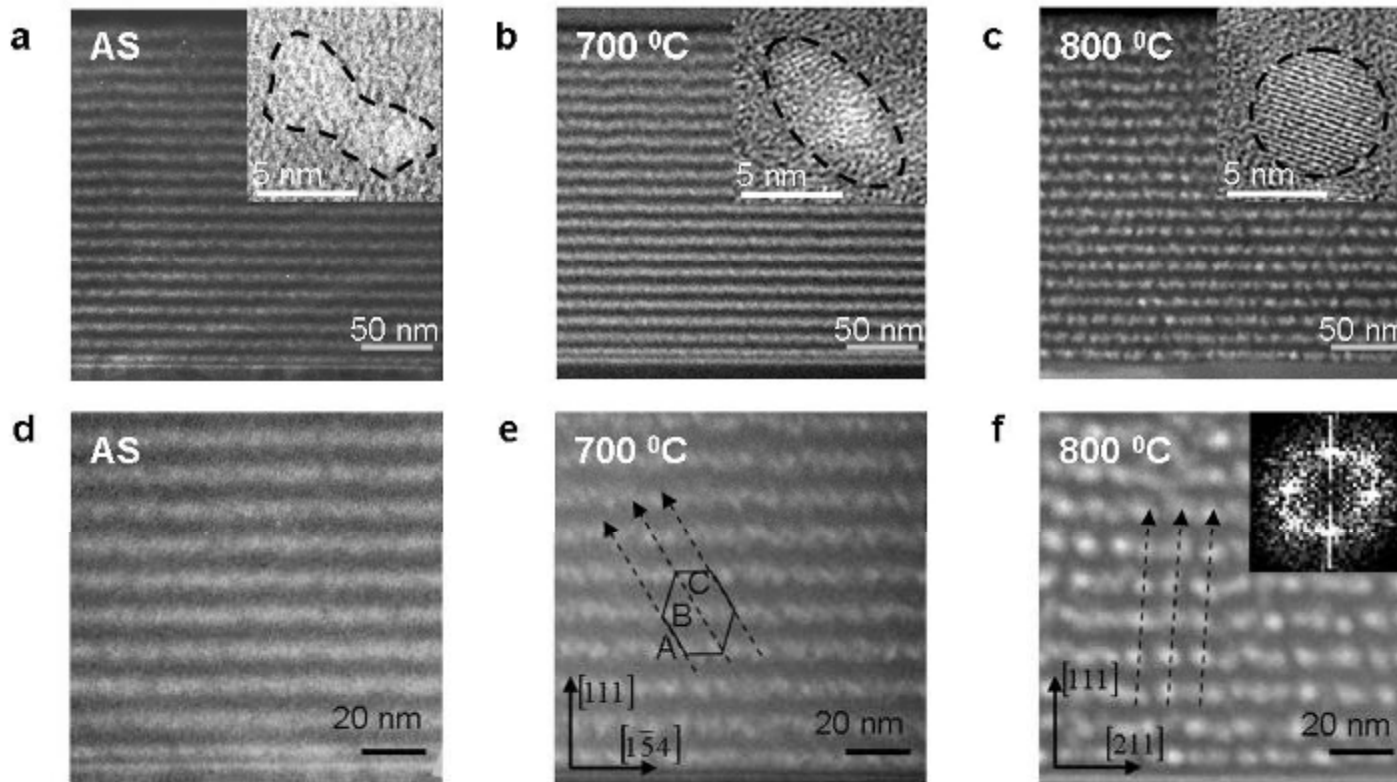




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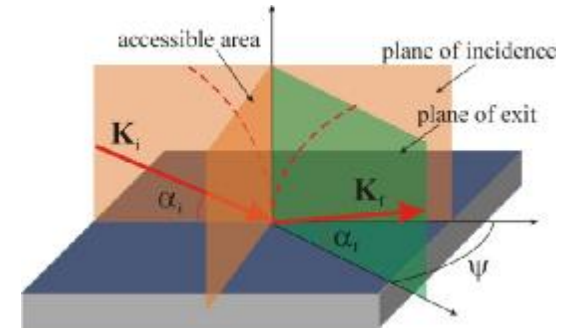
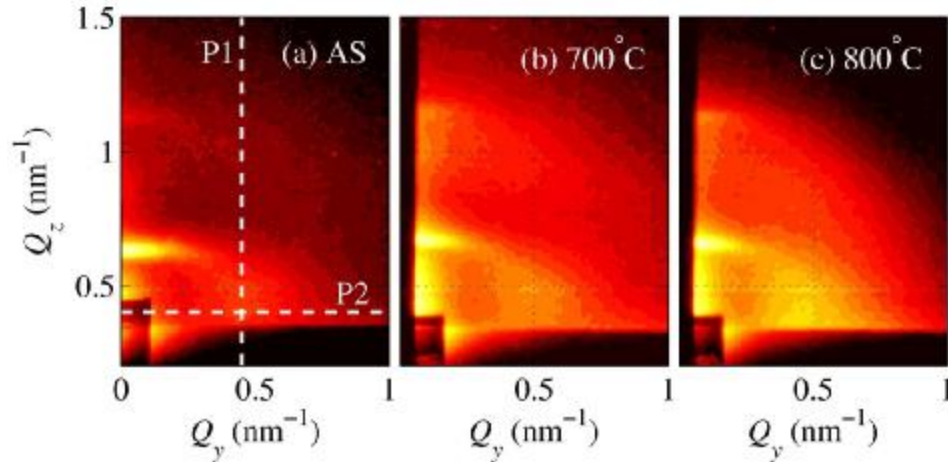


## Ge nanocrystals in Ge+SiO<sub>2</sub>/SiO<sub>2</sub> multilayer magnetron sputtering deposition (IRB Zagreb)



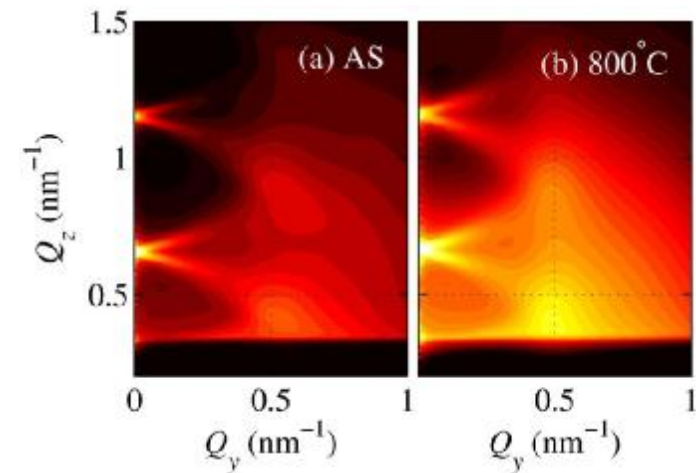
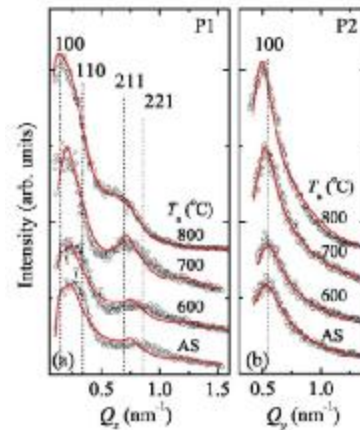


## GISAXS measurement (SAXS beamline @ ELETTRA)



simulated intensity maps

Fit to the theory –  
linescans:

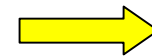
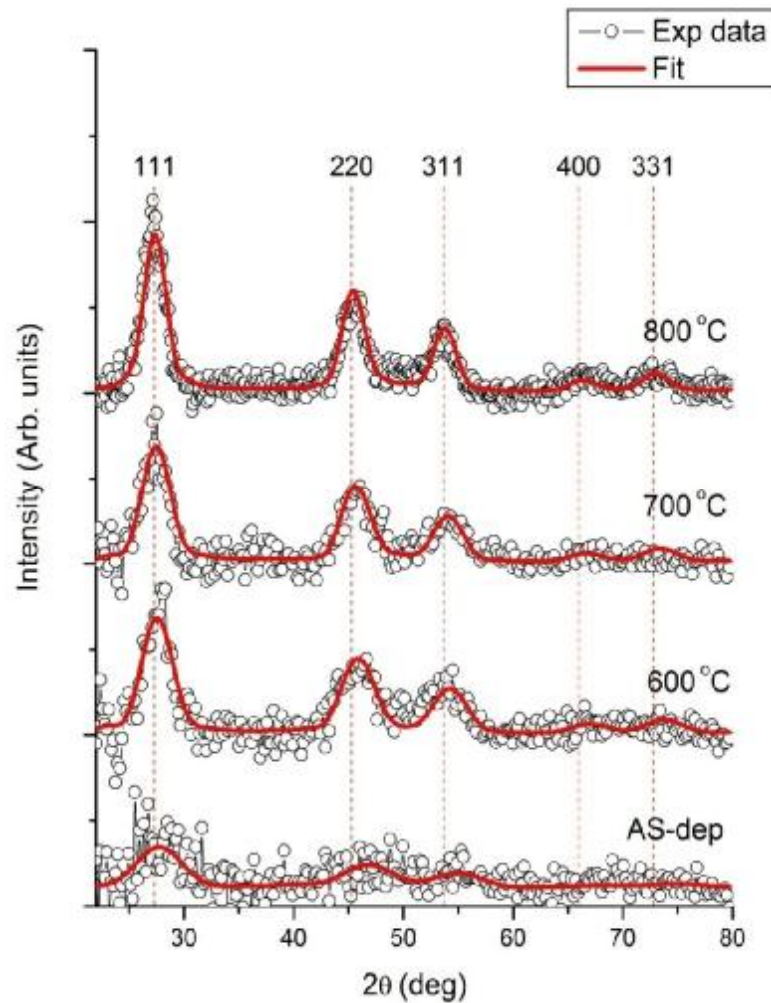


➡ size of individual nanocrystals, correlation parameters of their positions

See the talks by E. Majková and P. Šiffalovič



## X-ray diffraction – standard laboratory diffractometer, parallel-beam geometry

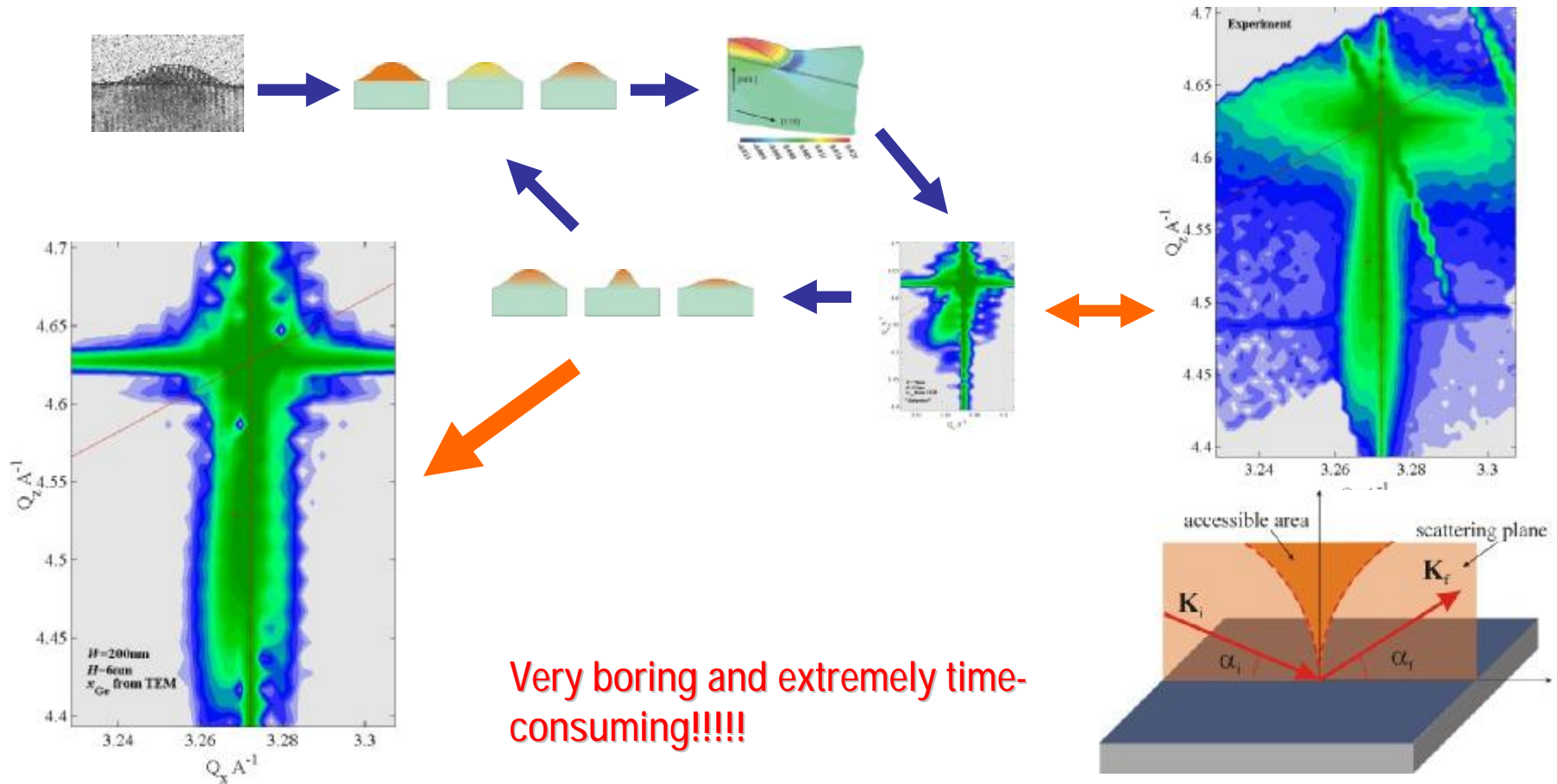


with increasing annealing temperature the density of defects (stacking faults, microtwins) increases



**GeSi free standing quantum dots on Si** sample grown by MBE, JKU Linz (Z. Zhong, G. Bauer, F. Schaeffler) ID10B@ESRF, J. Stangl, A. Hesse  
a "brute-force" approach

A. Hesse, J. Stangl, APL (2002)



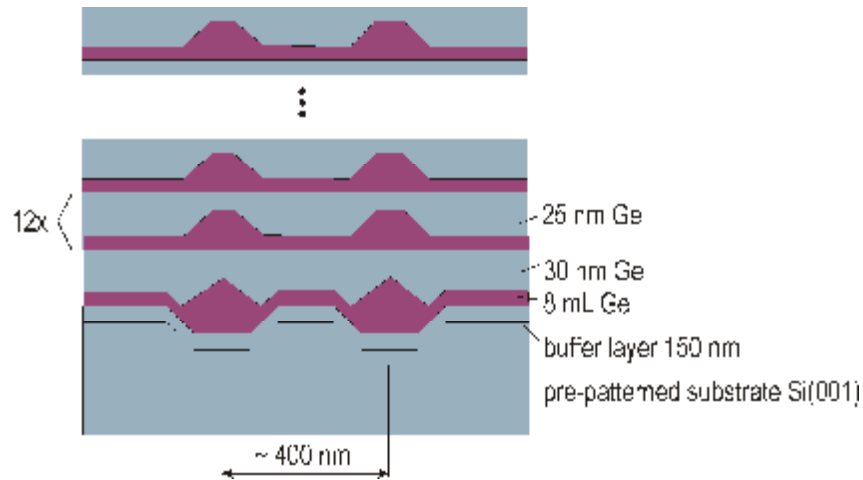
courtesy of J. Stangl

WSSR2011, Liptovský Ján

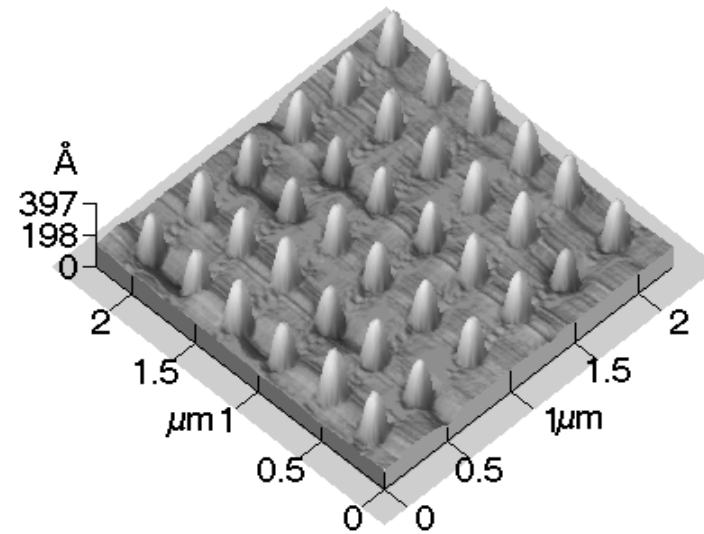


## Ge quantum dots in a Ge/Si superlattice grown on a pre-patterned substrate

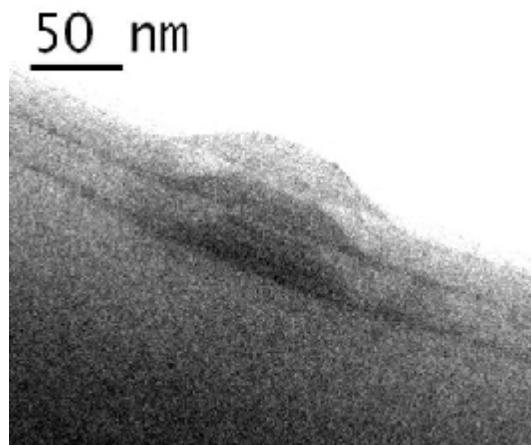
Samples grown by MBE at JKU Linz (prof. G. Bauer) and by D. Grützmacher (Jülich)



AFM: the first 2D layer of the dots

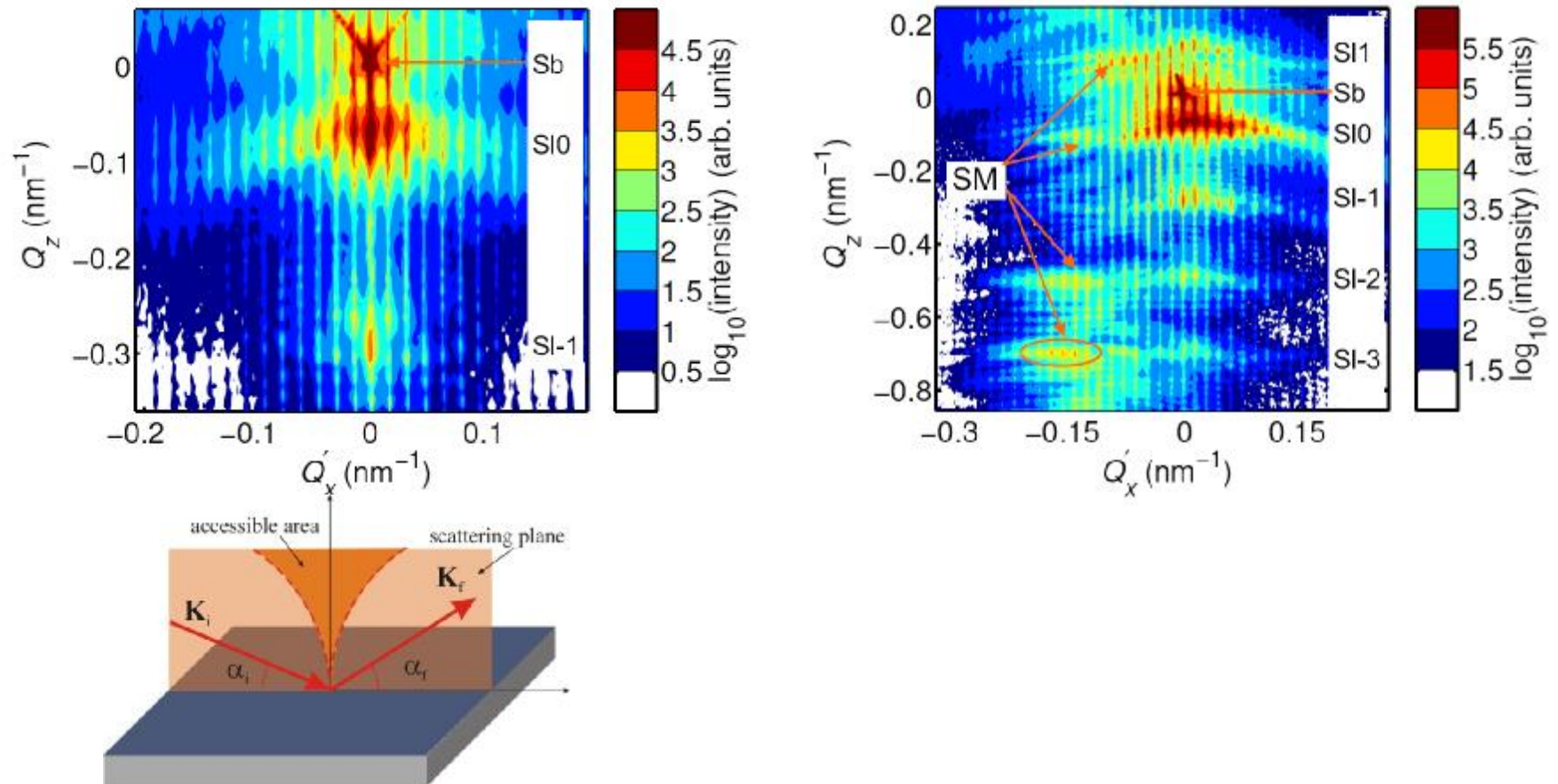


Cross-section TEM:



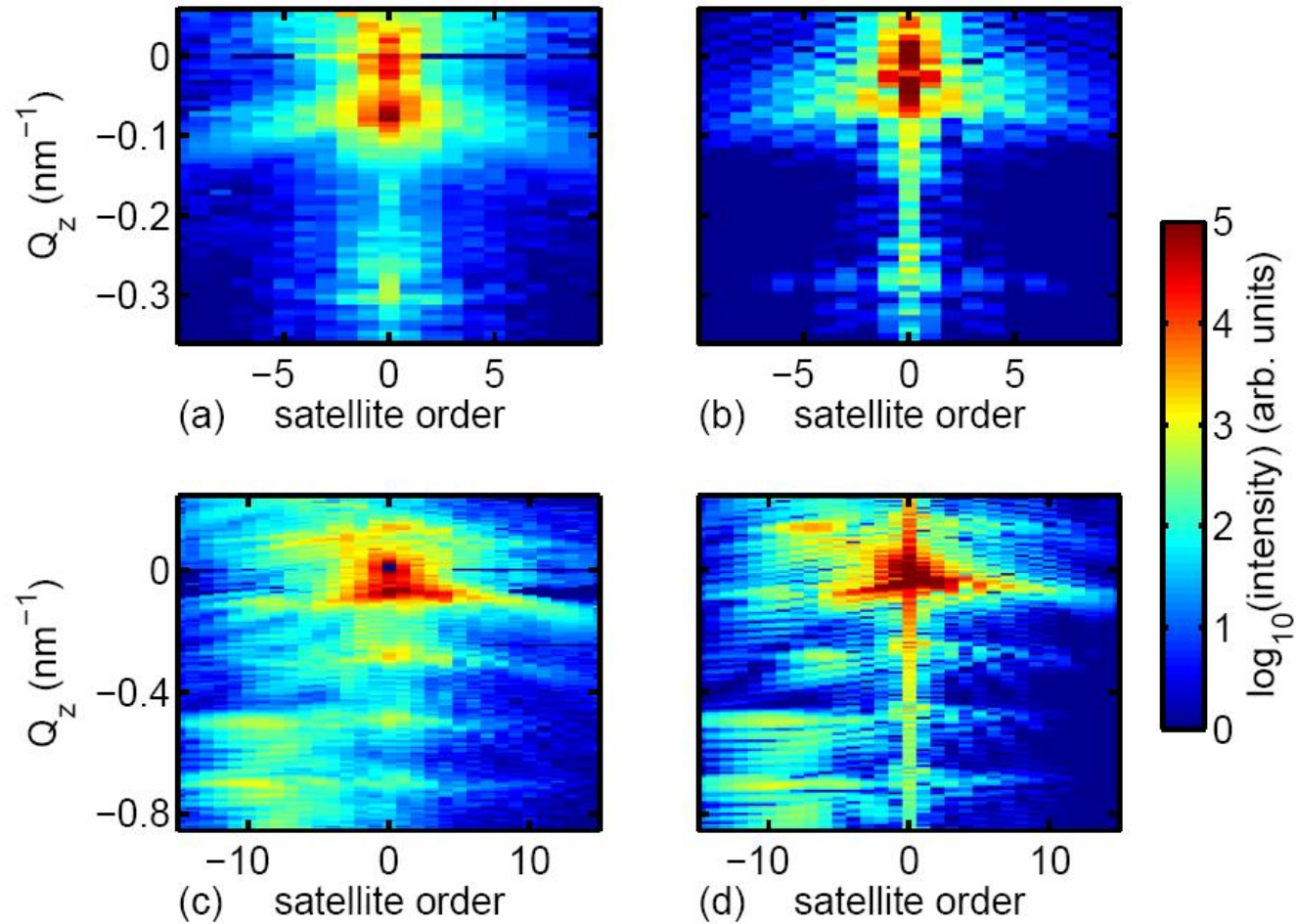


Coplanar 004 and 224 reciprocal space maps measured at the beamline ID10B, ESRF  
 $\lambda = 1.547 \text{ \AA}$

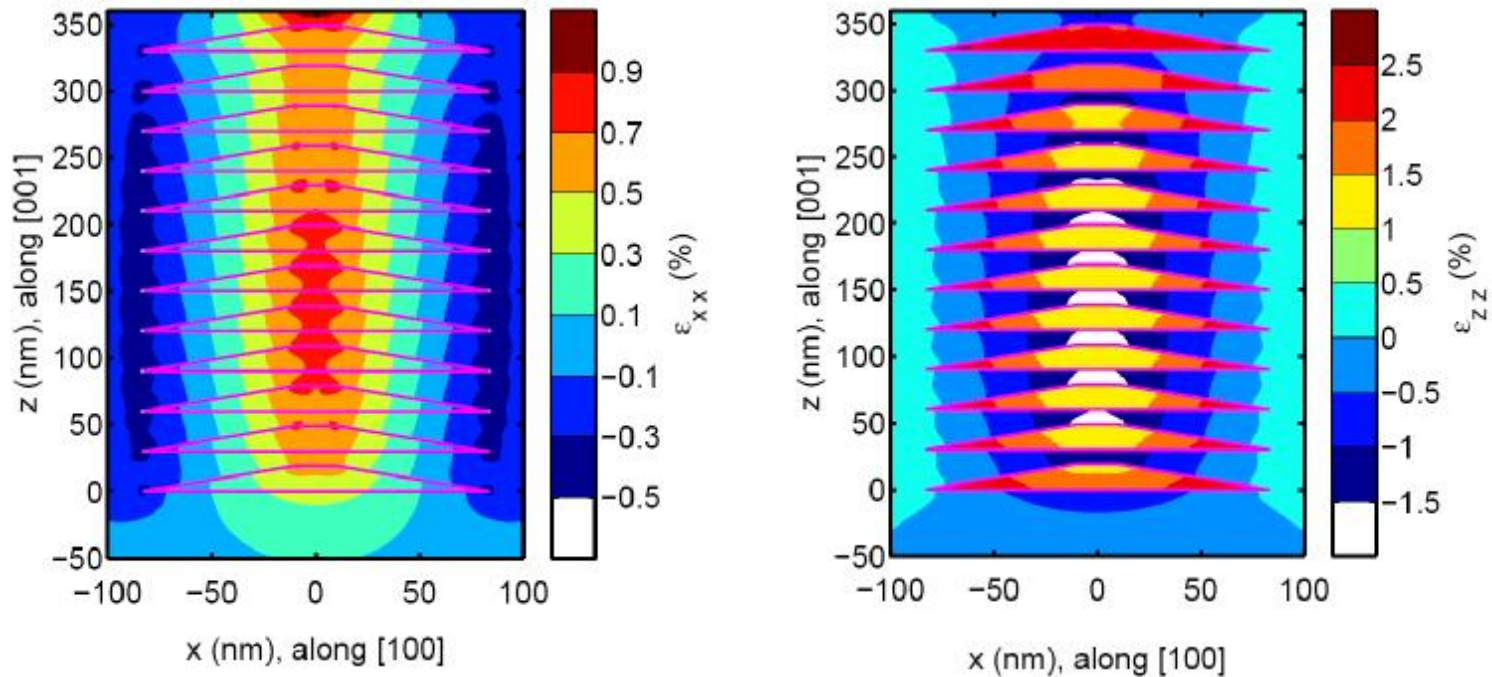




Experimental (left) and simulated (right) heights of the satellites in diffractions 004 (upper row) and 224 (lower row)







Resulting distribution of the strain tensor: the lateral (left) and the vertical components (the right panel). The Ge content increases from 30% (the bottom of the dot) to 45% at the dot apex.

J. Novák et al., J. Appl. Phys 98, 073517 (2005).

V. Holý et al. Phys. Rev. B 79, 035324 (2009).

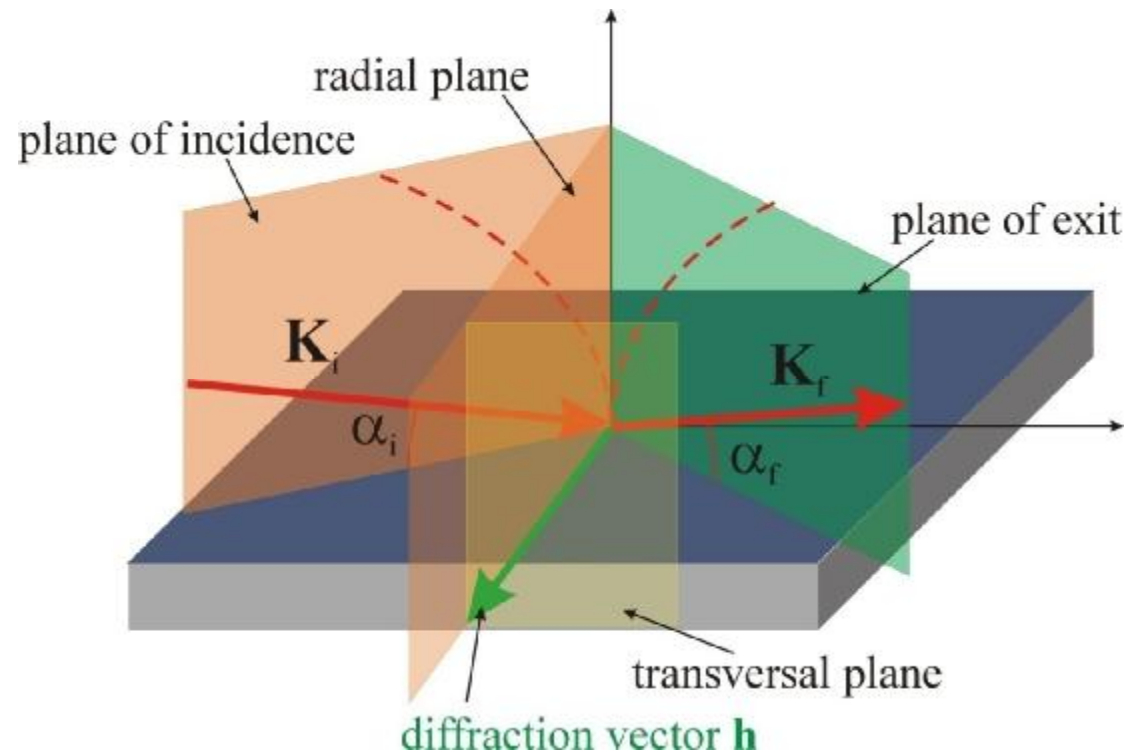


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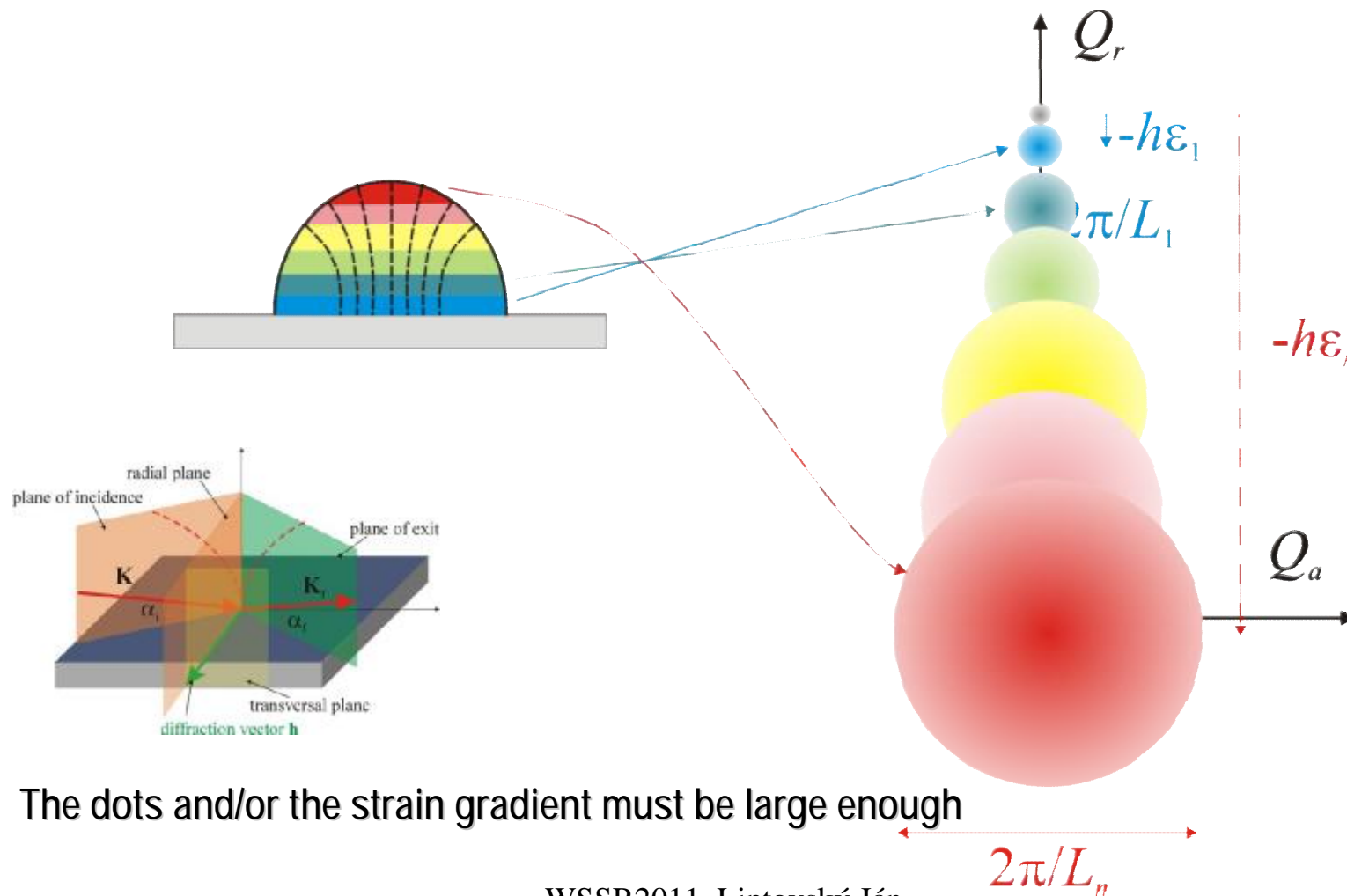
## Iso-strain method:

decomposition of an island into disks of constant strain





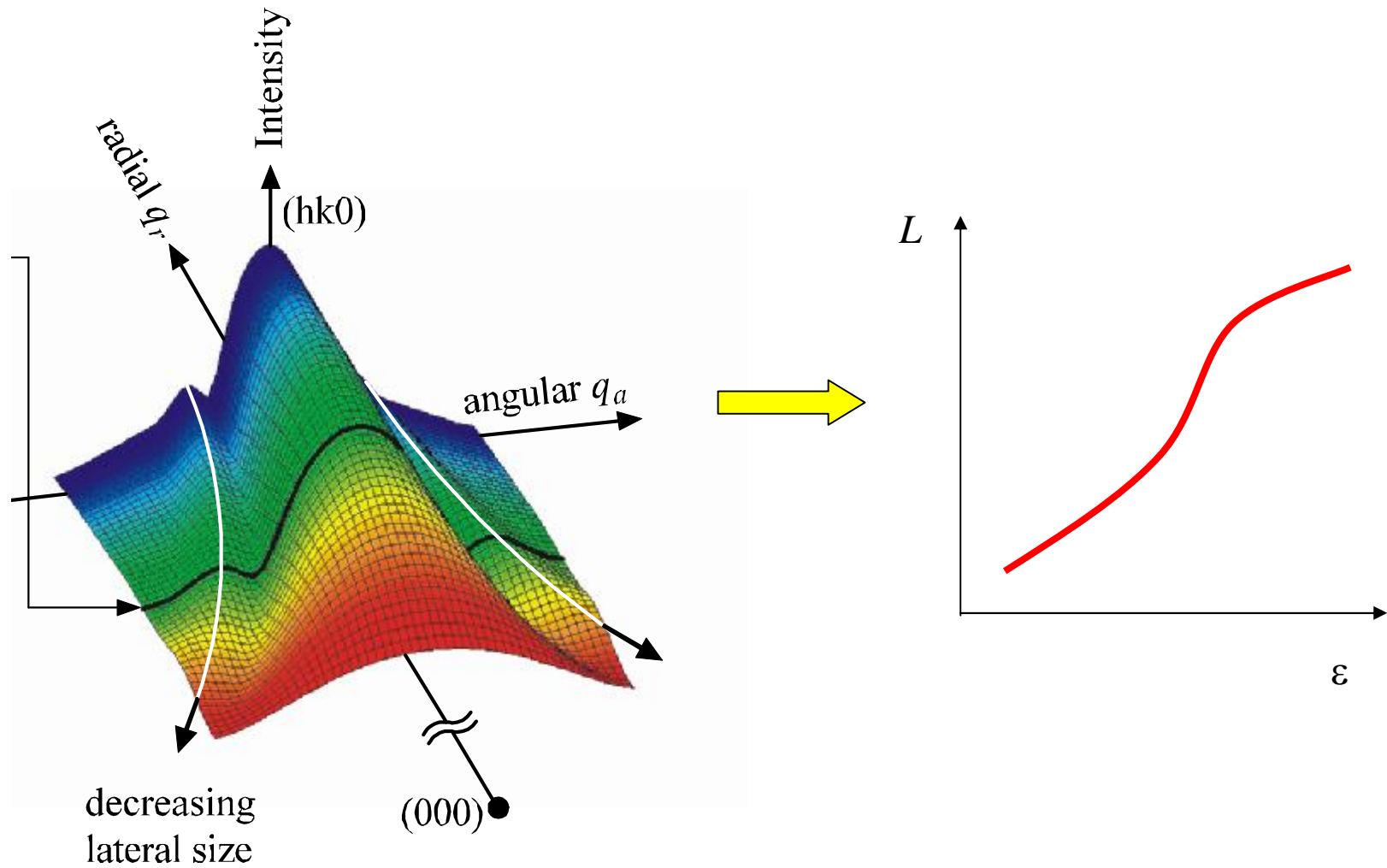
## lateral size of iso-strain volumes



The dots and/or the strain gradient must be large enough

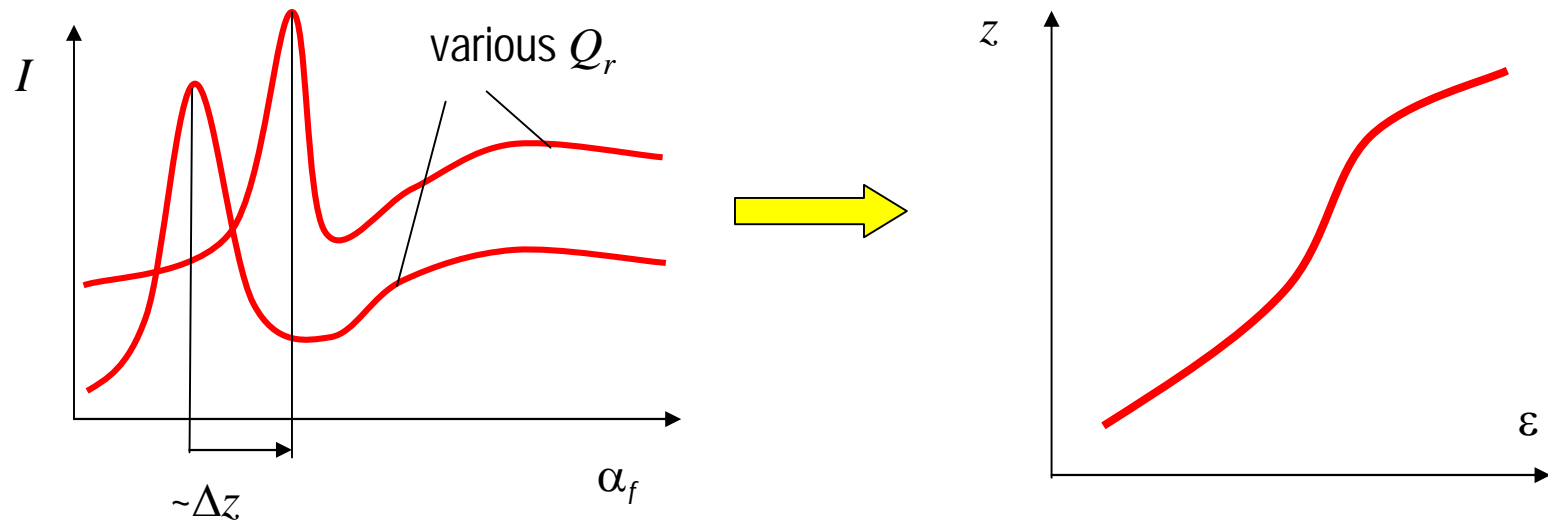
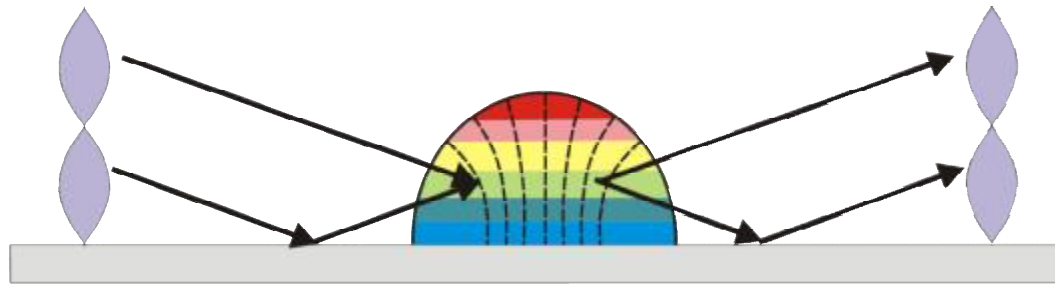


I. Kegel et al., Phys. Rev. Lett. 85, 1694 (2000); Phys. Rev. B 63, 035318 (2001).





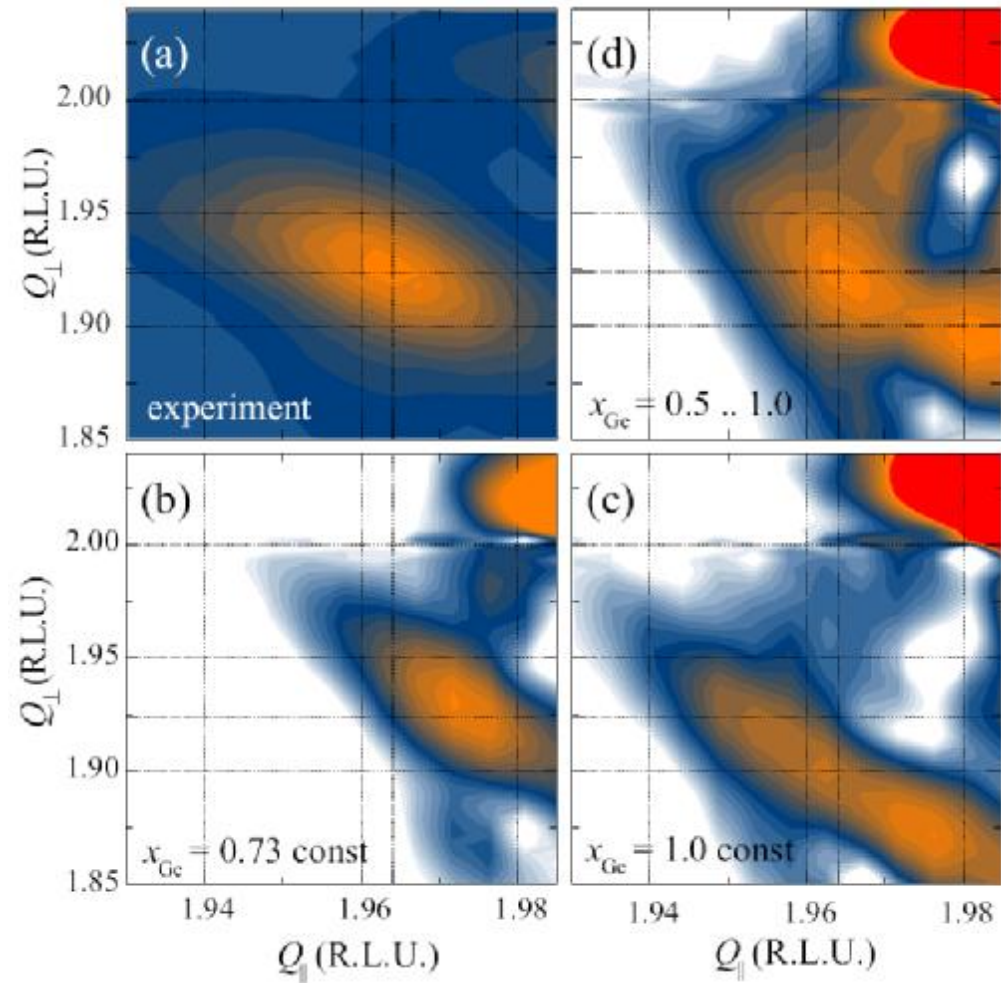
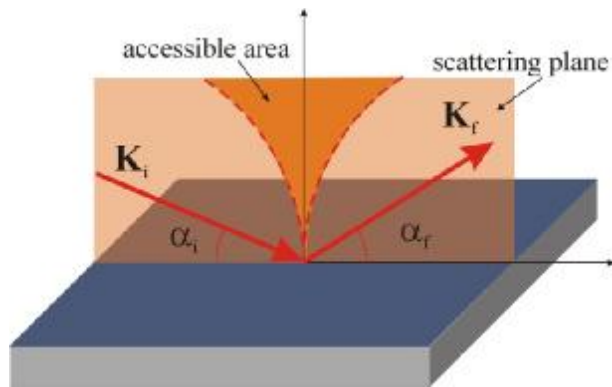
## Vertical positions of iso-strain volumes





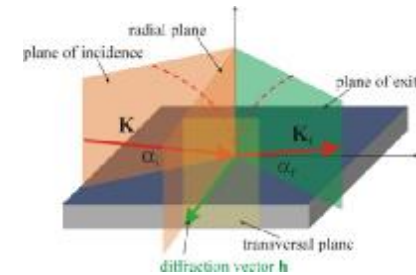
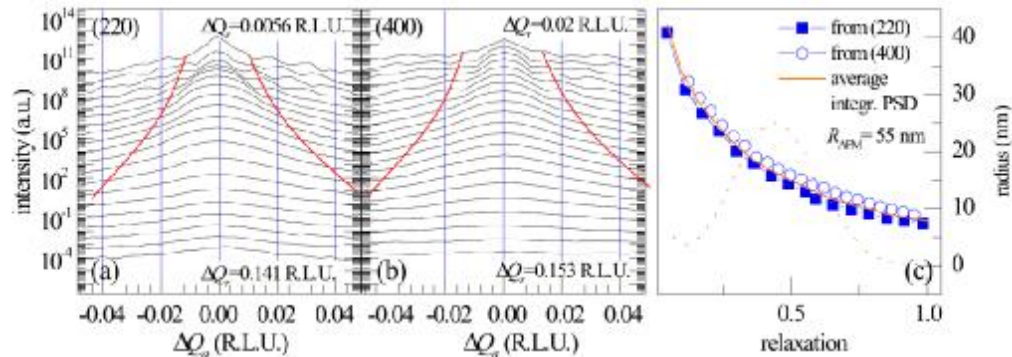
## Ge islands on Si:

Conventional 202 reciprocal space map and its simulations

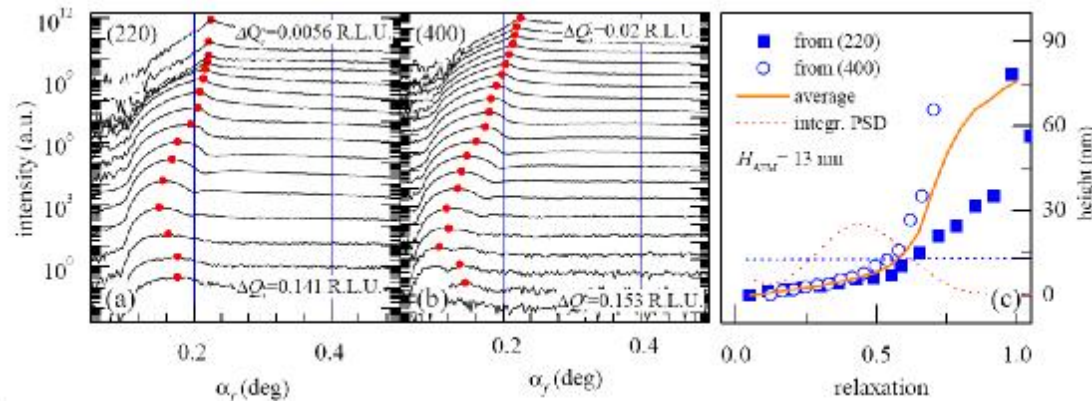




angular scans measured in 220 and 400 (a and b) and the resulting dependence of the radius of the iso-strain area on the relaxation degree (c)



the  $\alpha_f$  scans taken for various radial positions  $q_r$  in 220 and 400 diffractions (left) and the resulting dependence of the vertical position of the iso-strain area on the relaxation degree



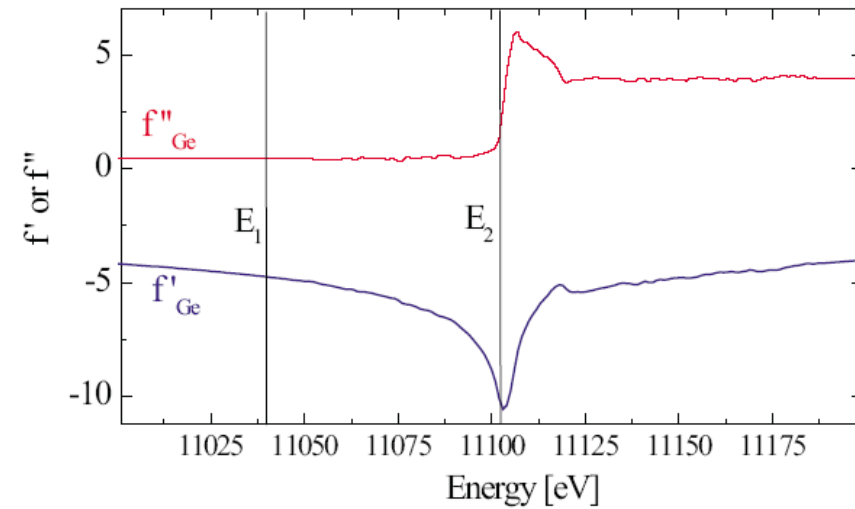
J. Stangl and V. Holy, *X-ray scattering methods for the study of epitaxial self-assembled quantum dots*, in: Quantum dots: Fundamentals, applications and frontiers, eds. B. A. Joyce, P. C. Kelires, A. G. Naumovets and D. D. Vvedensky, NATO Sci. Series II, vol. 190, Springer, Dordrecht 2005



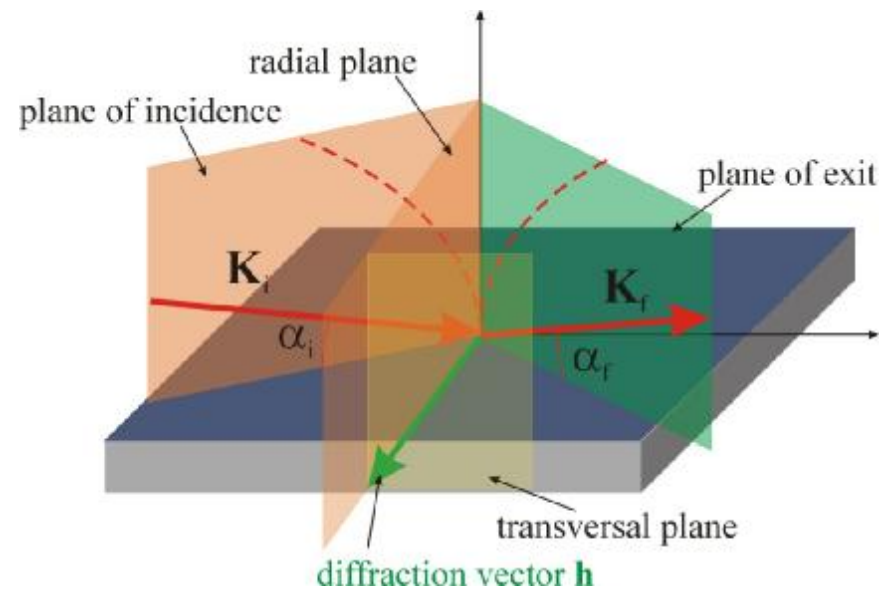


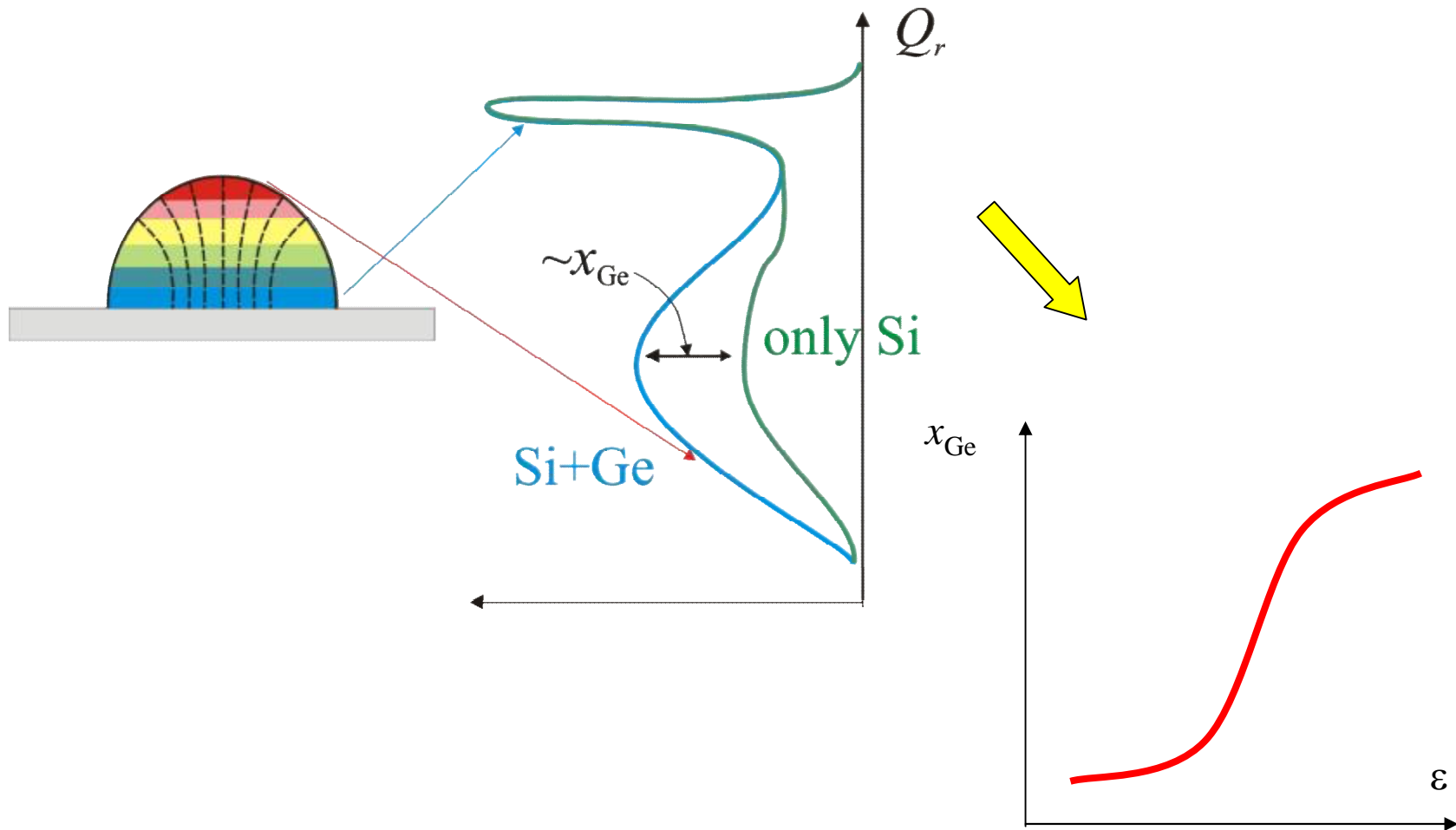
## Local chemical composition: Anomalous diffraction

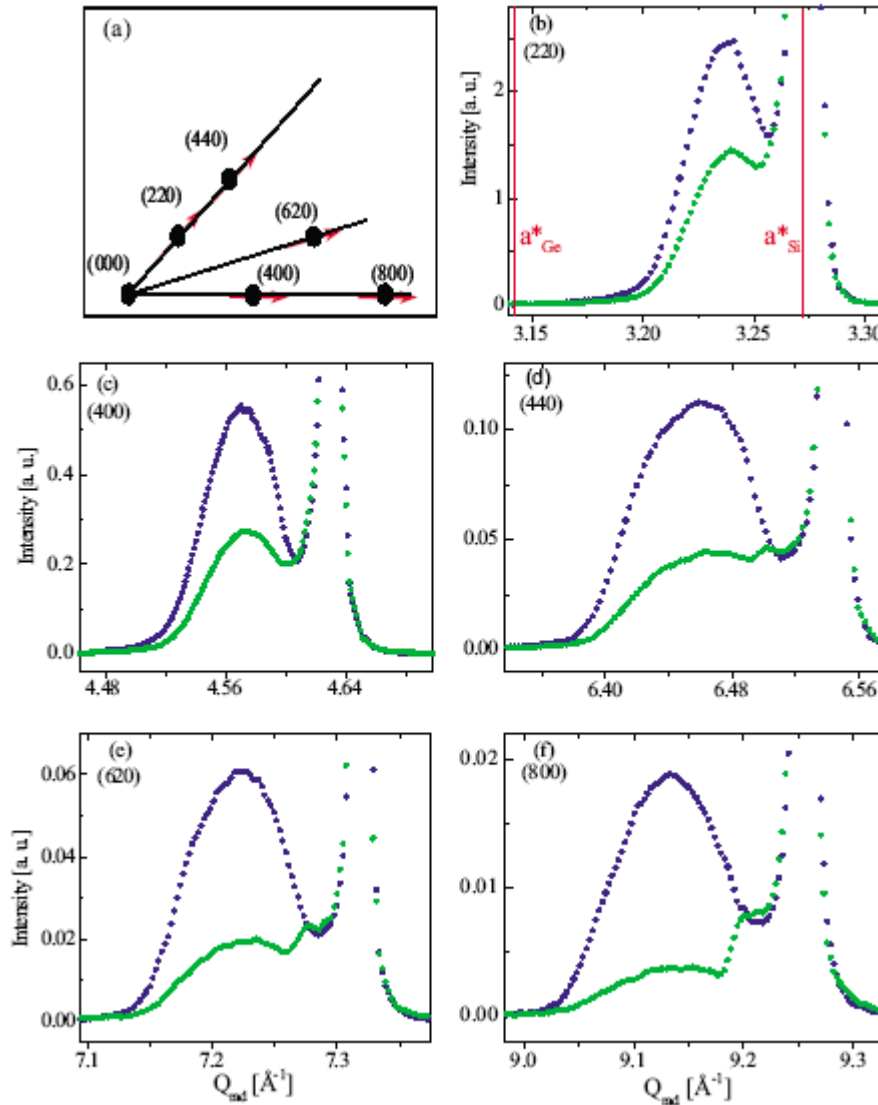
energy dependence of the dispersion corrections for Ge



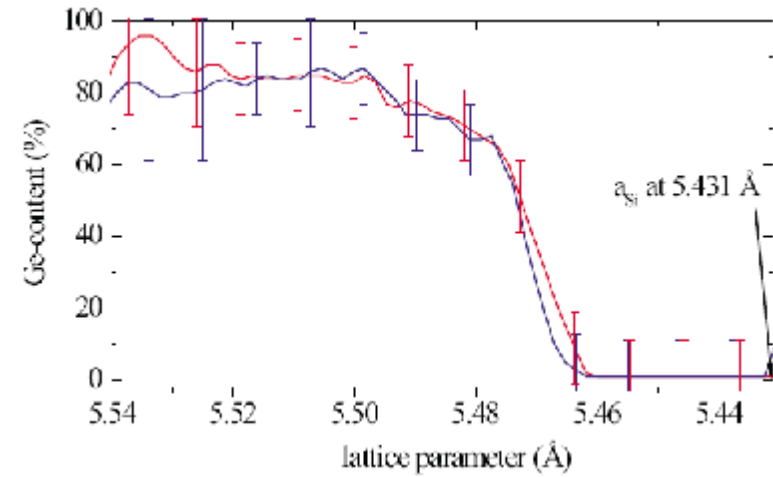
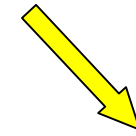
Grazing-incidence diffraction







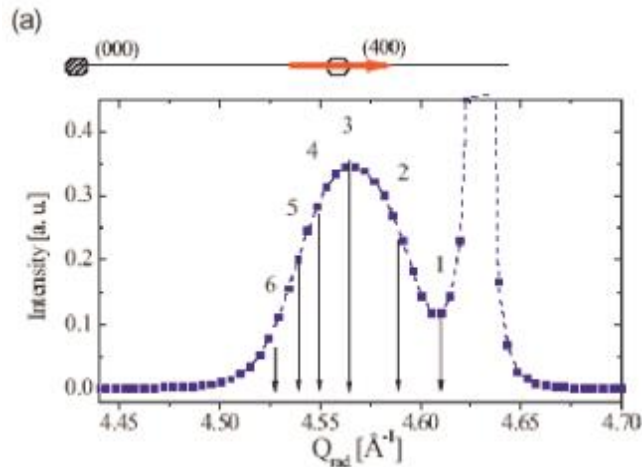
green dots – higher energy, only Si  
blue dots – lower energy



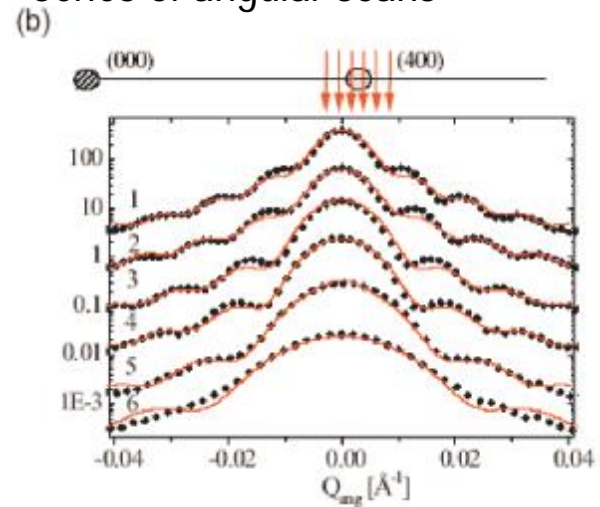
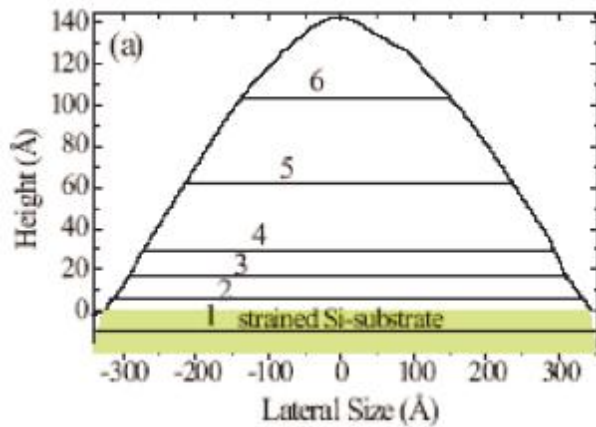
Courtesy of Tobias Schuelli, CEA Grenoble



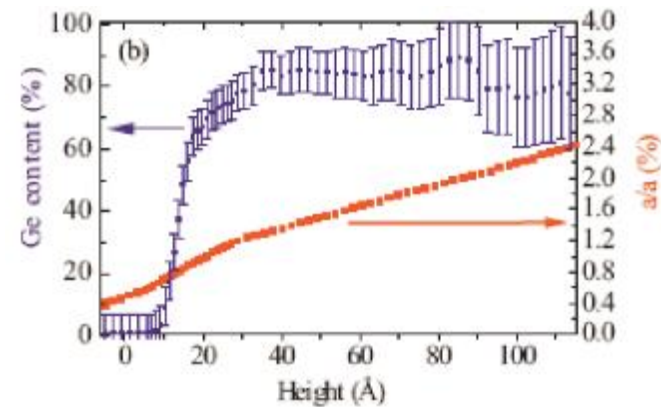
series of angular scans



reconstructed dot shape



reconstructed profile of Ge content



T. U. Schuelli et al, PRL 90, 066105 (2003).



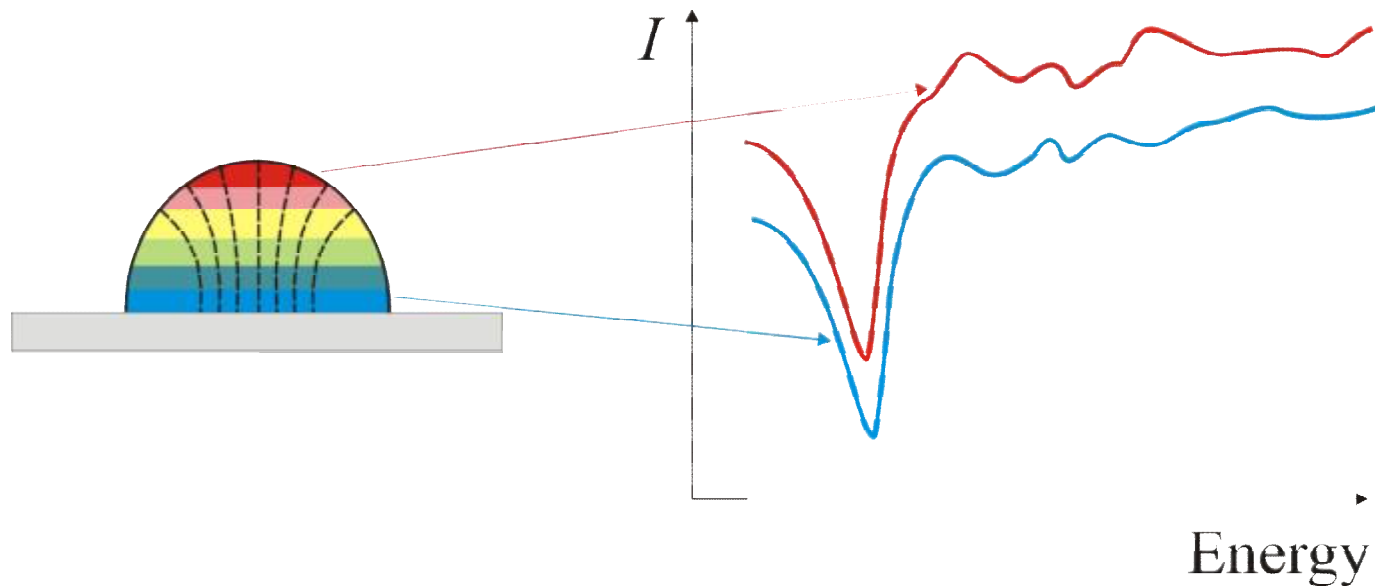
1. What can be done by x-ray scattering?
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**DAFS:** Combination of x-ray absorption spectroscopy (EXAFS) and diffraction – measurement of the energy dependence of the diffracted intensity **with constant scattering vector  $Q$**



The scattered intensity stems from a certain part of the sample with given strain (iso-strain volume)

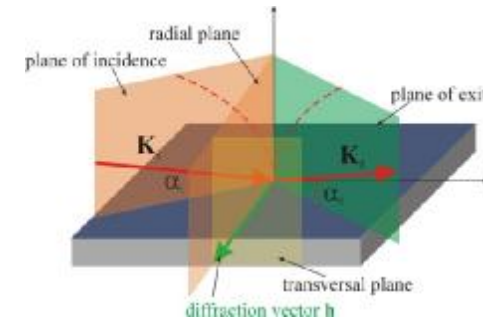




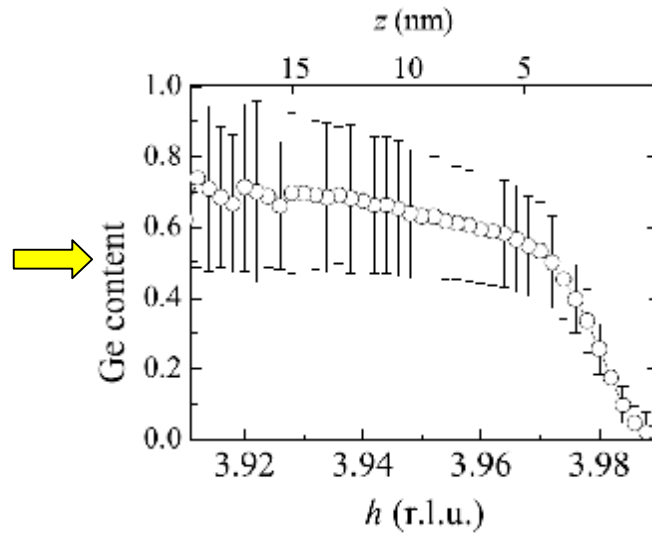
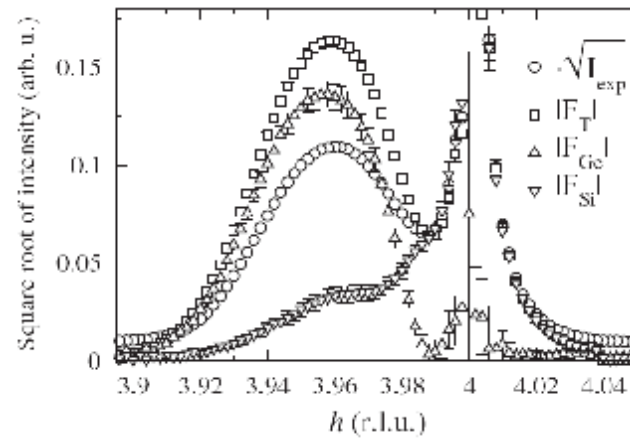
## Ge quantum dots on flat and pre-patterned Si(001) substrates

T. U. Schuelli et al., Phys. Rev. Lett. 102, 025502 (2009)

M.-I. Richard et al., Eur. Phys. J. 167, 3 (2009)

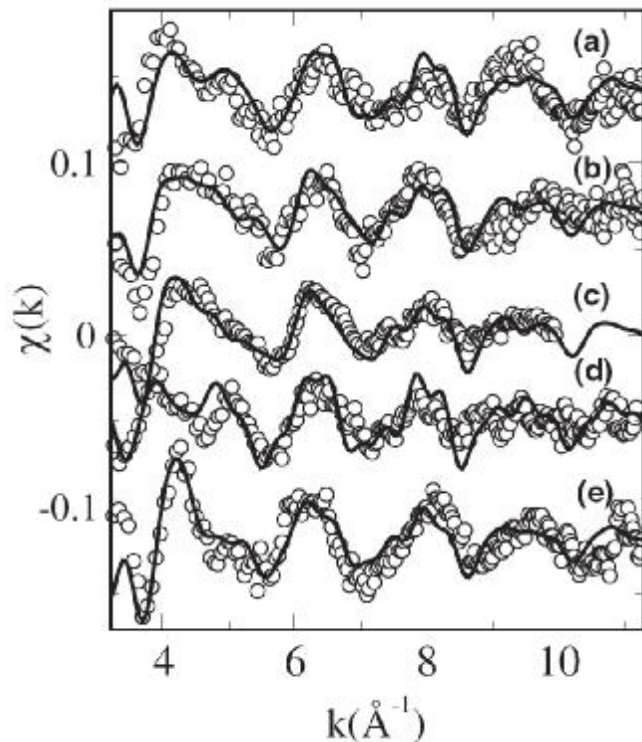
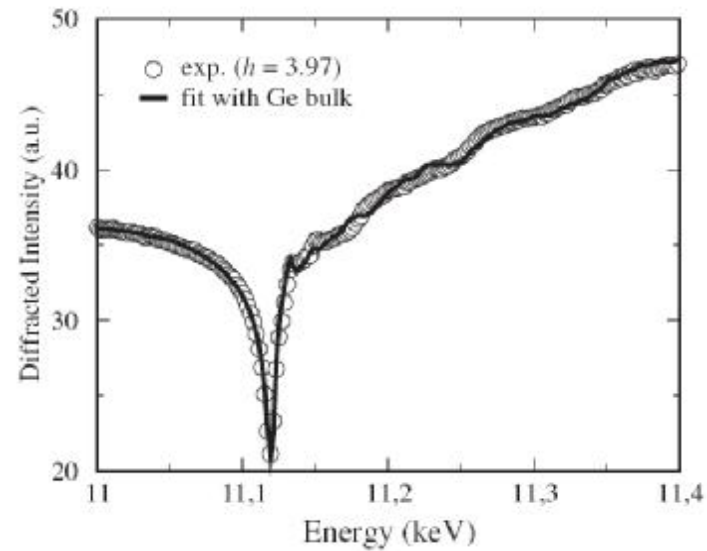


anomalous GID 400





example of a DAFS curve measured at (3.97 0 0)



measured (points) and VFF-simulated (lines) EXAFS oscillations measured in various reciprocal lattice points

Bond lengths and local Ge content in various parts of the dot volume





## CdSe/ZnSe quantum dots

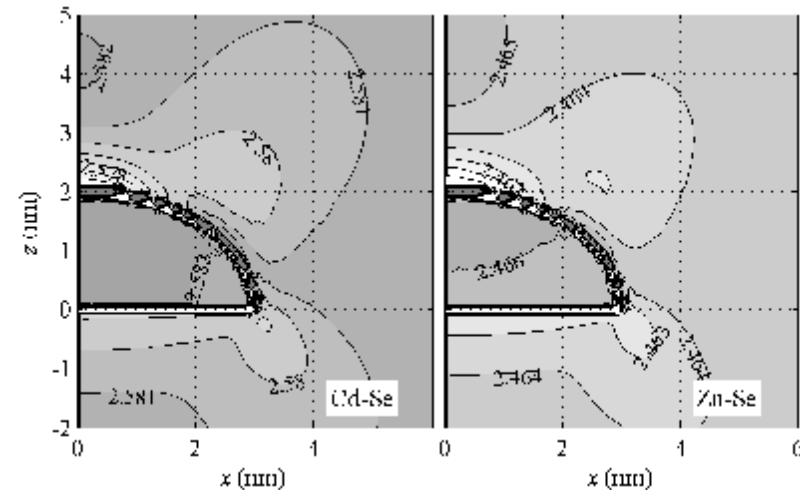
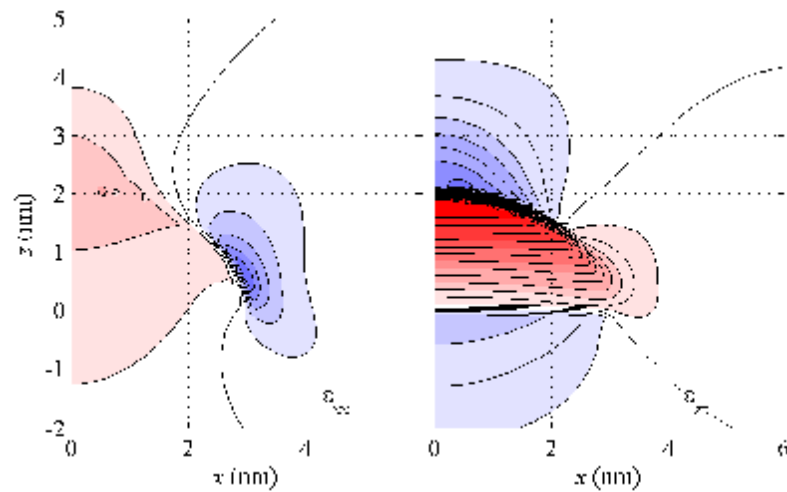
Samples grown by MOVPE, D. Hommel, Univ. Bremen, measurement BM02@ESRF

DAFS measurement around the SeK edge

Inhomogeneities in local elastic strains in and around the dots

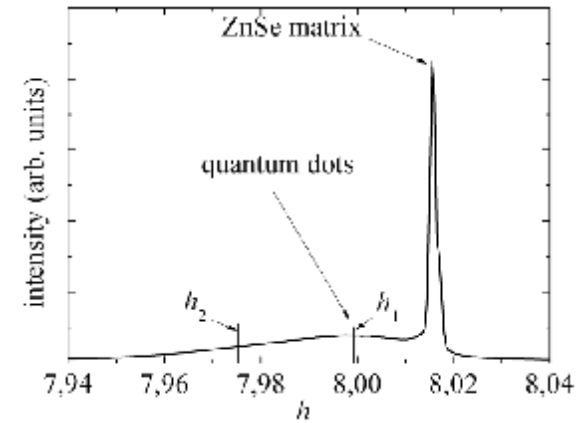
lead to

inhomogeneities in the Zn-Se and Cd-Se bondlengths  
(calculated by the simplified valence-force field model – d’Acapito)

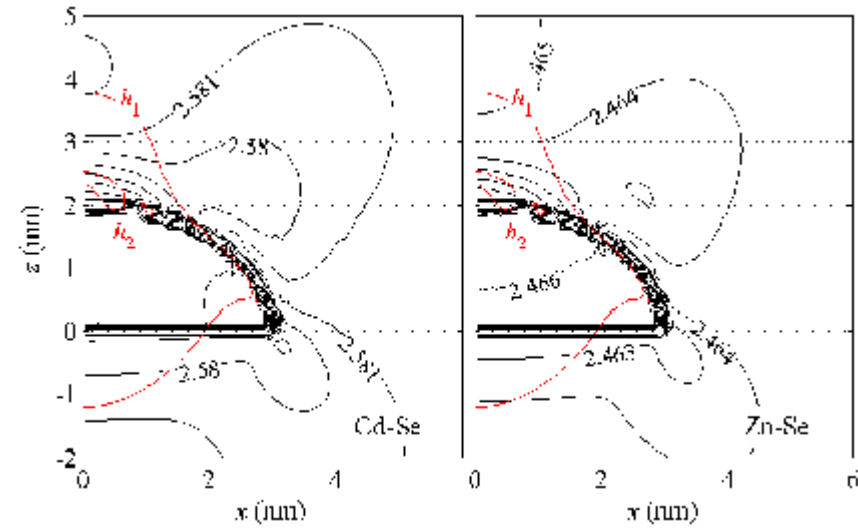




# GID, diffraction 800, radial scans

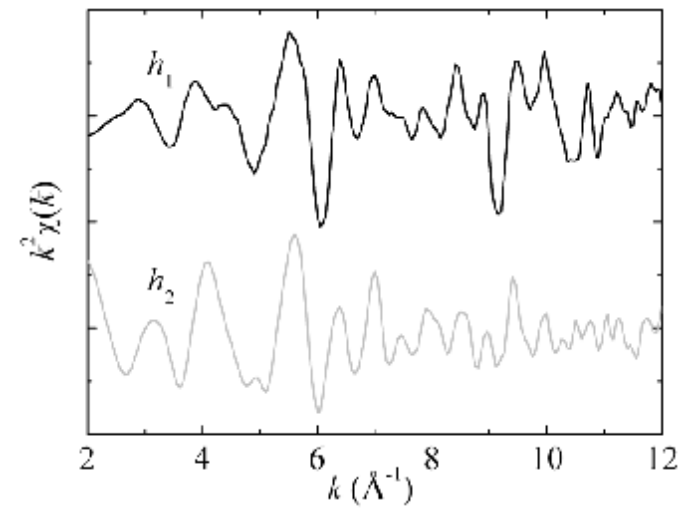
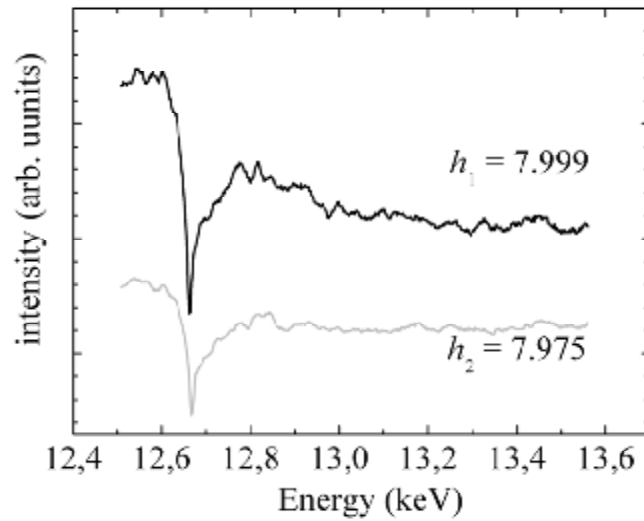


Choosing the working-point ( $h_1$  or  $h_2$ ) we change the iso-strain volume:





and the DAFS spectra are completely different



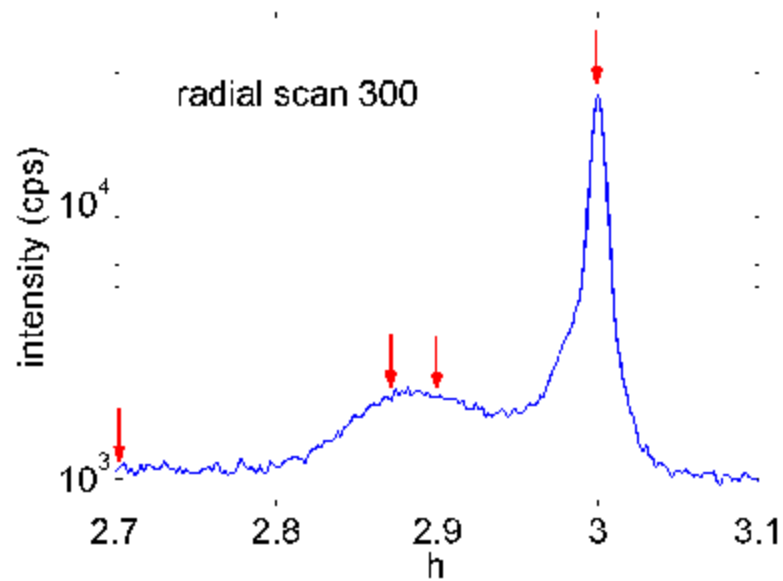
Not finished yet ☹



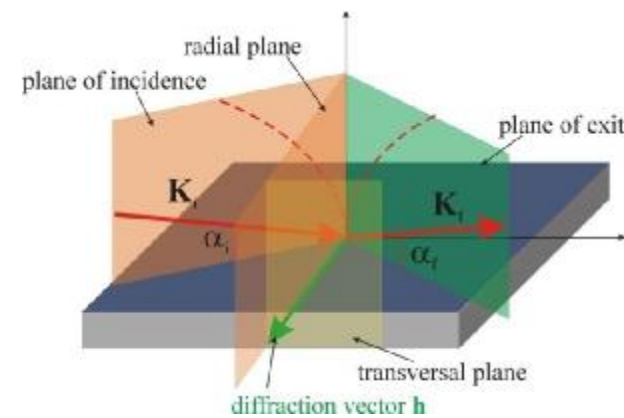
## Free-standing InGaN quantum dots on GaN:

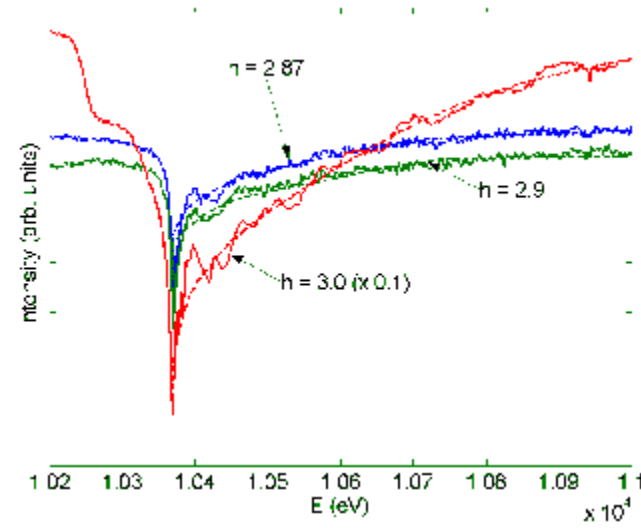
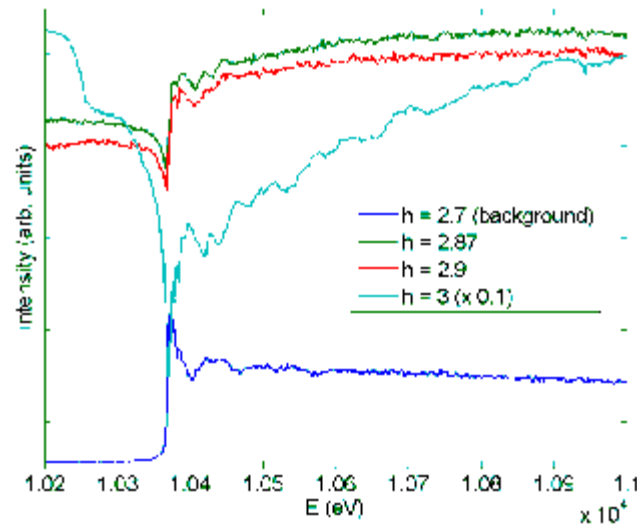
Samples grown by MOVPE, D. Hommel, Univ. Bremen, measurement BM02@ESRF

DAFS measurement around the GaK edge, the anomalous atoms (Ga) are everywhere ☹



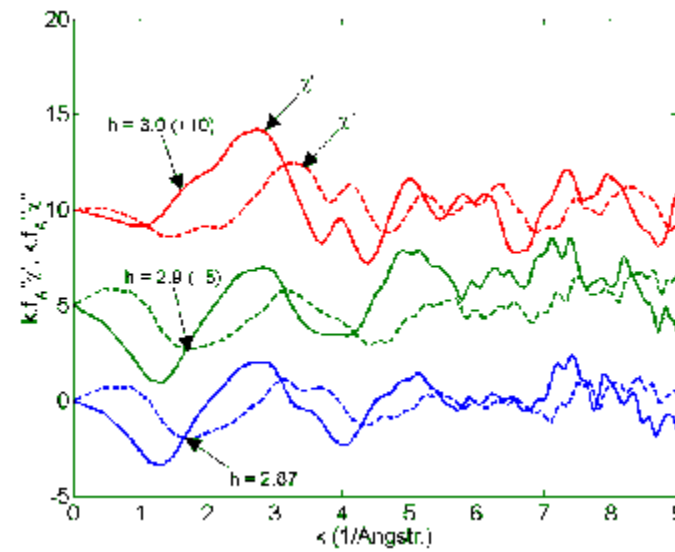
The DAFS scans were measured in the points denoted by arrows





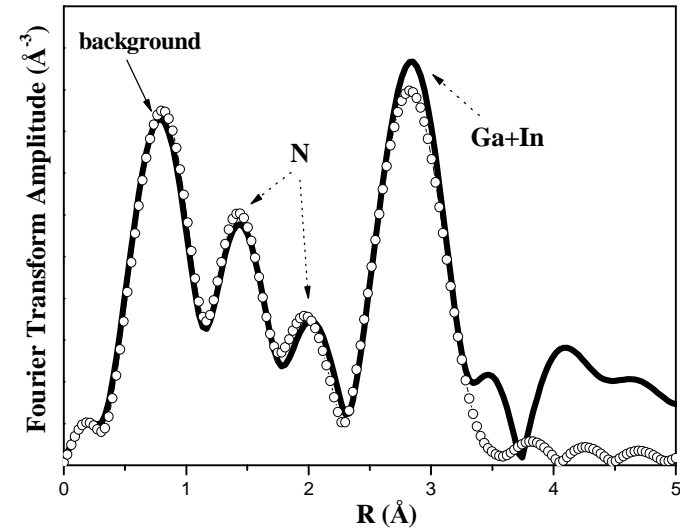
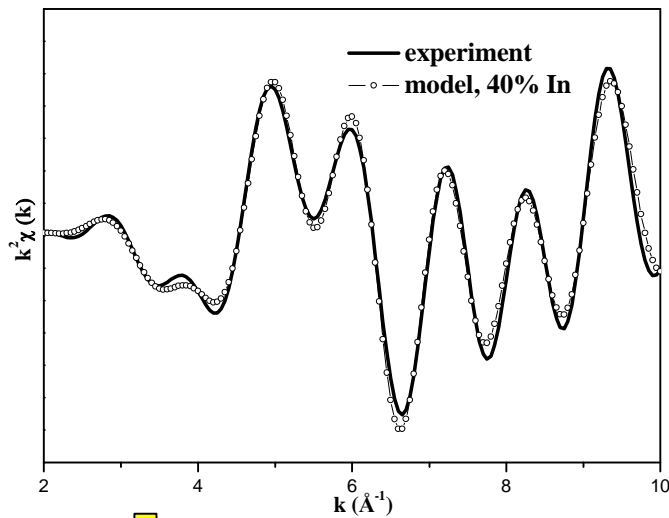
After fluorescence subtraction and fitting by  $I_{\text{smooth}}$

Resulting profiles of the oscillatory parts of the Ga scattering factors





## Fitting to the $\chi''(E)$ profile to ab-initio simulation results



close to the dot base ( $h=3.0$ ) the lattice is strained and it contains  $(20 \pm 5)\%$  In  
at the dot apex ( $h=2.9$ ) the lattice is relaxed and it contains  $(40 \pm 5)\%$  In

E. Piskorska et al., phys. stat. sol. (c) 3, 1662 (2006).



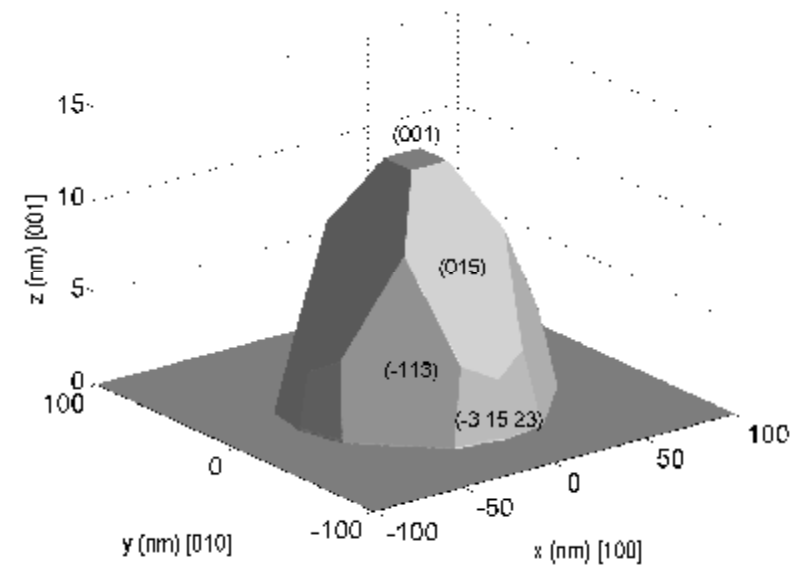
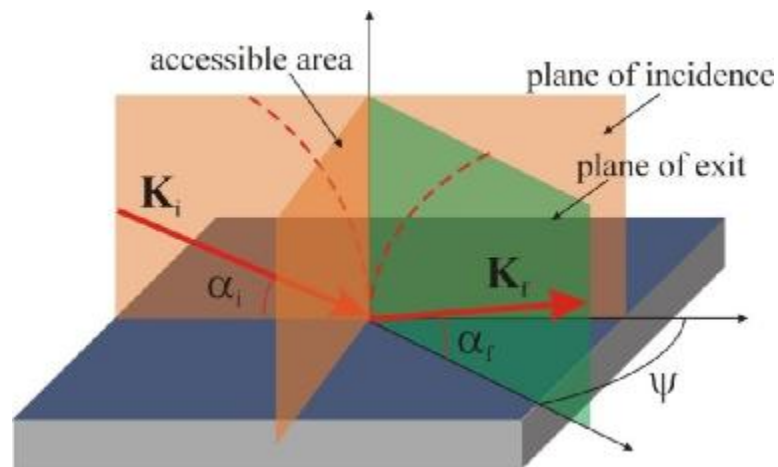
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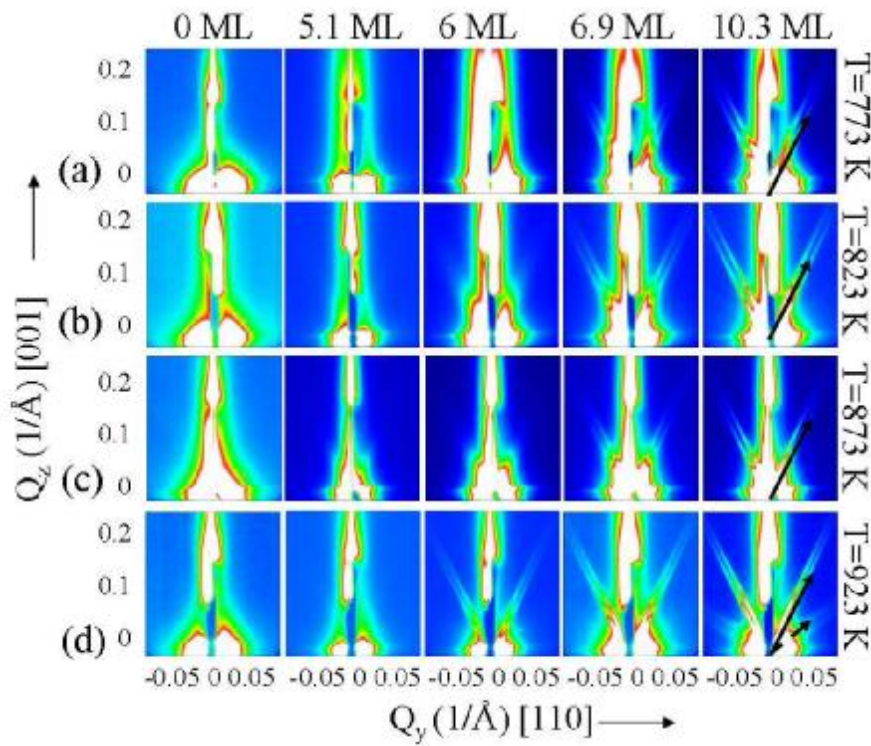
MBE growth chamber attached to the BM32 beamline at ESRF

## In-situ GISAXS study of the growth of Ge domes on Si(001)

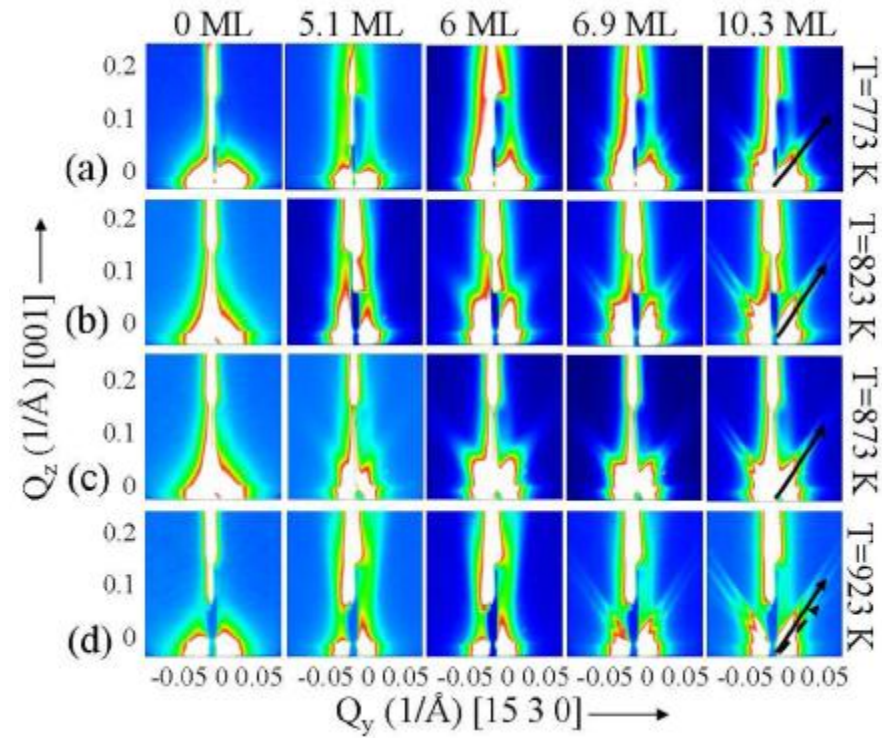
measurement M.-I. Richard, T. Schuelli, G. Renaud



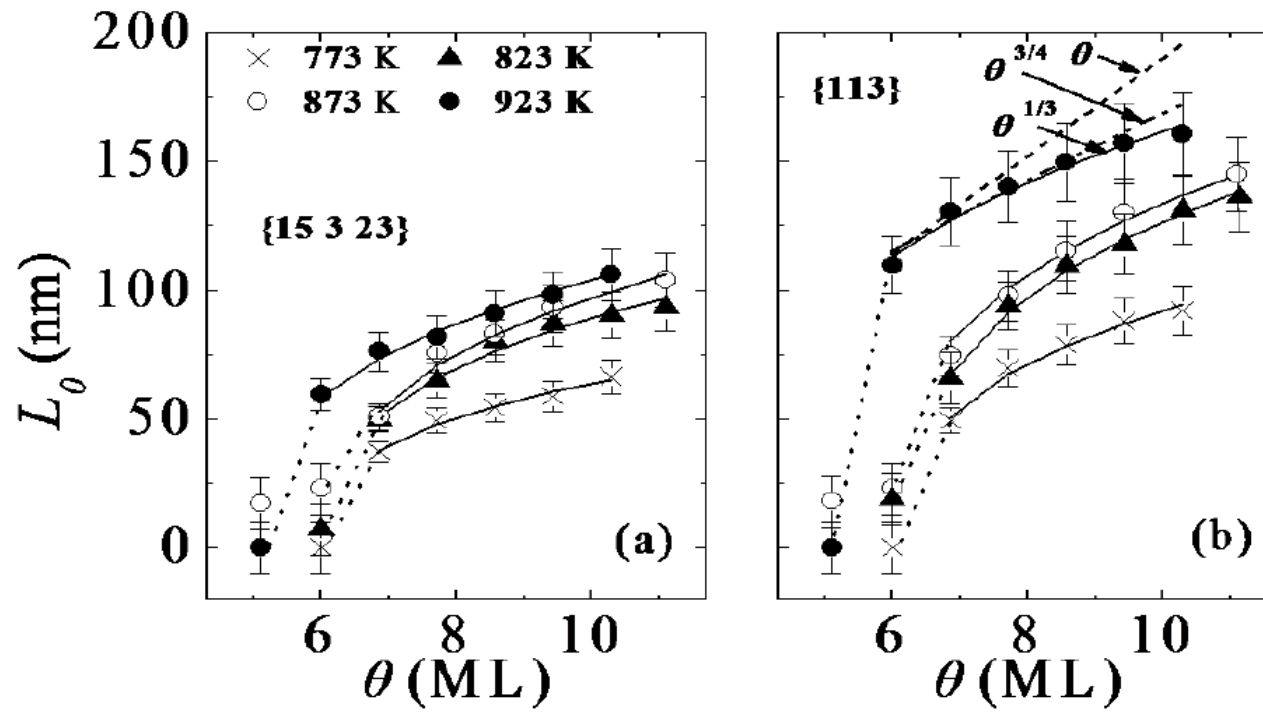




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 - - - - - → {111}



—→ {15 3 23}  
 - - - - - → {20 4 23}

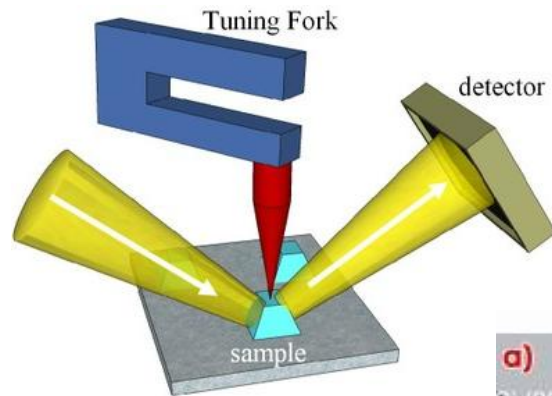


the dependence on the mean facet size on the Ge coverage  $\theta$

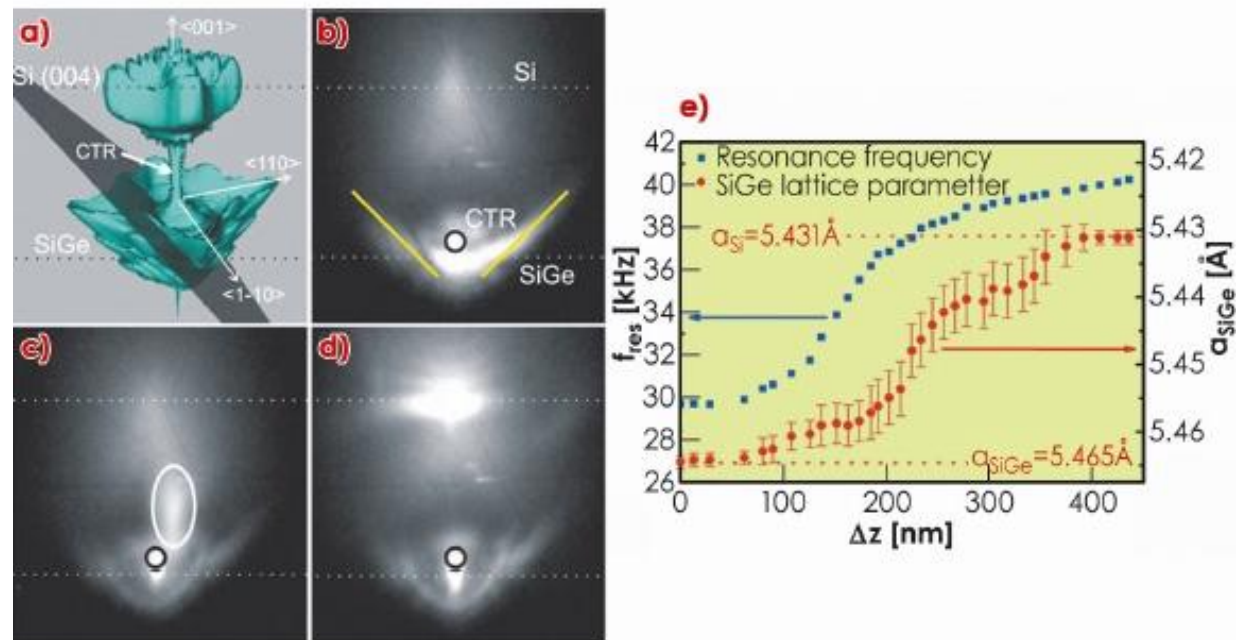
M.-I. Richard, T. U. Schüllli, G. Renaud, E. Wintersberger, G. Chen, G. Bauer, and V. Holý, Phys. Rev. B 80, 045313 (2009).



## Combining x-ray diffraction and AFM in-situ (ID01 at ESRF)



T. Scheler et al., Appl. Phys. Lett. 94, 023109 (2009)



From ESRF Highlights



1. What can be done by x-ray scattering?
2. Standard applications – rather boring
3. Iso-strain method & anomalous diffraction
4. DAFS
5. In-situ scattering
6. Nanobeams & coherent scattering
7. Summary

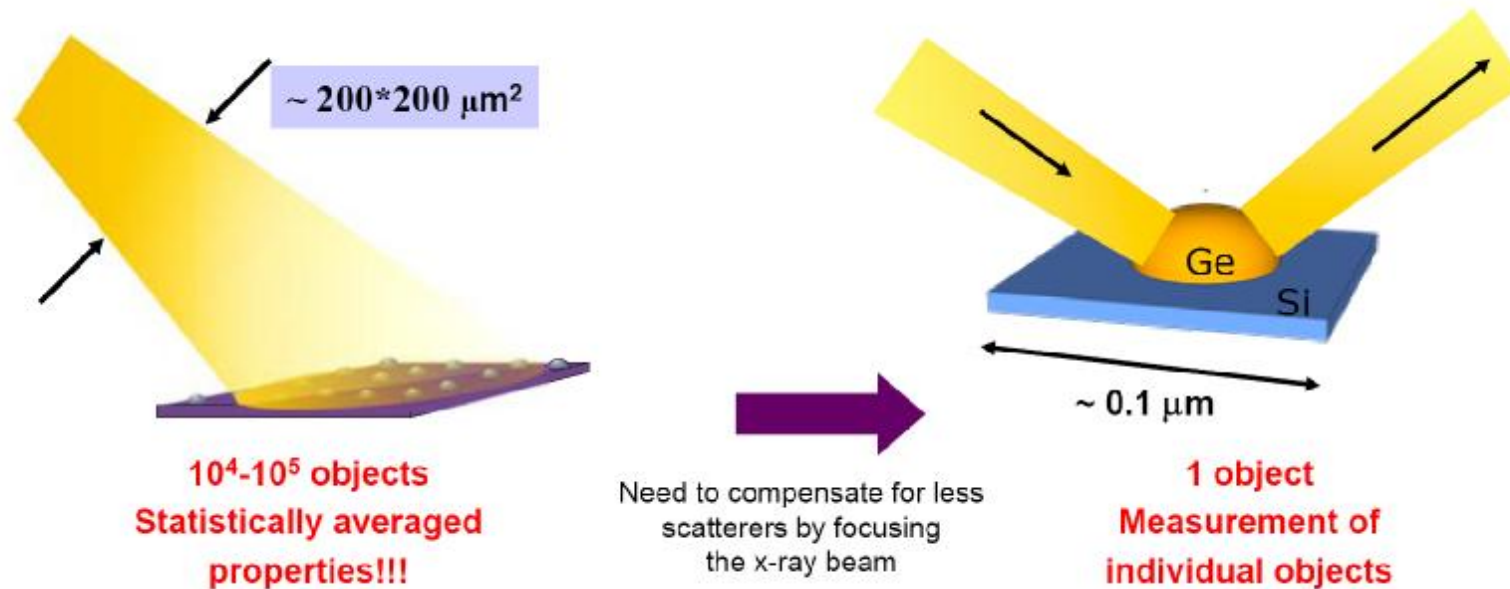


In previous examples, the measured intensity was an incoherent superposition of intensities scattered from a very large numbers of nano-objects

☺ the measured signal has a good statistical relevance

☹ details of the intensity distributions are smeared out, since the objects are not completely identical

To irradiate only one single object



courtesy of Cristian Mocuta, ESRF

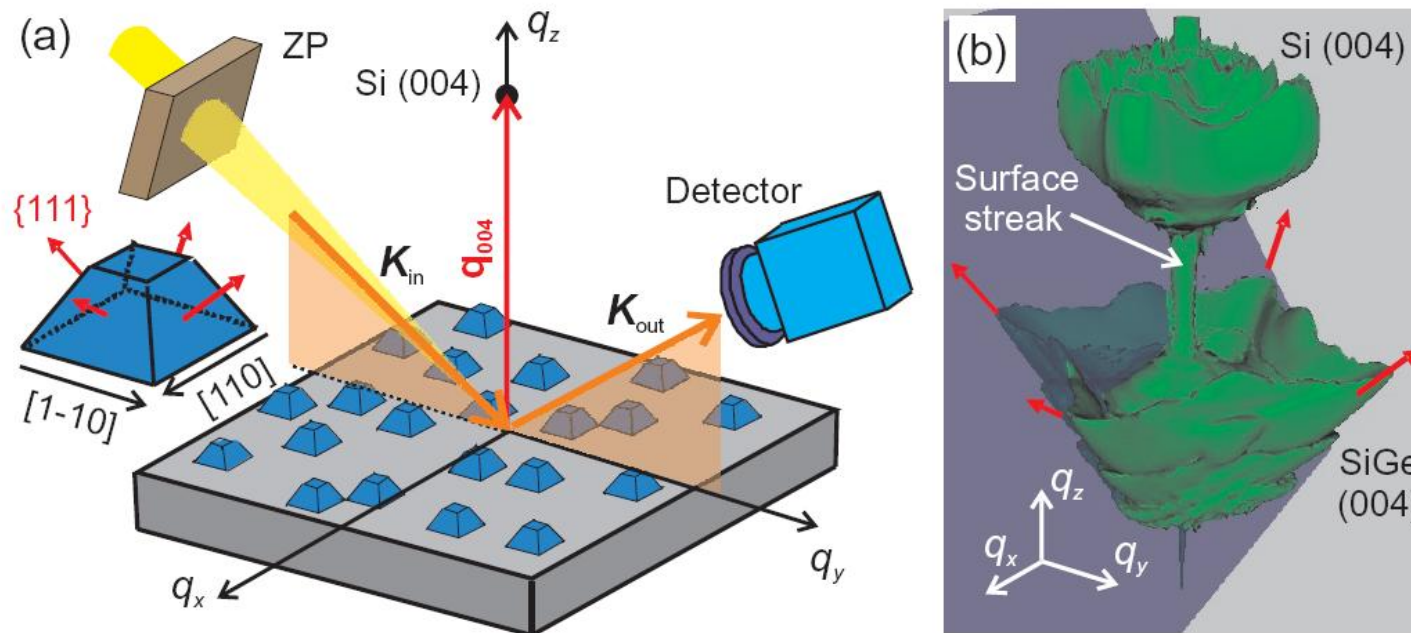


Then, the nano-object can be completely coherently irradiated

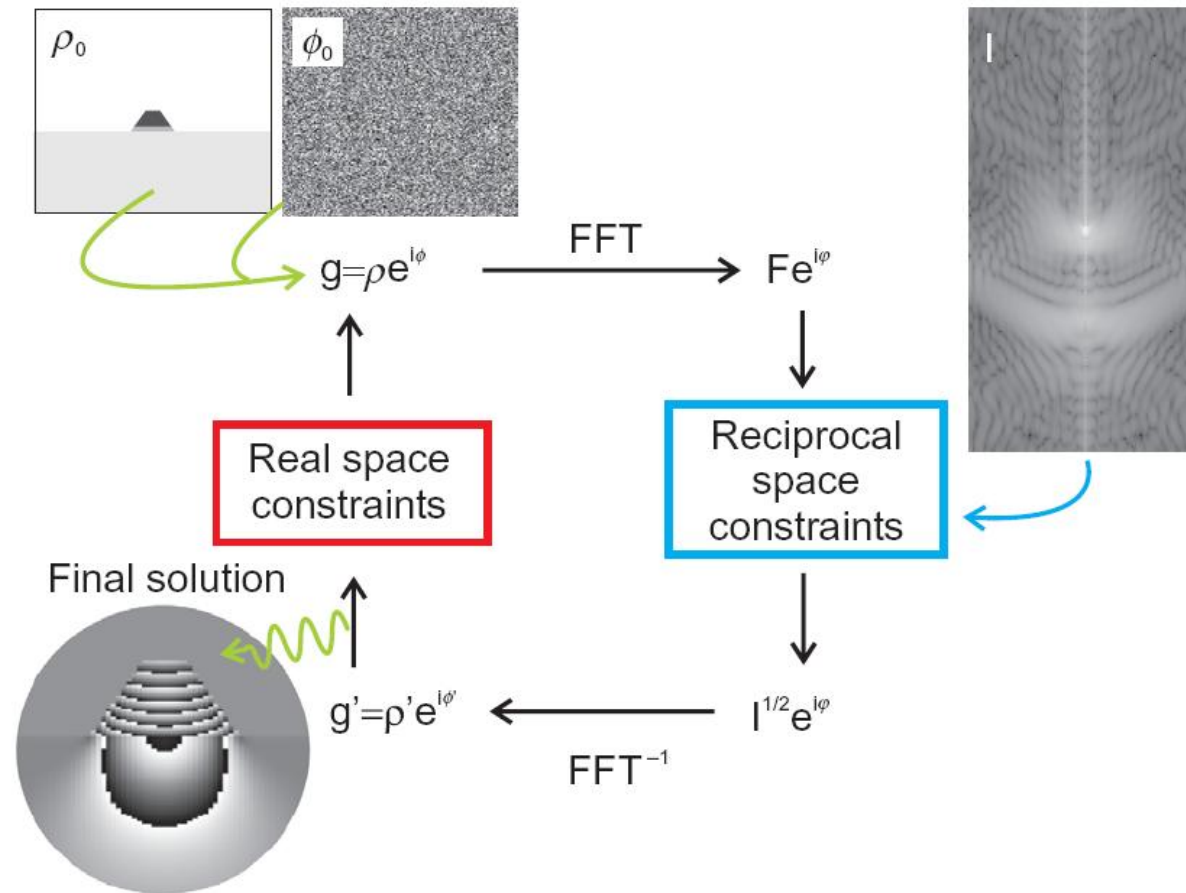
The phase-retrieval method makes it possible to reconstruct the shape and strain field directly without any structure model

Ana Diaz et al., New J. Phys. 12 035006, (2010)

### Ge dots on Si(001)



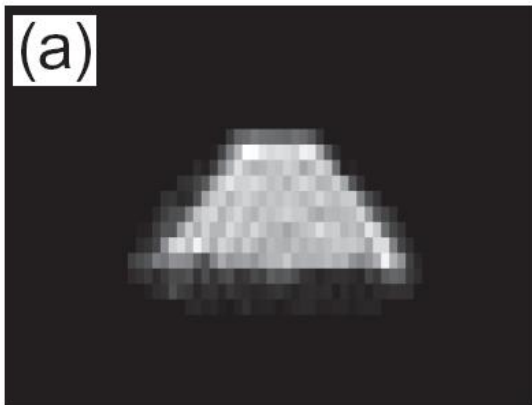
courtesy of A. Diaz, ESRF



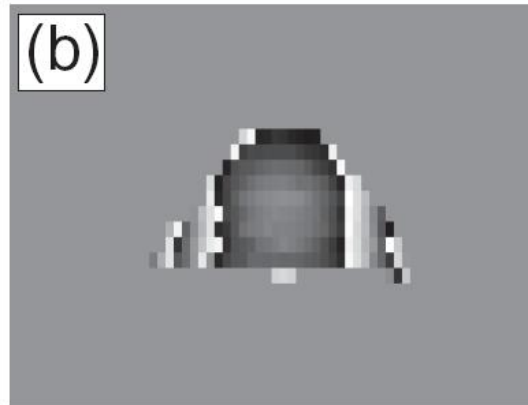
courtesy of A. Diaz, ESRF



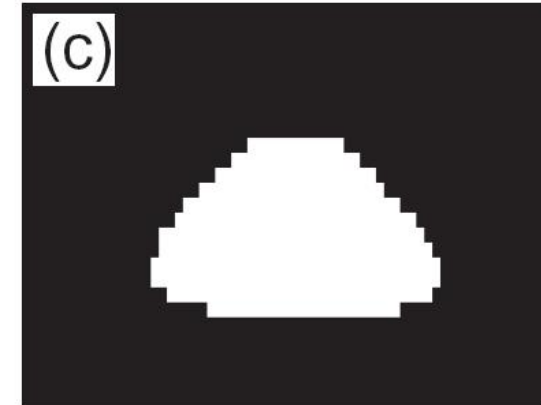
reconstructed amplitude



reconstructed phase



optimized real-space constraint



courtesy of A. Diaz, ESRF



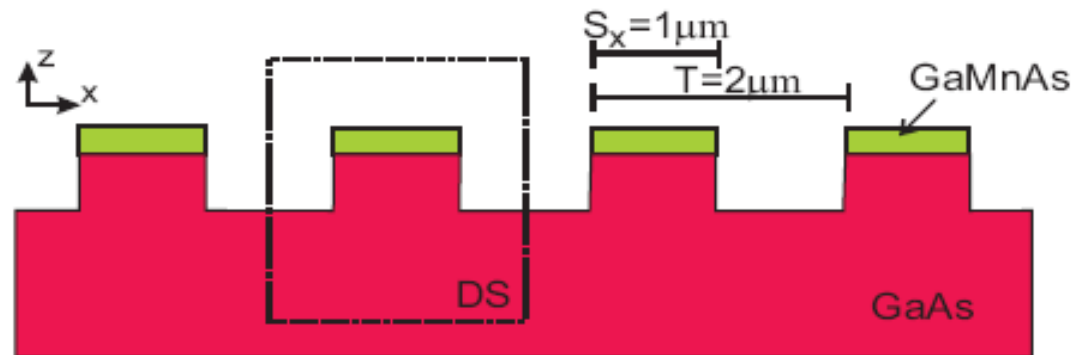


## GaMnAs wires in GaMnAs/GaAs(001) layers

Samples grown by MBE at Institute of Physics, ASCR Prague and patterned at University of Cambridge, UK

ID10@ESRF, phase retrieval algorithm by A. Minkevich, ANKA Karlsruhe

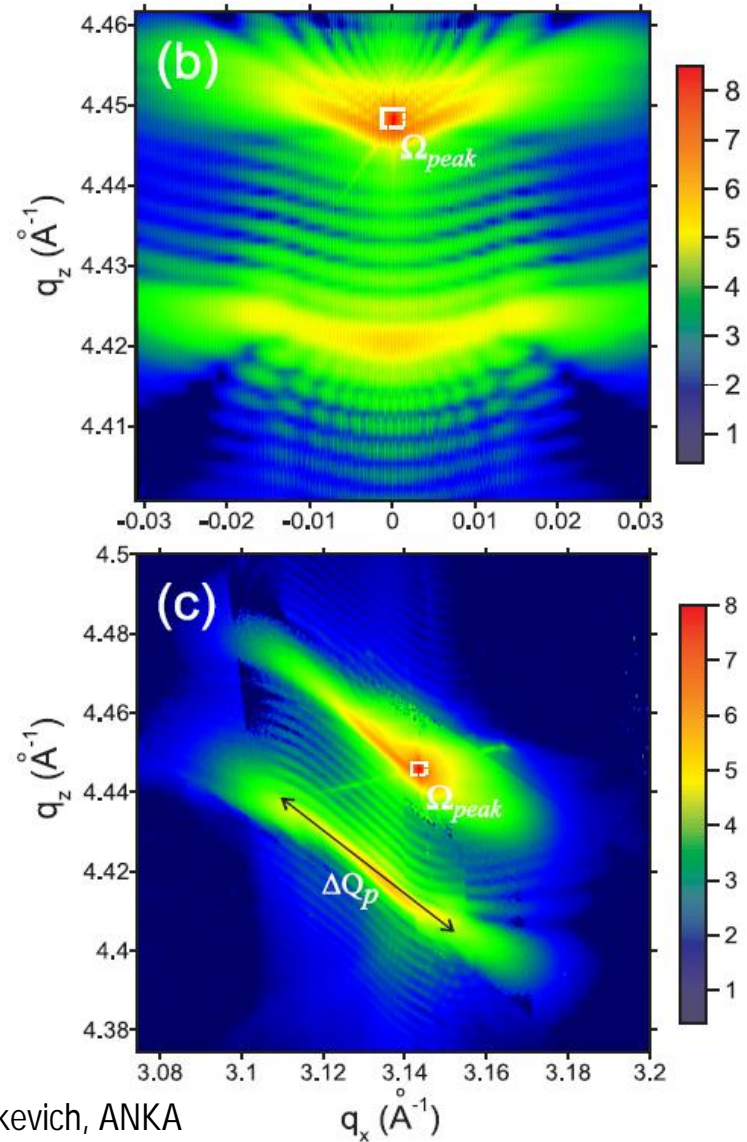
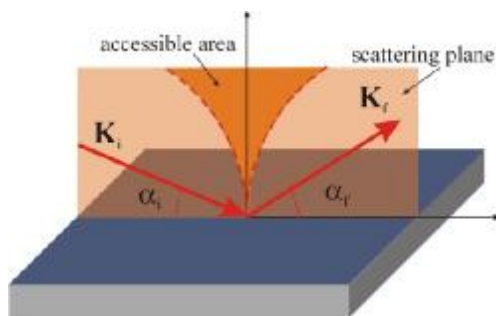
sample cross-section



courtesy of A. Minkevich, ANKA



measured intensity distribution, coplanar symmetric diffraction 004 (upper panel) and 224 (lower panel)

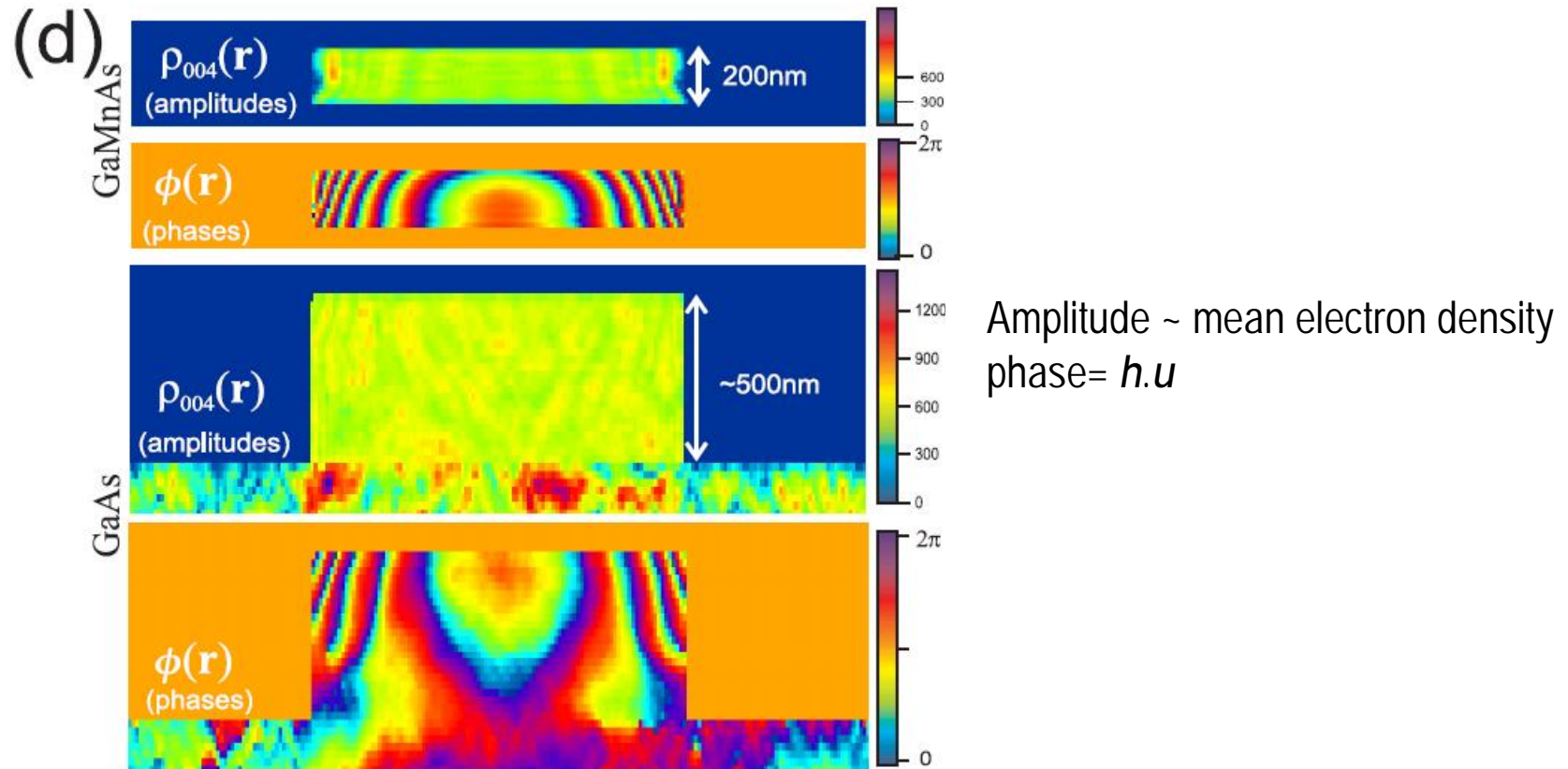


courtesy of A. Minkevich, ANKA

WSSR2011, Liptovský Ján



Amplitudes and phases of the field diffracted from the GaMnAs and GaAs volumes, diffraction 004

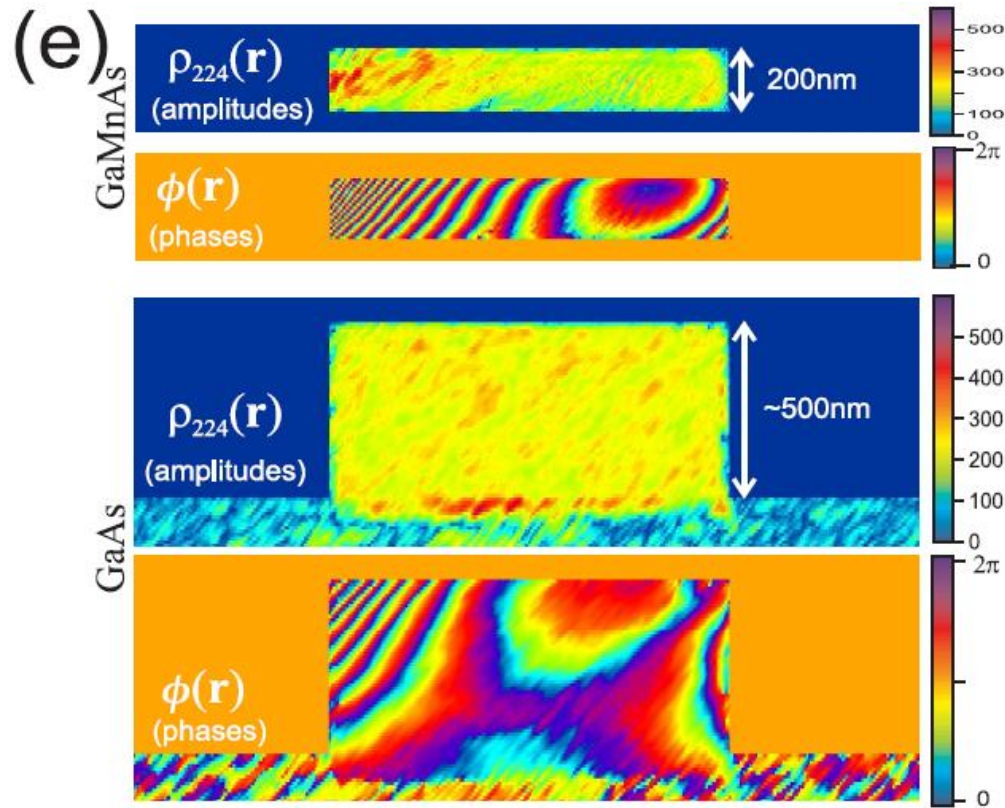


courtesy of A. Minkevich, ANKA

WSSR2011, Liptovský Ján



Amplitudes and phases of the field diffracted from the GaMnAs and GaAs volumes, diffraction 224



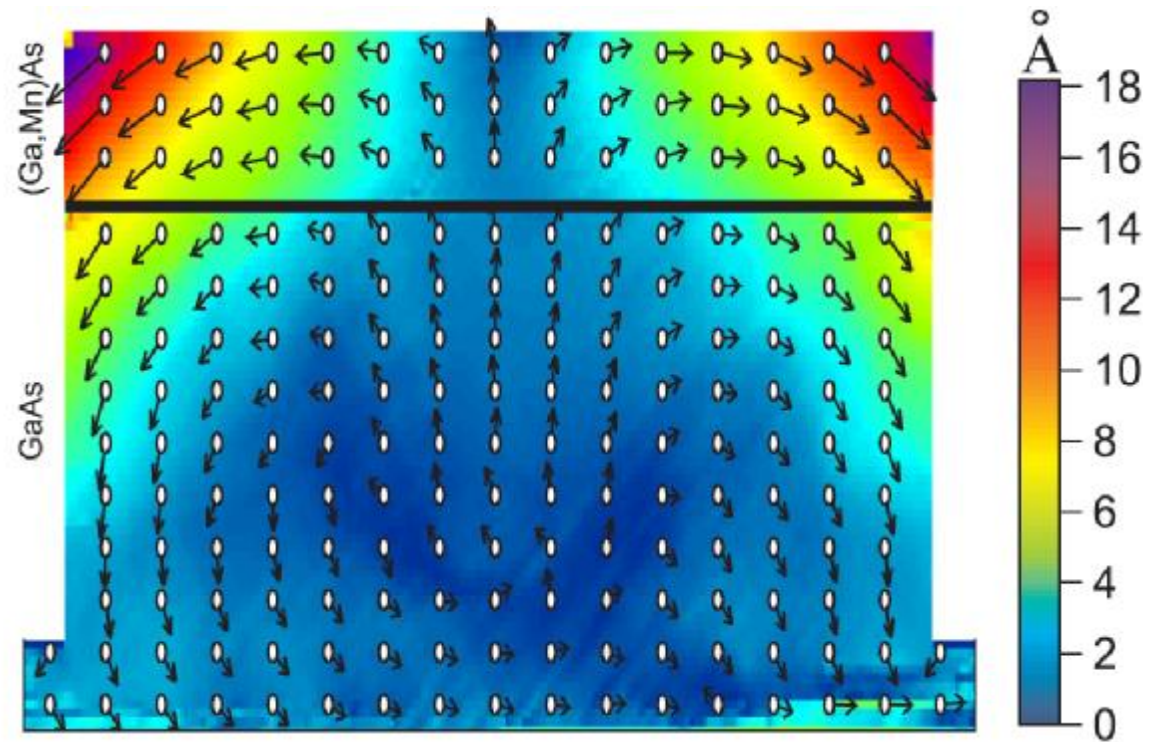
Amplitude ~ mean electron density  
phase =  $h \cdot u$

courtesy of A. Minkevich, ANKA

WSSR2011, Liptovský Ján



Displacement field reconstructed from the x-ray data



A. Minkevich et al., submitted

courtesy of A. Minkevich, ANKA

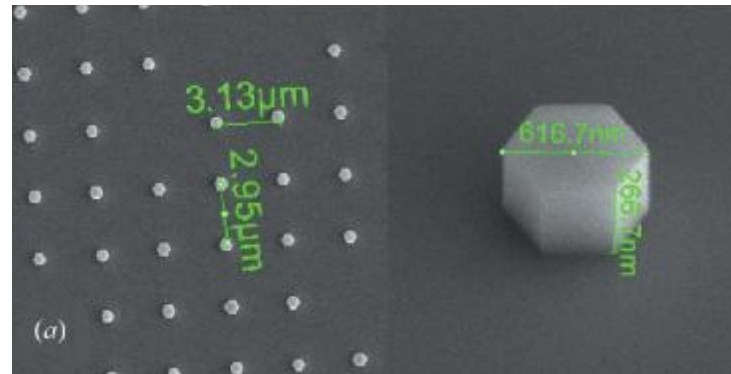
WSSR2011, Liptovský Ján



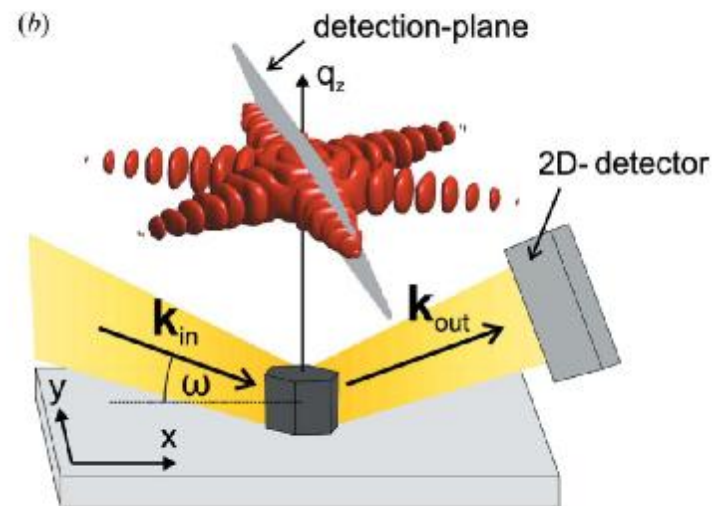
Another example: **coherent diffraction from single nanorods**

A. Biermanns et al., J. Synchr. Rad. 16, 796 (2009)

V. Favre-Nicolin et al., New Journal of Physics 12, 035013 (2010).

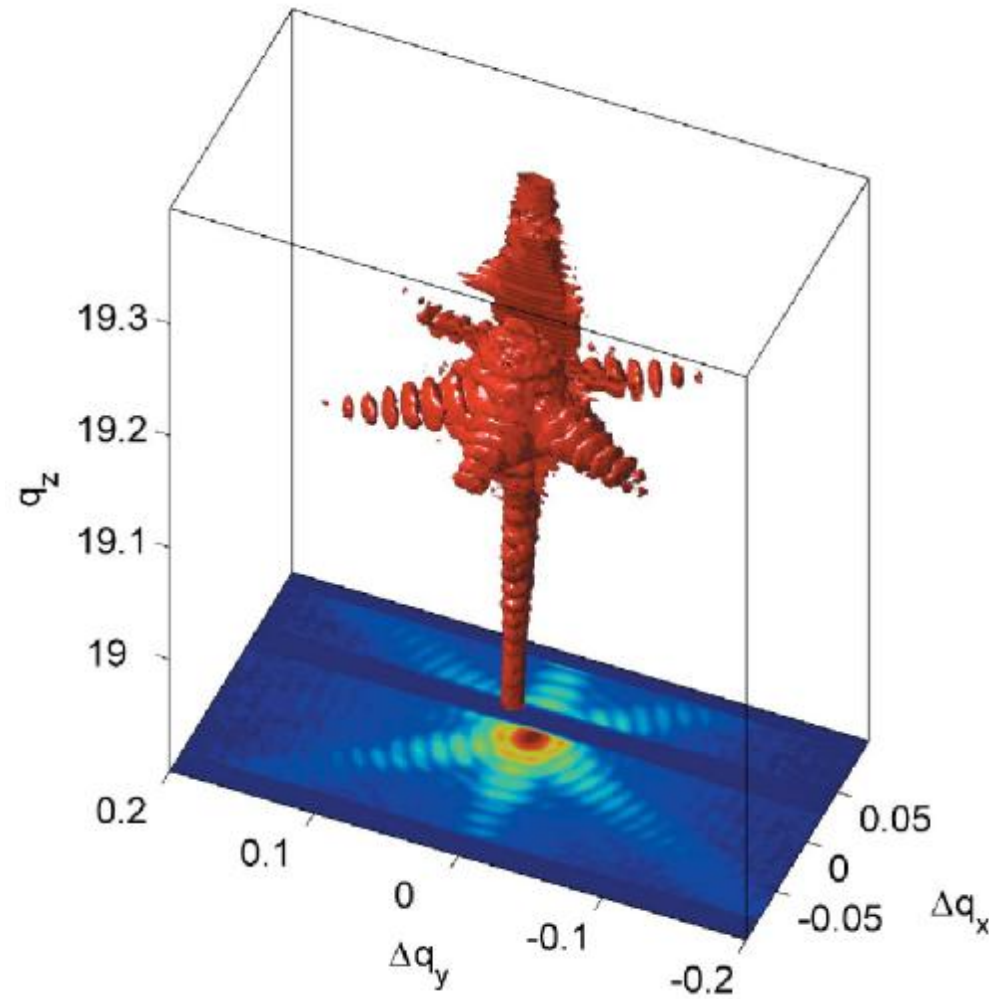


A. Biermanns et al.:



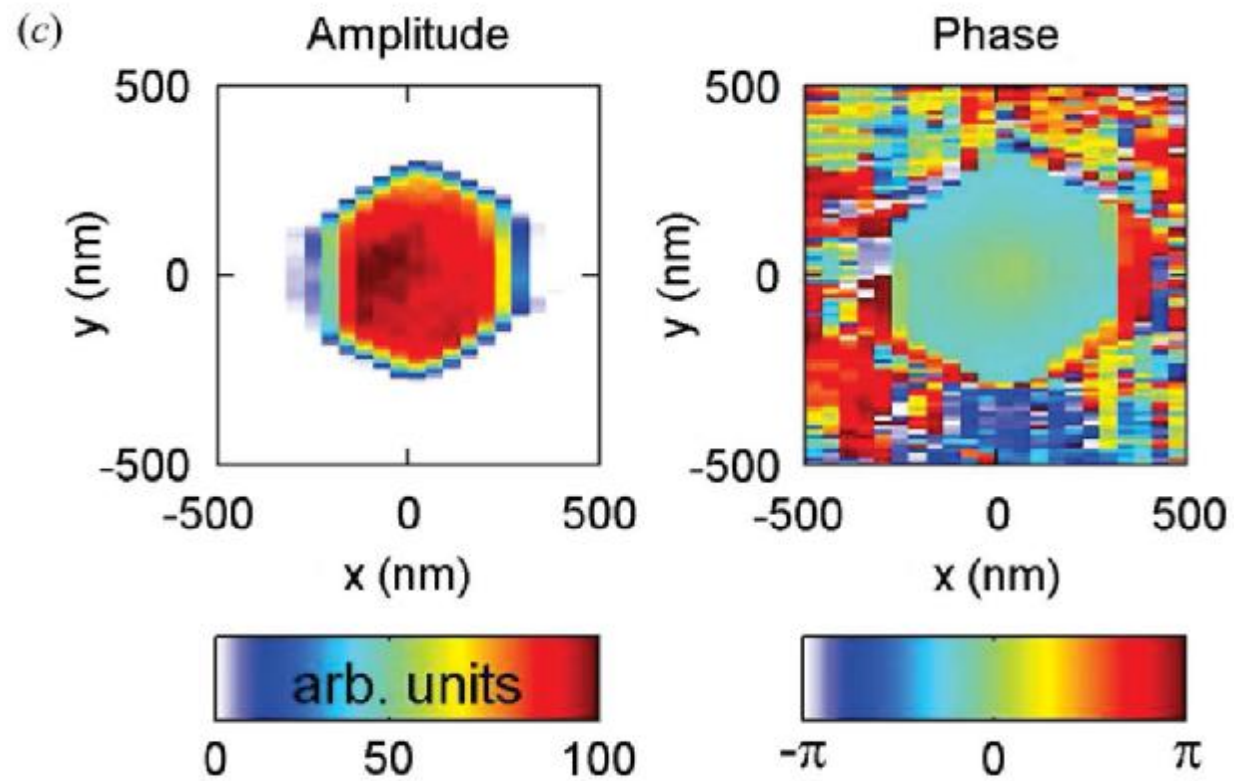


## 3D distribution of coherently scattered intensity in reciprocal space





## Results of the phase-retrieval algorithm







1. What can be done by x-ray scattering?
2. Standard applications – rather boring
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5. In-situ scattering
6. Nanobeams & coherent scattering

## 7. Summary



## Recent trends using synchrotron radiation:

1. Multiwave anomalous diffraction (MAD) method
2. In-situ studies (during deposition, annealing, oxidation, elastic deformation etc)
3. Investigation of a single nanoobject
4. Phase-retrieval method

## Substantial progress in laboratory instrumentation:

1. **THANK YOU FOR YOUR KIND ATTENTION**
2. In-situ measurements (annealing, oxidation etc.)
3. Better detectors (2D detectors with extreme dynamical range, very low noise but for a very high price ☹)

Projects: KAN400100652, EU project 214499 NAMASTE

8th Autumn School on X-ray Scattering from Surfaces and Thin Layers Smolenice 2011 - Mozilla Firefox

Soubor Úpravy Zobrazení Historie Záložky Nástroje Nápověda

http://cmd.karlov.mff.cuni.cz/Smolenice\_2011/

Nejnavštěvovanější Jak začít Přehled zpráv

31 Kalendář Google Gmail - Doručená pošta - vasekhol... Program - Winter School of Synchr... 8th Autumn School on X-ray Sca... XPS File Extension - Open .XPS files

**Smolenice 2011**  
*Smolenice Castle, Slovak Republic, October 4 - 7 2011*

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- ✓ About castle
- Countdown

**8<sup>th</sup> Autumn School on X-ray Scattering from Surfaces and Thin Layers**  
4.-7. October 2011  
Smolenice Castle, Slovakia

The autumn school is the 8<sup>th</sup> in row on the topic of high-resolution X-ray measurements and interpretation from thin films and surfaces. As the last one, held in 2007, the conference is organized on a romantic castle in Smolenice close to Bratislava (Slovakia). In order to welcome students and young scientists, we are planning to keep to costs low as possible again. The conference site also allows us to invite a number of Eastern Europe scientists as well. As in previous years, graduate and post-graduate students and other young scientists are called to present their actual and often still unpublished results. Professors and other "Chiefs" are welcome but they will not give talks. Because no proceeding will appear the discussion can be very lively and fruitful. The school is organized in the conference site of the Slovak Academy of Sciences in Smolenice (Slovakia).

Hotovo

Windows taskbar: Newsletter\_Holy - ... Total Commander 7... 8th Autumn School ... Microsoft PowerPoi...

System tray: EN < > 21:16