

WINTER SCHOOL OF SYNCHROTRON RADIATION

**31.01. – 04.02. 2011 Liptovský Ján,
Slovakia**

Š. Michalik: X-ray Diffraction and X-ray Absorption Spectroscopy in the Study of Materials with Disordered Structure

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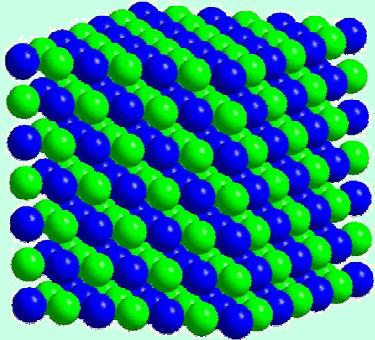
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Múdrost' minulosti – poznatky prítomnosti – vzdelanie budúcnosti.



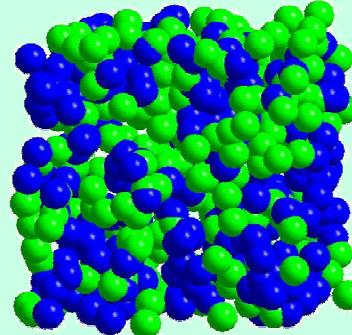
Materials with Disordered Structure

Ordered (crystalline) structure



- periodic structure
- long range order
- symmetry

Disordered (amorphous) structure



- no periodicity
- short range order
- no symmetry

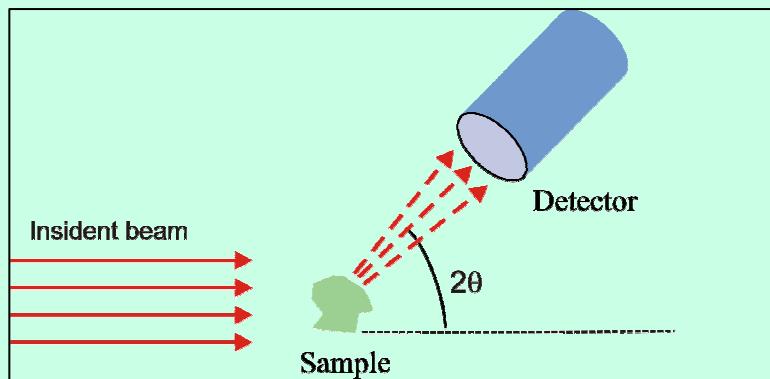
Real life example



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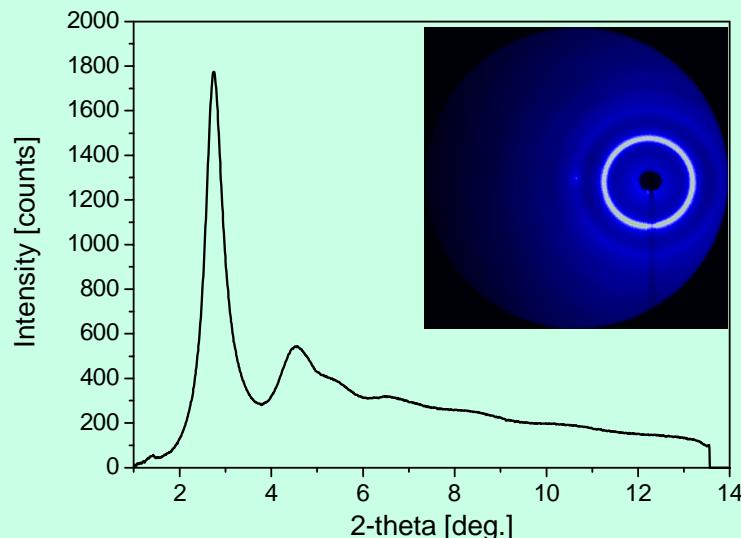
X-ray diffraction experiments



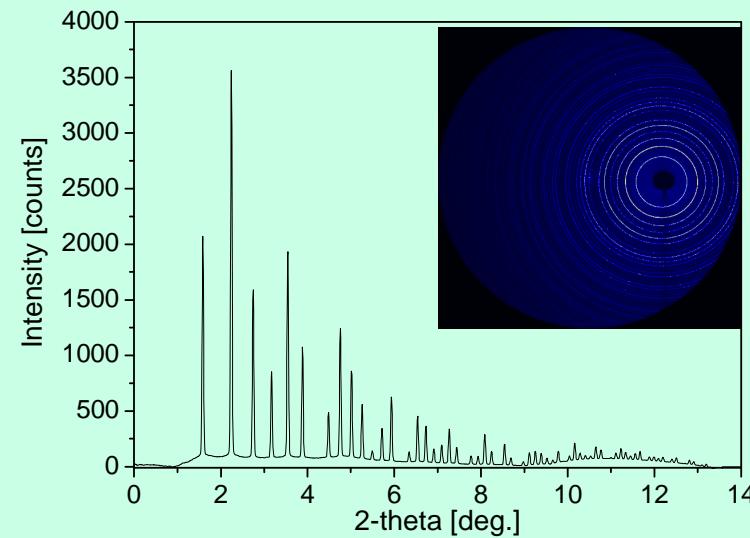
The main idea is to measure the variation in scattered intensity due to interference effects as a function of the (diffracted) angle

$$Q = \frac{4\pi}{\lambda} \sin(\theta) \quad - \text{a wave transfer vector}$$

Amorphous material



Crystalline material



Experimental station BW5

BW5 is dedicated to X-ray scattering experiments at energies between **60 and 150 keV**.



Present applications are:

- charge density studies of single crystals
- structural phase transitions
- *analysis of the pair distribution function of amorphous material and liquids*
- texture analysis of powder materials

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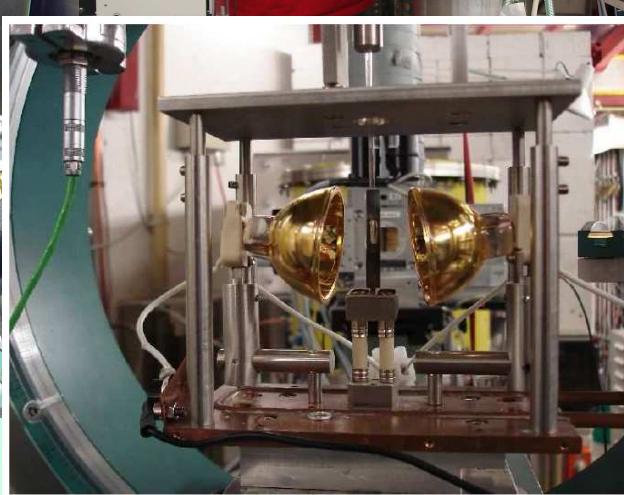
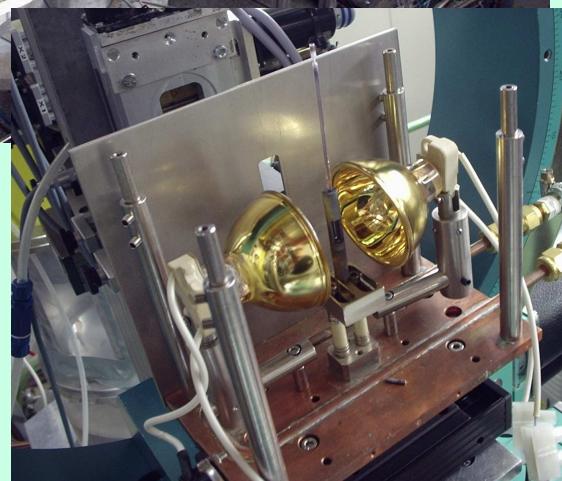
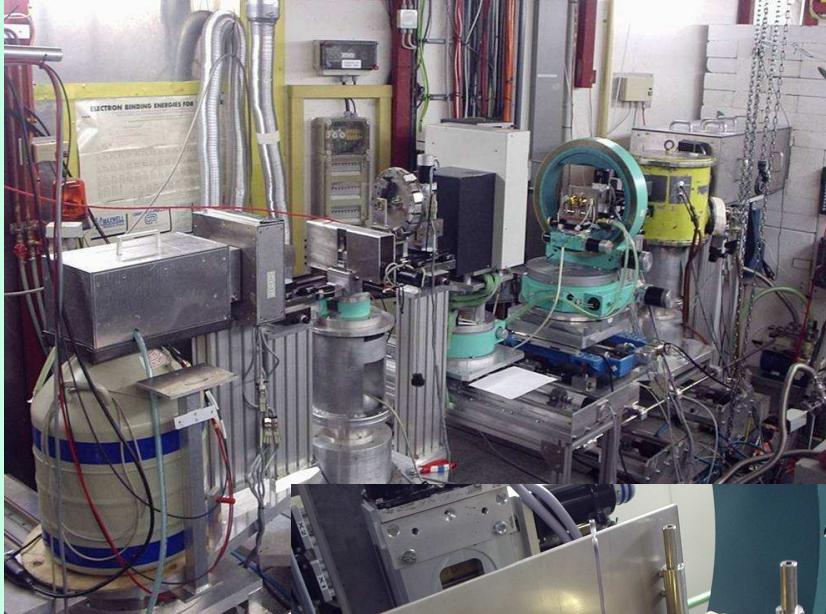


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Experimental station BW5



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Photon energy, wavelength and penetration depth

$$\lambda[\text{\AA}] = \frac{hc}{\epsilon} = \frac{12.398}{\epsilon[\text{keV}]}$$

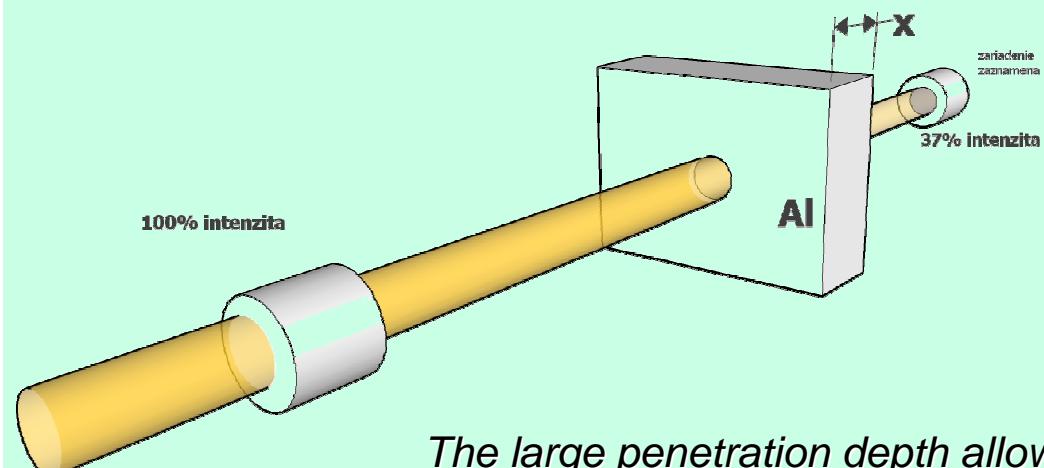
$$\text{\AA} = 10^{-10} \text{ m}$$

$$\text{eV} = 1.602 \times 10^{-19} \text{ J}$$

200 eV = 62 Å soft x-rays

8.05 keV = 1.54 Å standard x-ray Cu lamp

100 keV = 0.124 Å hard x-rays



200 eV $x = 0.1 \mu\text{m}$

9 keV $x = 0.103 \text{ mm}$

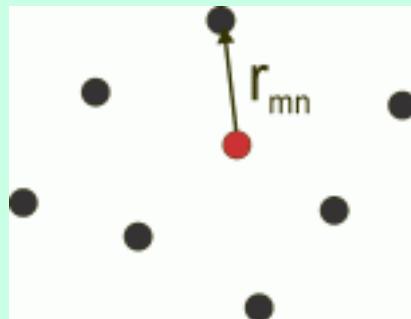
100 keV $x = 2.2 \text{ cm } !!!$

The large penetration depth allows the investigation of bulk materials and complex sample environments.



Theory of diffraction on disordered materials

The scheme of amorphous alloys



$$I(\mathbf{Q}) = \sum_m^N f_m(\mathbf{Q}) e^{i\mathbf{Q} \cdot \mathbf{r}_m} \sum_n^N f_n(\mathbf{Q}) e^{-i\mathbf{Q} \cdot \mathbf{r}_n}$$

$$S(Q) \equiv \frac{I(Q)/N - f^2(Q)}{f^2(Q)} = 1 + \int_0^\infty \underbrace{4\pi r [\rho(r) - \rho_0]}_{D(r)} \frac{\sin(Qr)}{Q} dr$$

where $\rho(r)$ and ρ_0 are the local and average atomic number densities. $D(r)$ – the reduced pair distribution function

Fourier transformation

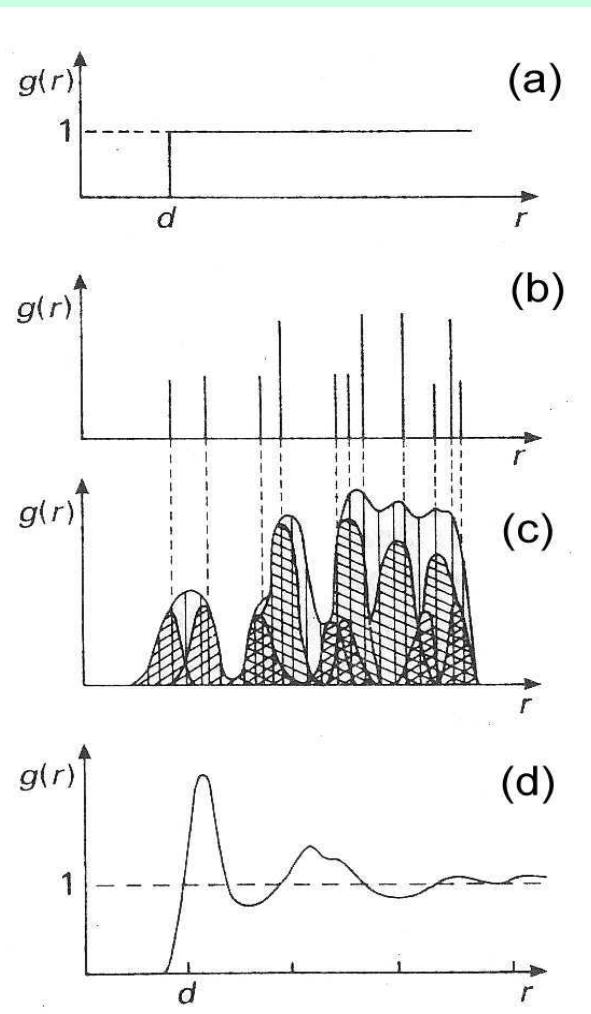
$$S(Q) = \int_0^\infty D(r) \frac{\sin(Qr)}{Q} dr$$

$$D(r) = \frac{2}{\pi} \int_0^\infty Q [S(Q) - 1] \sin(Qr) dQ$$

We are able **directly** to perform the real space analysis (in the space of interatomic distances)



Pair distribution functions



gas

ideal
crystal

real
crystal

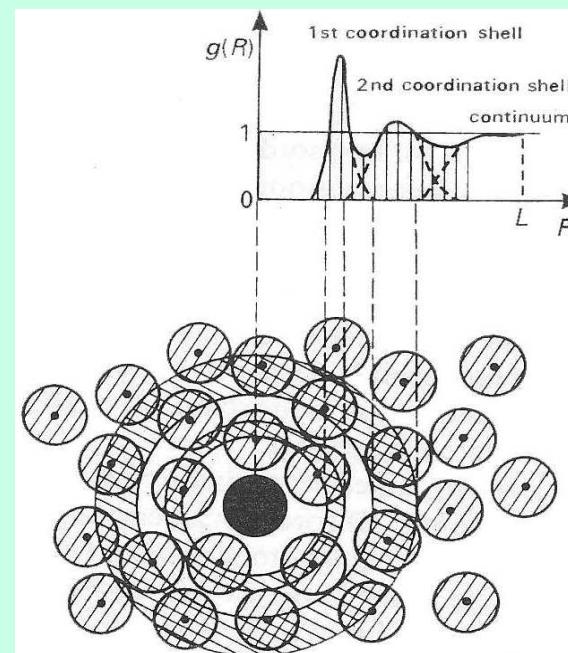
amorphous
alloy

The reduced distribution function $D(r)$

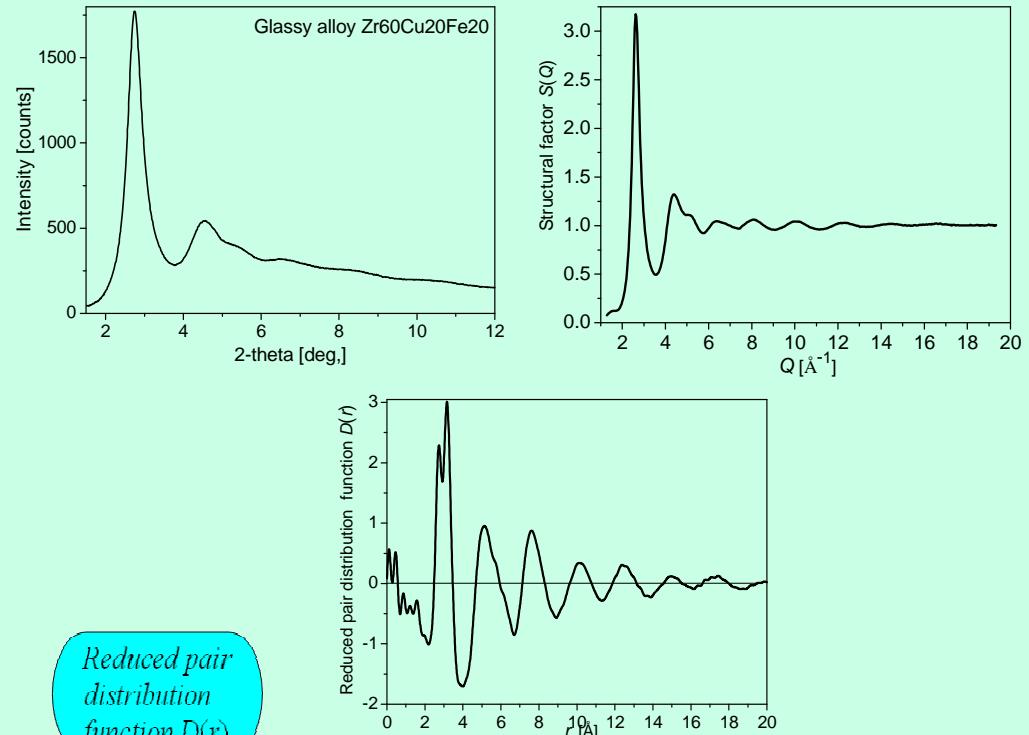
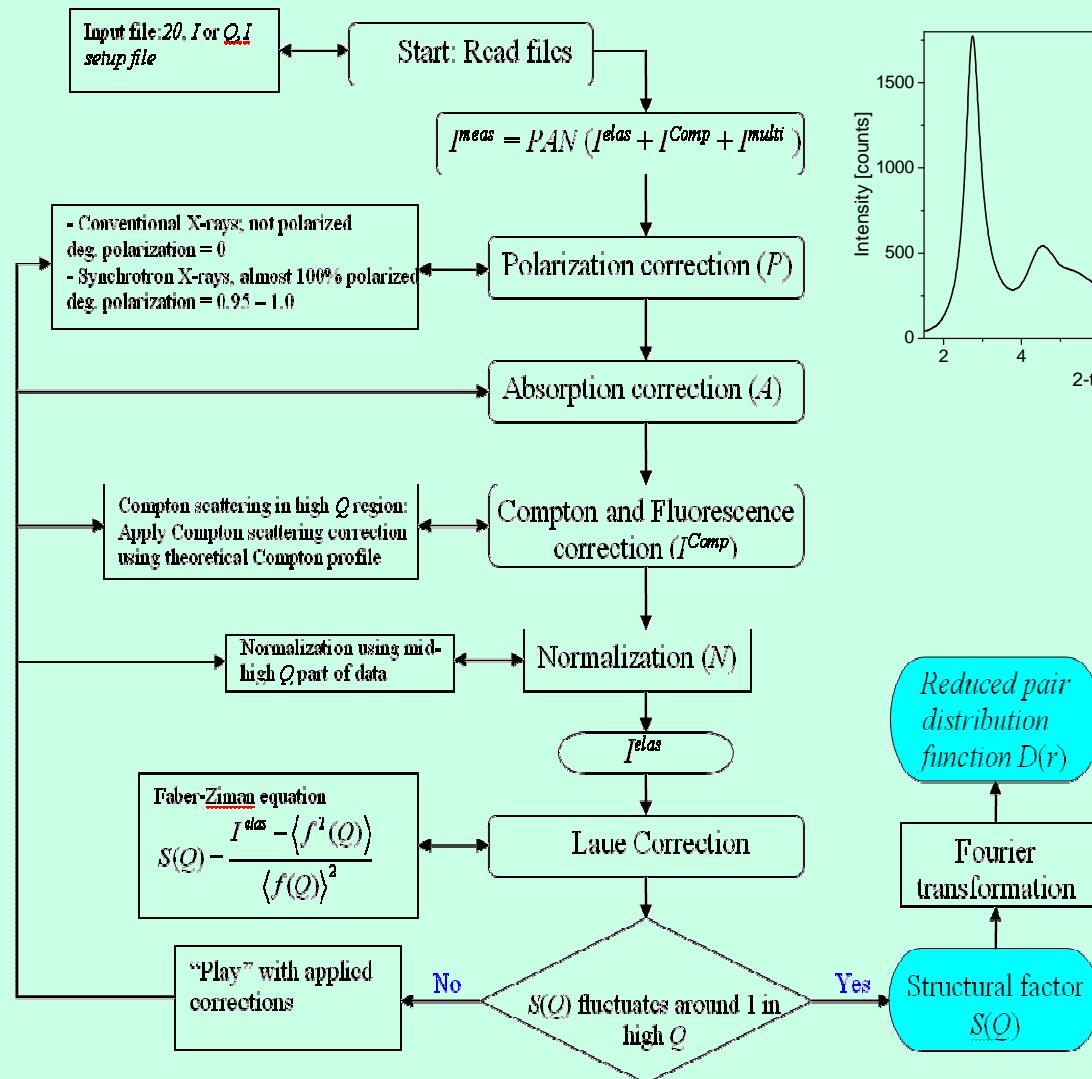
$$D(r) = 4\pi r[\rho(r) - \rho_0]$$

The atomic pair distribution function $g(r)$

$$g(r) = \rho(r) / \rho_0$$



The structural factor extraction



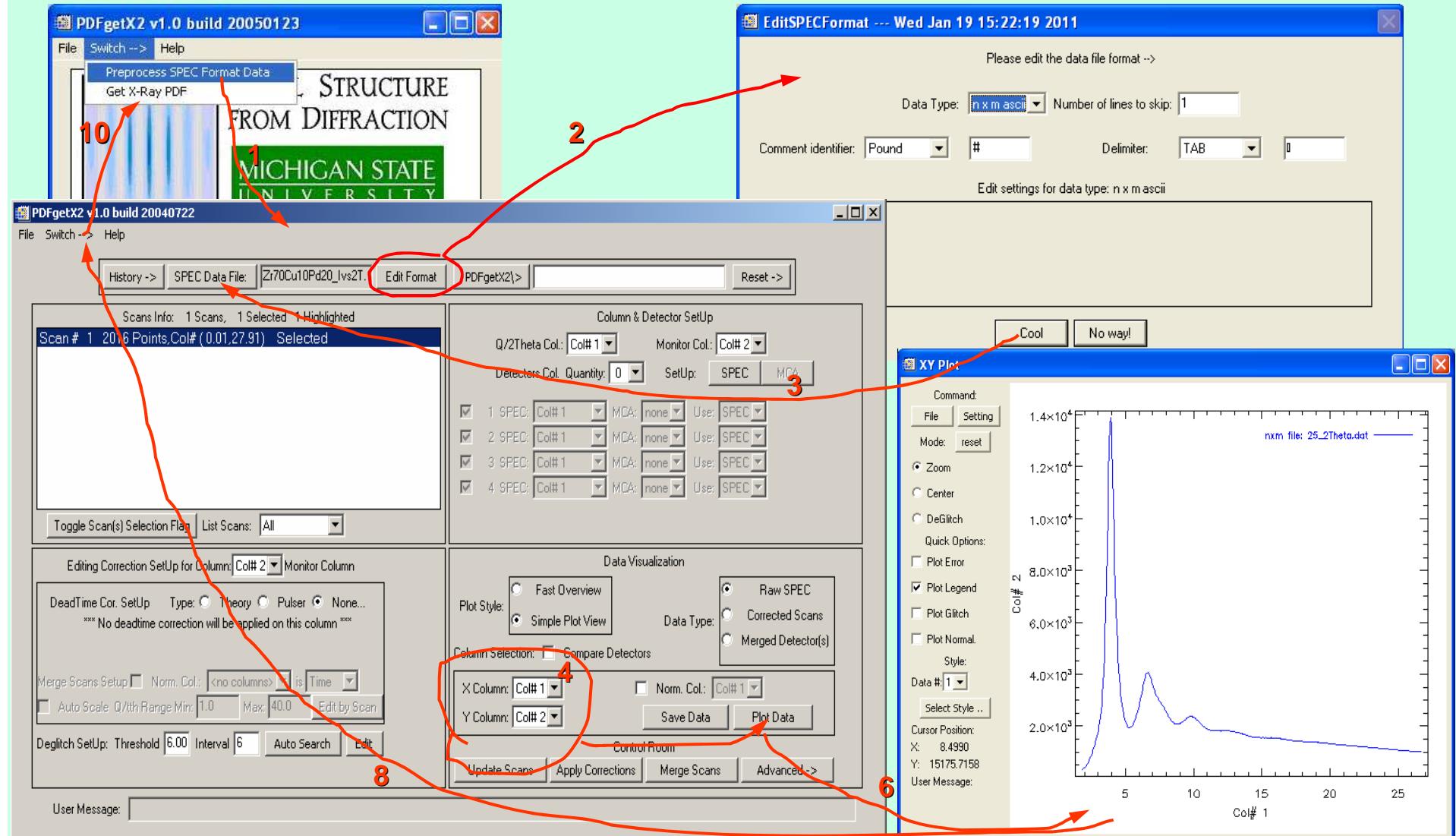
How to do it?

Software PDFGetX2

<http://www.pa.msu.edu/cmp/billinge-group/programs/PDFgetX2/>



Preprocess a data file



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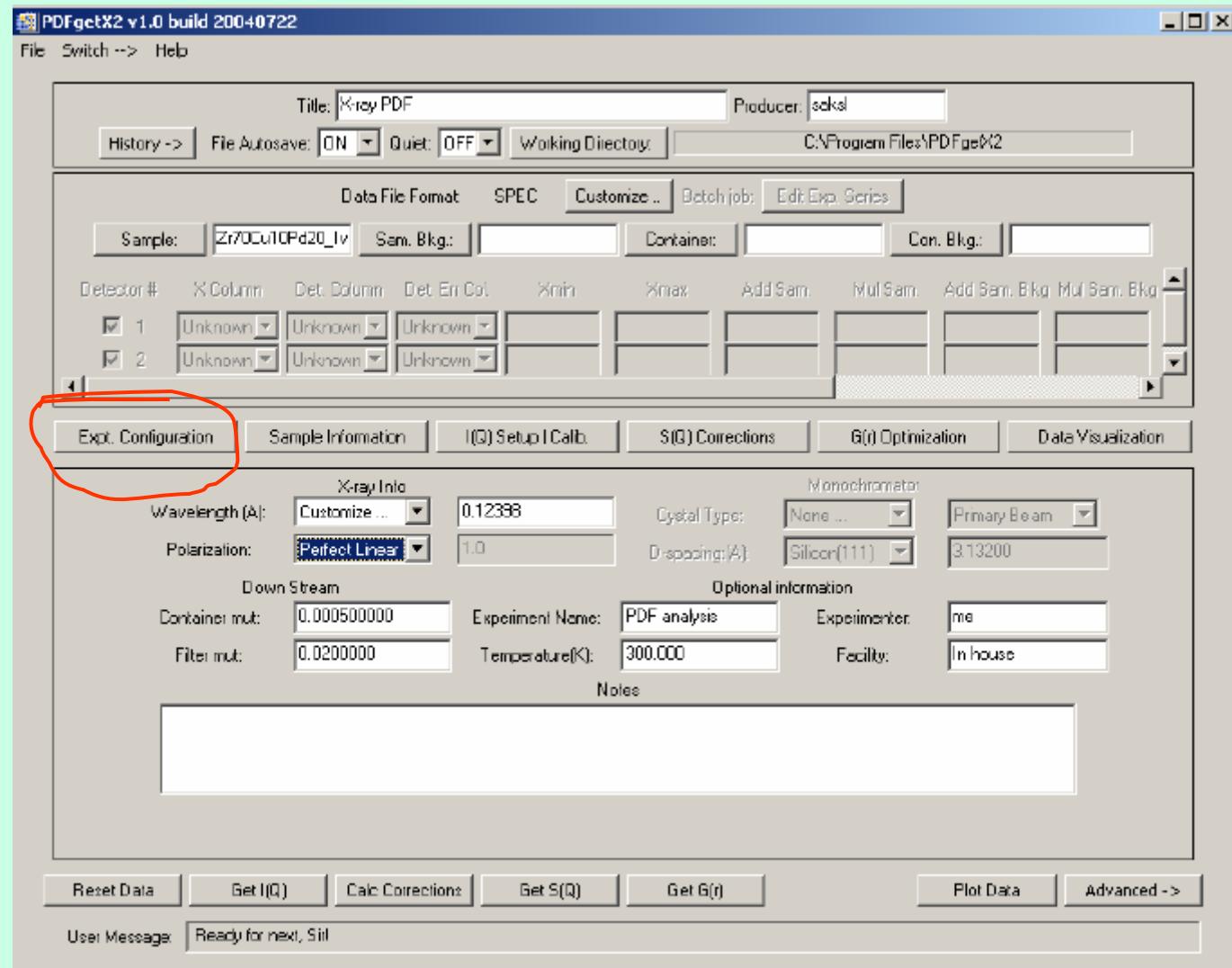


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Setup an exp. configuration



Define wavelength
and polarization.

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Setup the sample information

The screenshot shows the PDFgetX2 software interface. The 'Sample Information' tab is selected. On the right, there is a grid of elements for selection. A red arrow points from the title bar to the 'Sample Information' tab. A red circle highlights the 'Sample Information' tab. A third red arrow points from the 'Sample Information' tab to the element selection grid.

Sample Information Tab Content:

- Add/Remove Atoms:** Buttons for 'Add' and 'Remove' atoms.
- Sample Geometry:** Set to 'Cylindrical Capillary'.
- Attenuation Coef. (mut):** 0.78697000.
- Element Table:** Shows atomic symbols, Z values, Valence, Fraction, User f1, User f2, and User macrof for Zr, Cu, and Pd.
- Optional Info.:** Number Density: 0.0485000, Thickness/Diameter(mm): 2.00000, Packing Fraction: 0.500000, Theoretical mut: 0.786970.
- Buttons:** Reset Data, Get I(Q), Calc Corrections, Get S(Q), Get G(r), Plot Data, Advanced ->.
- User Message:** [Mon Jun 19 09:41:08 2006] Start to calculate corrections ... successfully finished!

Element Selection Grid (Right Side):

- Grid Labels:** H, Li, Be, Na, Mg, B, C, N, O, F, Ne, Al, Si, P, S, Cl, Ar, K, Ca, Sc, Ti, V, Cr, Mn, Fe, Co, Ni, Cu, Zn, Ga, Ge, As, Se, Br, Kr, Rb, Sr, Y, Zr, Nb, Mo, Tc, Ru, Rh, Pd, Ag, Cd, In, Sn, Sb, Te, I, Xi, Cs, Ba, Lu, Hf, Ta, W, Re, Os, Ir, Pt, Au, Hg, Tl, Pb, Bi, Po, At, Rn, Fr, Ra, Lr, Rf, Db, Sg, Bh, Hs, Mt, uun, uuu, uub, uut, uuq, uup, uuh, uus, uuo.
- Lanthanoids:** La, Ce, Pr, Nd, Pm, Sm, Eu, Gd, Tb, Dy, Ho, Er, Tm, Yb.
- Actinoids:** Ac, Th, Pa, U, Np, Pu, Am, Cm, Bk, Cf, Es, Fm, Md, No.

Selected Elements: Zr, Cu, Pd

Buttons: Done!, Cancel!

Define the sample chemical composition, the sample geometry, the number density and other parameters.

Push “Reset Data”, “Get I(Q)” and then “Calc Corrections” and so the “theoretical mut” will be calculated. The calculated value put into “Attenuation Coef. (mut)”.

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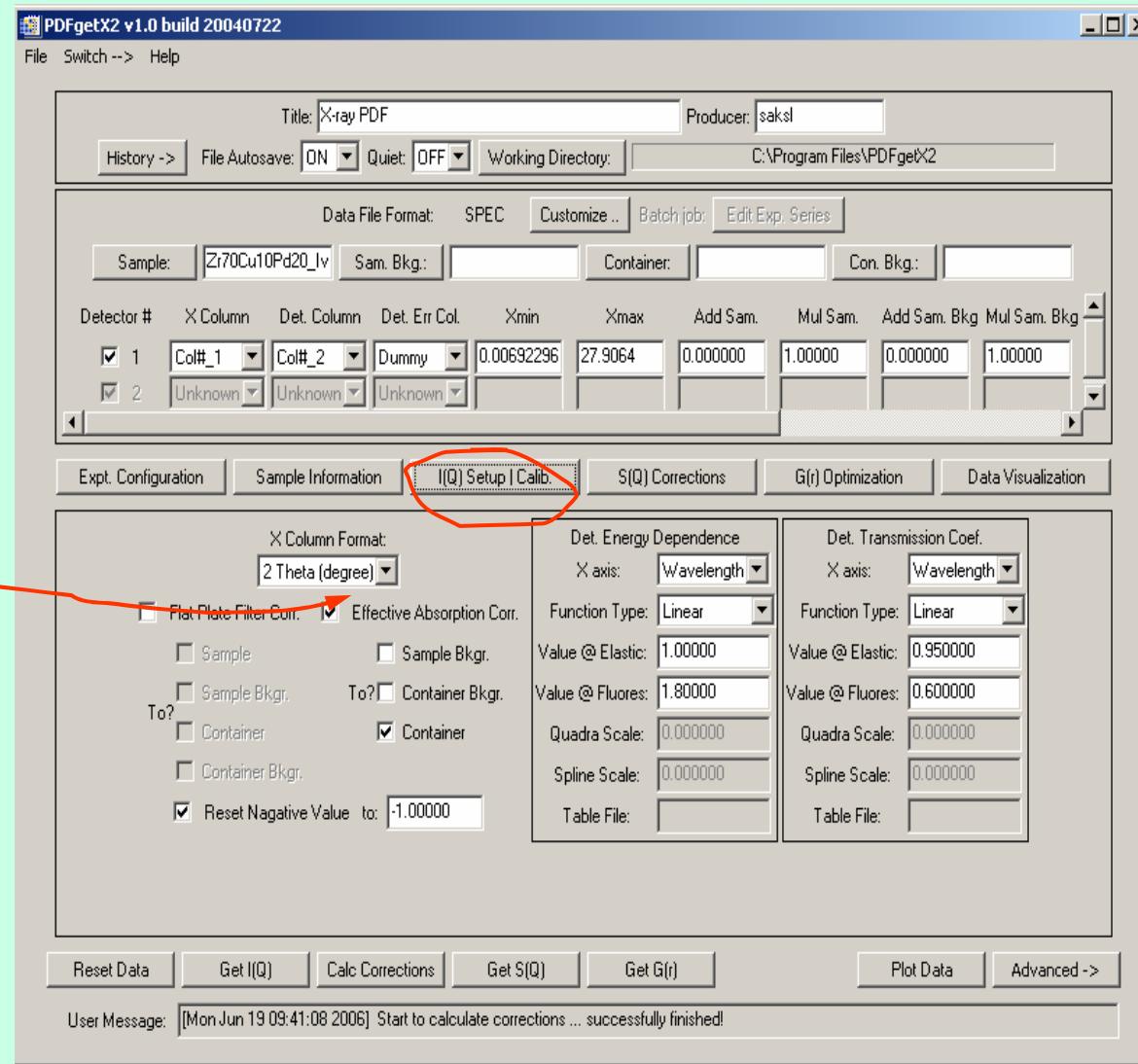


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I(Q) setup



Choose a correct a
X Coumn Format

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Setup $S(Q)$ corrections

PDFgetX2 v1.0 build 20040722

File Switch -> Help

Title: X-ray PDF Producer: saks1
History -> File Autosave: ON Quiet: OFF Working Directory: C:\Program Files\PDFgetX2

Data File Format: SPEC Customize... Batch job: Edit Exp. Series

Sample: Zr70Cu10Pd20_lv Sam. Bkg.: Container: Con. Bkg.:

| Detector # | X Column | Det. Column | Det. Err Col. | Xmin | Xmax | Add Sam. | Mul Sam. | Add Sam. Bkg | Mul Sam. Bkg |
|------------|----------|-------------|---------------|---------|---------|----------|----------|--------------|--------------|
| 1 | Colt#_1 | Colt#_2 | Dummy | 1.20000 | 19.0000 | 0.000000 | 1.000000 | 0.000000 | 1.000000 |
| 2 | Unknown | Unknown | Unknown | | | | | | |

Expt. Configuration Sample Information I(Q) Setup | Calib. S(Q) Corrections G(r) Optimization Data Visualization

Method: Generic I(Q) Simu. SetUp Elastic

Sample Self-Absorption Compton Scattering profile: empirical form

Multiple Scattering (2nd only) Transmissoin Coeff.: 0.980000 Rulandwin Func. Width: 0.020000

Oblique Incidence Breit-Dirac Factor Expo: 2

Fluorescence Type: Constant Scale: 430.0000 Breit-Dirac Factor Expo: 2

X-ray Polarization Energy Dependent: 1/E Linear a: 0.035000 b: 9.999999

Laue Diffuse Scattering Weighting Function Type: $\langle \sin^2 \theta \rangle^2$ Qstart: 3.000000 Width: 100 Cycles: 600

Edit S(Q) w/: Const. + 0.00000 * 1.000000 Qstart: 3.000000 Width: 100 Cycles: 600

Smoothing Qmin: 12.0000 Width: 9 Damp F(Q) Type: Gaussian Width (/Å): 23.00

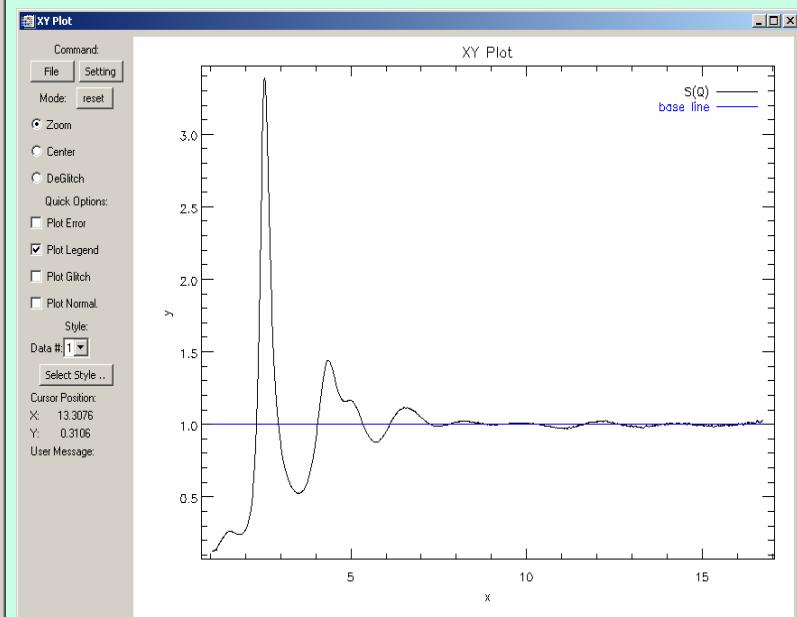
Interpolate Qmin to 0.0 Linear

Auto Scale manual: 0.172768 High Q Range: 60 % calc: 0.17276858

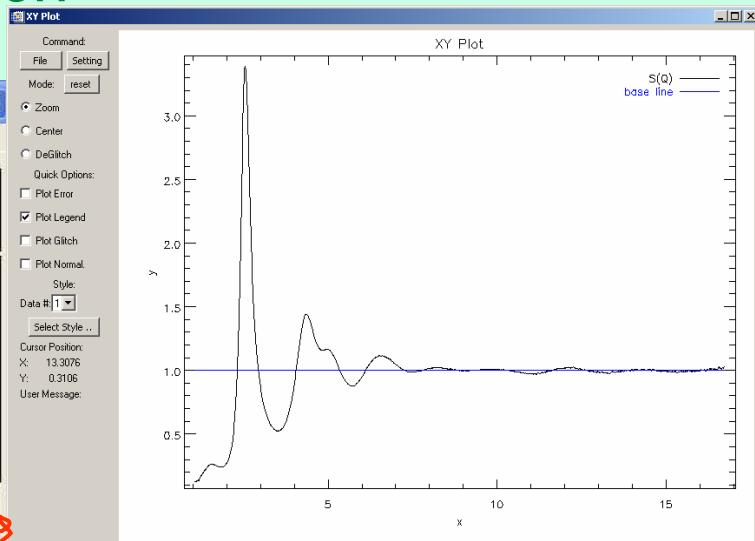
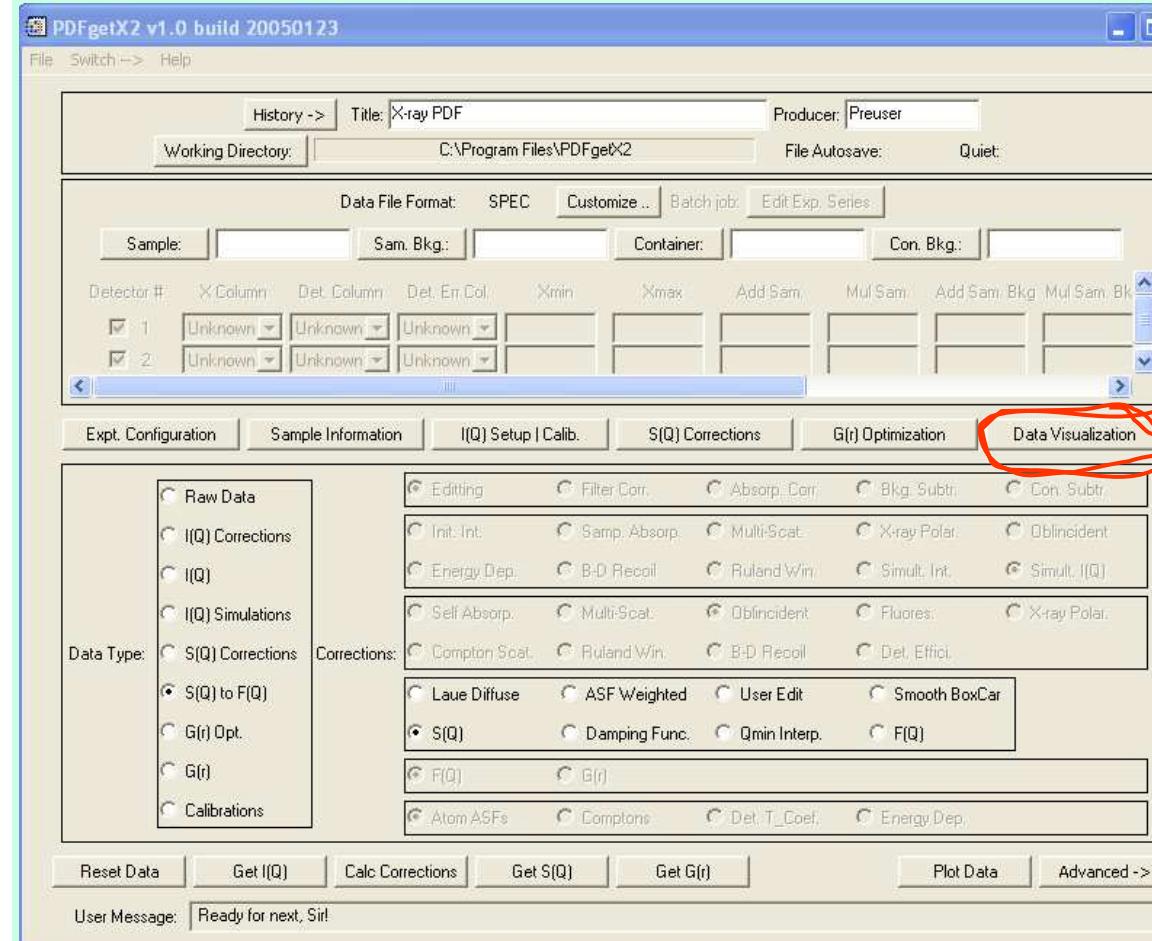
Reset Data Get I(Q) Calc Corrections Get S(Q) Get G(r) Plot Data Advanced ->

User Message: [Mon Jun 19 10:04:16 2006] Start to obtain S(q)... successfully completed!

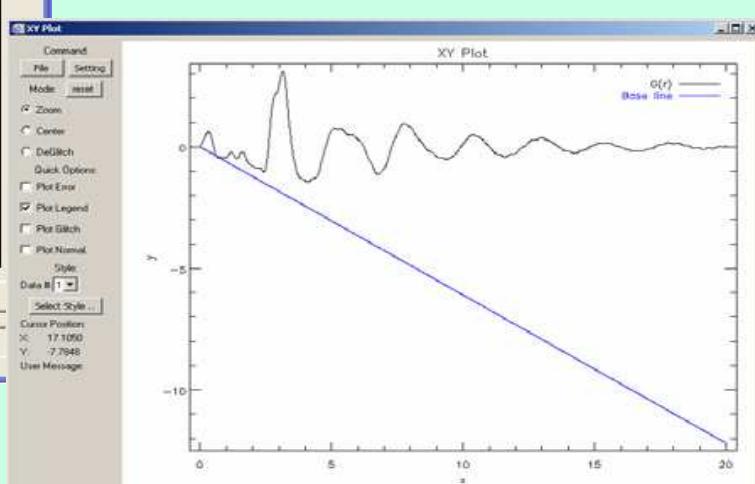
Apply all necessary correction in order that $S(Q)$ oscillates around 1.
Push "Get S(Q)"



Data visualization



Push "Get G(r)"



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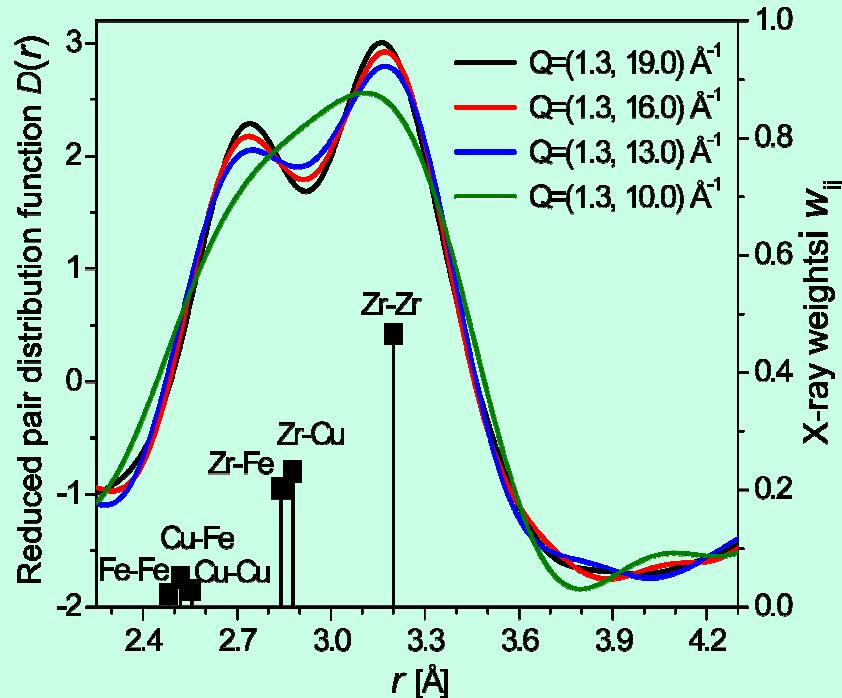
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One example – is the high Q needed?

First coordination shell of $D(r)$ of the $Zr_{60}Fe_{20}Cu_{20}$ glassy alloy.



Shortening of interatomic Zr-Fe and Zr-Cu distances about 3.5 % is observed.

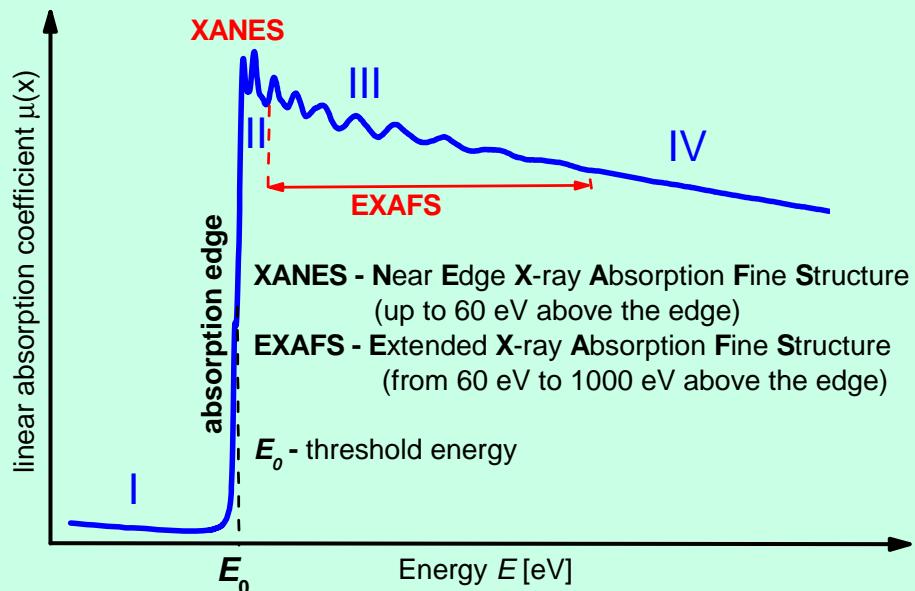
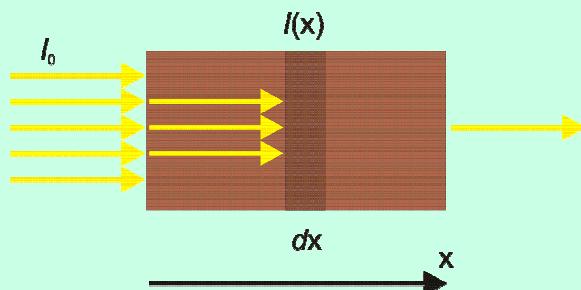
| Atomic pairs | Interatomic bond lengths [\AA] | X-ray weights w_{ij} |
|--------------|---|------------------------|
| Zr-Zr | 3.200 | 0.4660 |
| Zr-Cu | 2.878 | 0.2313 |
| Zr-Fe | 2.841 | 0.2020 |
| Cu-Fe | 2.519 | 0.0501 |
| Cu-Cu | 2.556 | 0.0287 |
| Fe-Fe | 2.482 | 0.0219 |

We need the Q vector as high as possible in order to improve the resolution Δr in the real space.

$$\Delta r = (Q_{\max} - Q_{\min}) / \pi$$

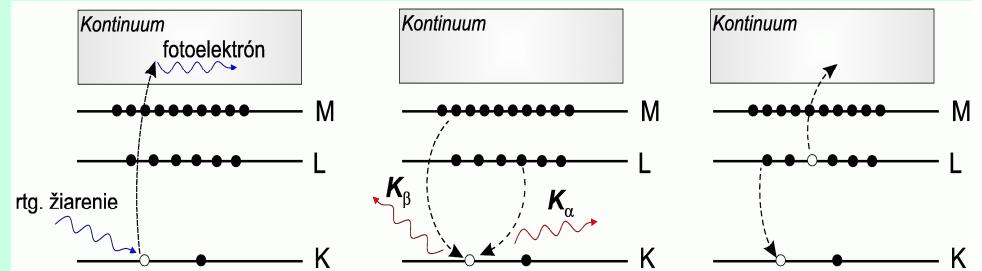


X-ray absorption



Fermi golden rule

$$\mu(E) \propto \sum_f \left| \langle \psi_f | \hat{\mathbf{e}} \cdot \mathbf{r} | \psi_i \rangle \right|^2 \delta(E_f - E_i - \hbar\omega)$$



emission of a
photoelectron

characteristic x-
rays are emitted

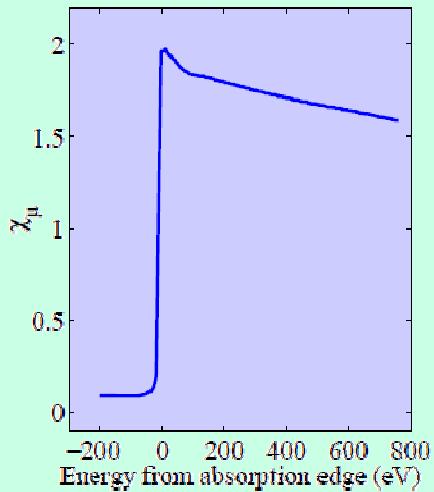
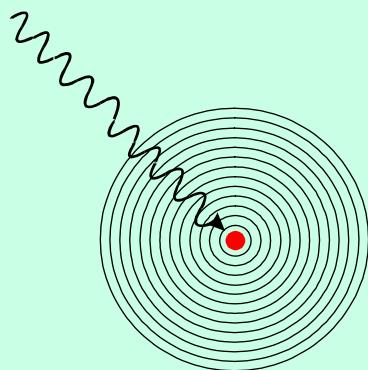
Auger's electrons
are produced



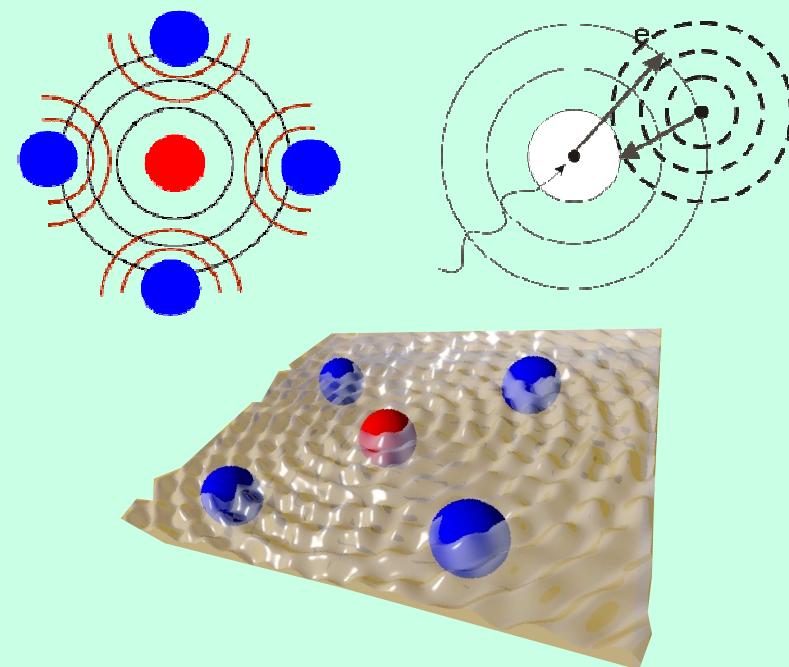
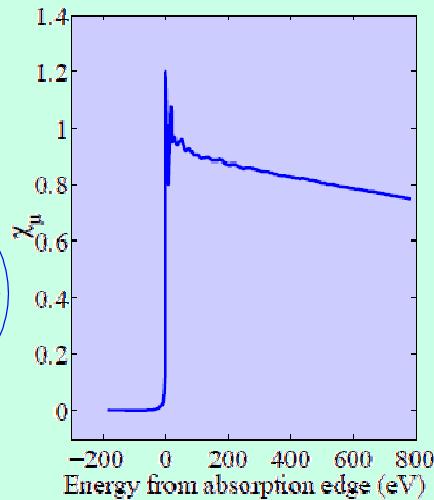
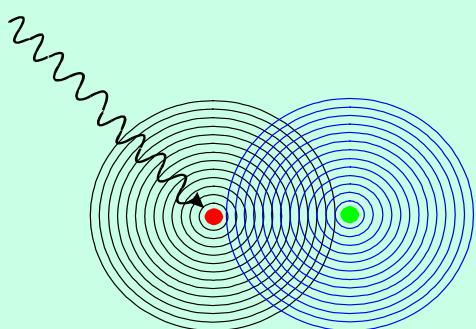
X-ray absorption fine structure signal (XAFS)

Explanation

XAFS of one atom
(inert gases)



XAFS of two atoms



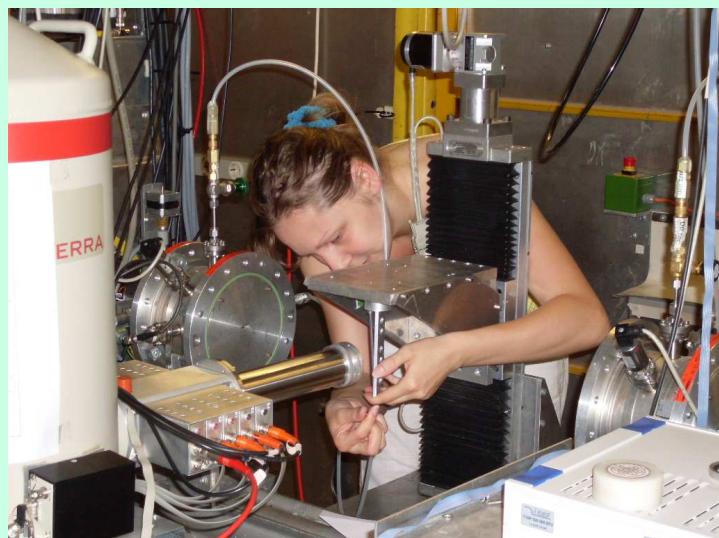
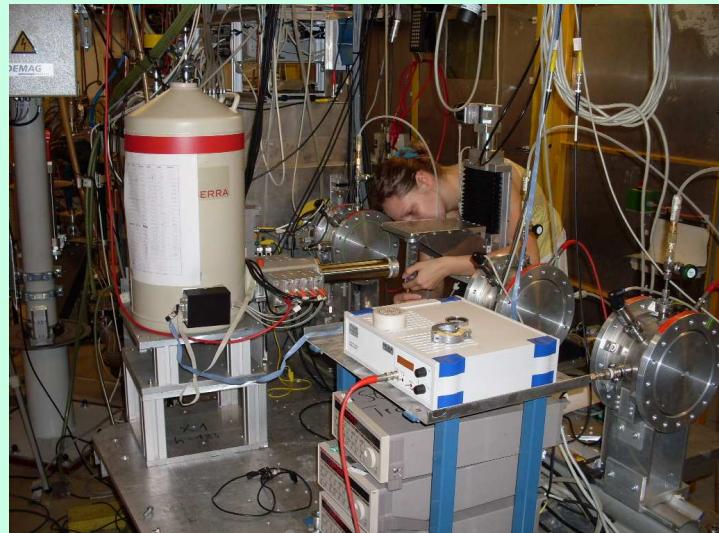
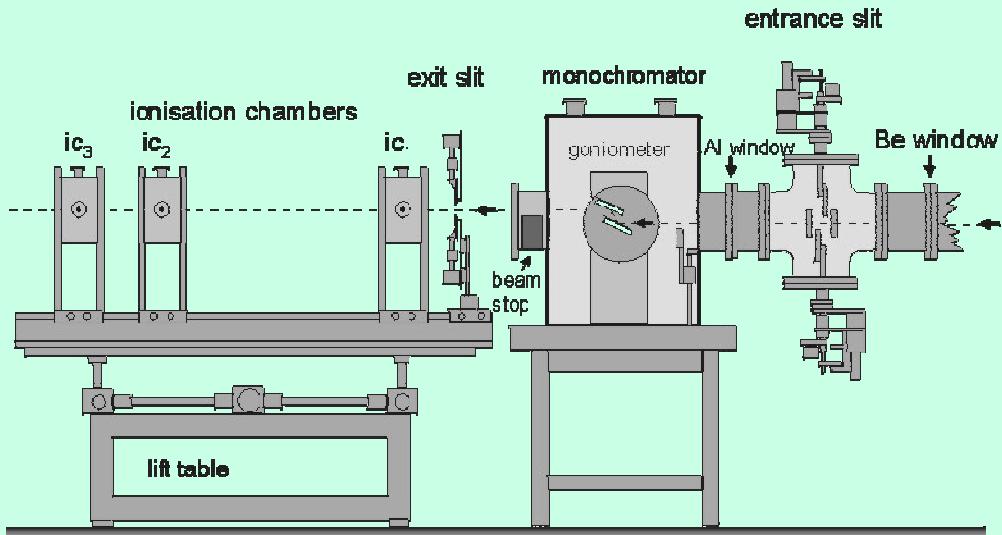
XAFS is the result of interference effects of outgoing and backscattered photo-electrons.

Oscillations of $\mu(E)$ are an unique fingerprint of a local atomic structure around an absorbing atom.



Experimental realization

Setup of the X1 experiment



Modes of realization

Transmission mode: the most preferable, easiest for realization and data correction

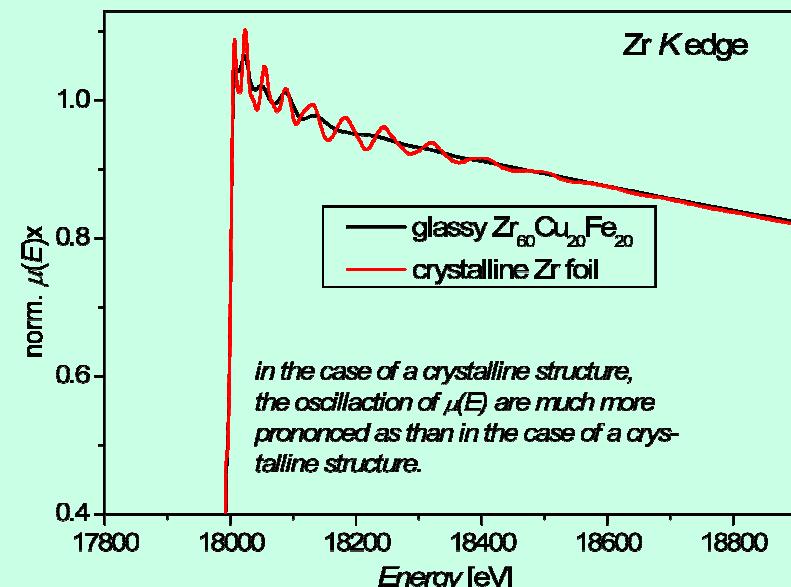
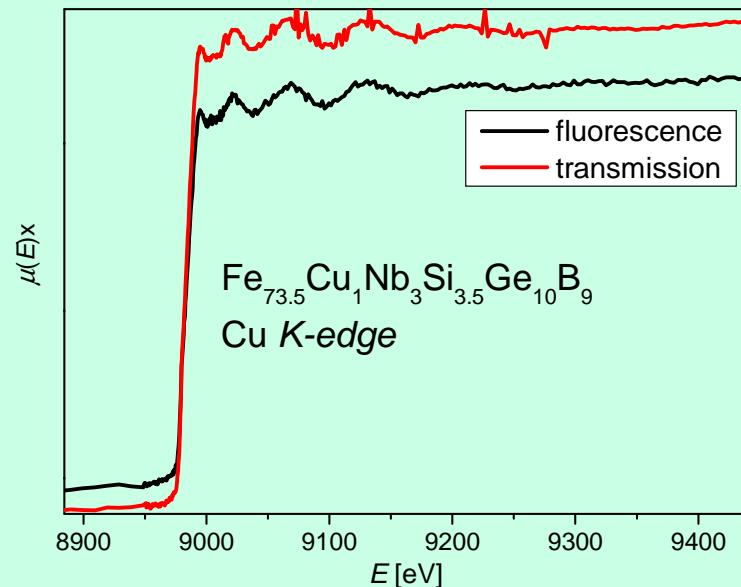
Fluorescence mode: characteristic x-rays are detected

Total electron yield mode: electrons are detected



Experimental realization – practical aspects

Transmission mode – a sample must be homogeneous with constant thickness, absorption $\mu(E)x < 2.6$, not too low concentration of absorbing atoms



Example for a $Fe_{73.5}Cu_1Nb_3Si_{3.5}Ge_{10}B_9$ ribbon:

absorption $\mu(E)x = 2.0$

Fe K-edge: $x = 8.5 \mu m$

Nb K-edge: $x = 85 \mu m$

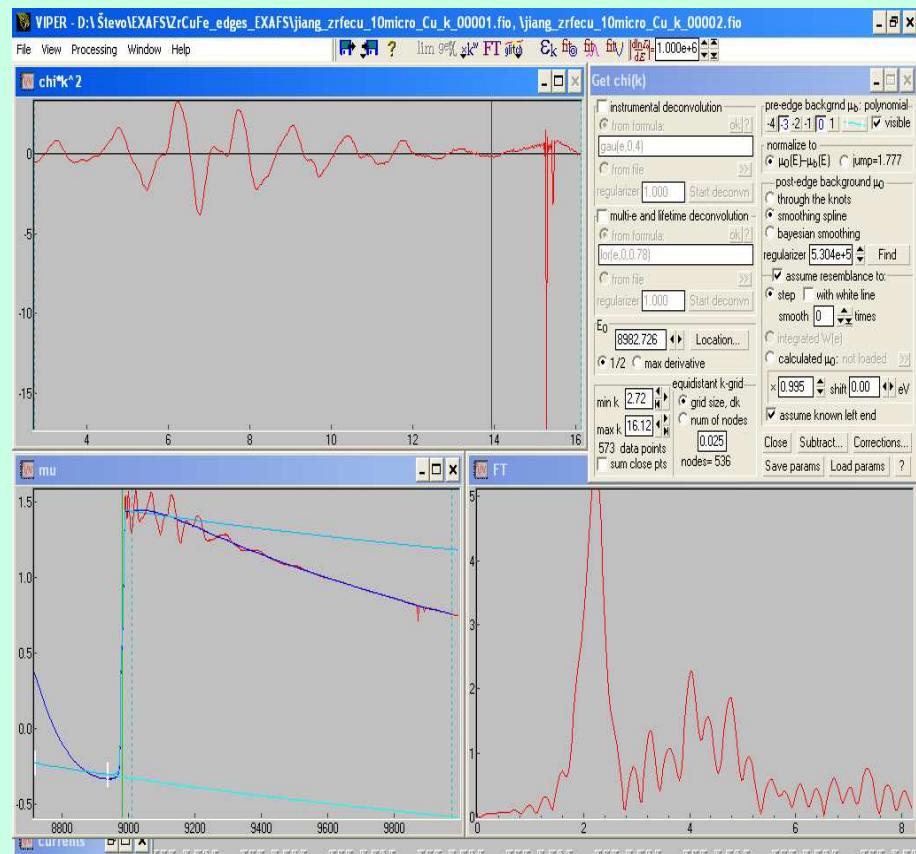
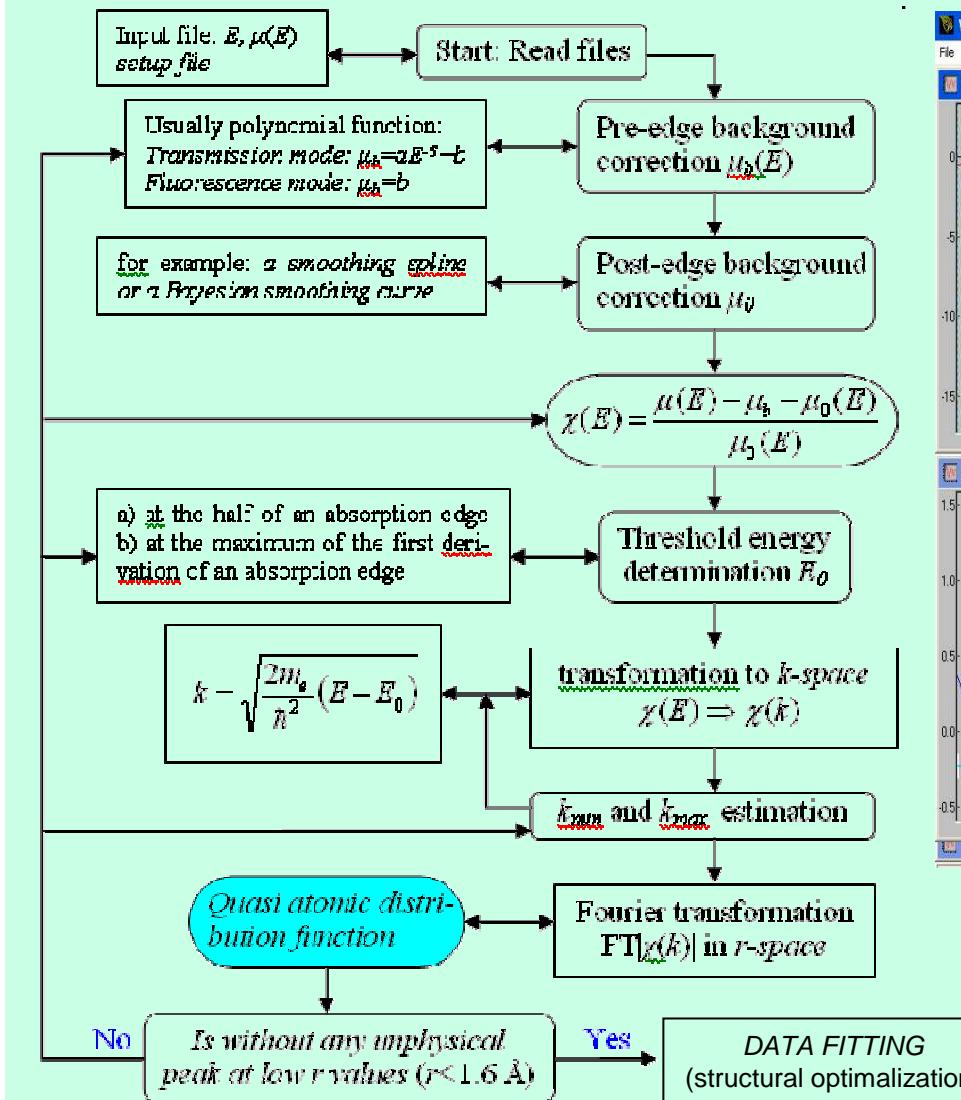
Cu K-edge: $x = 15.2 \mu m$

Ge K-edge: $x = 21.9 \mu m$

low absorbing element concentration results in spikes and strong background tail contribution



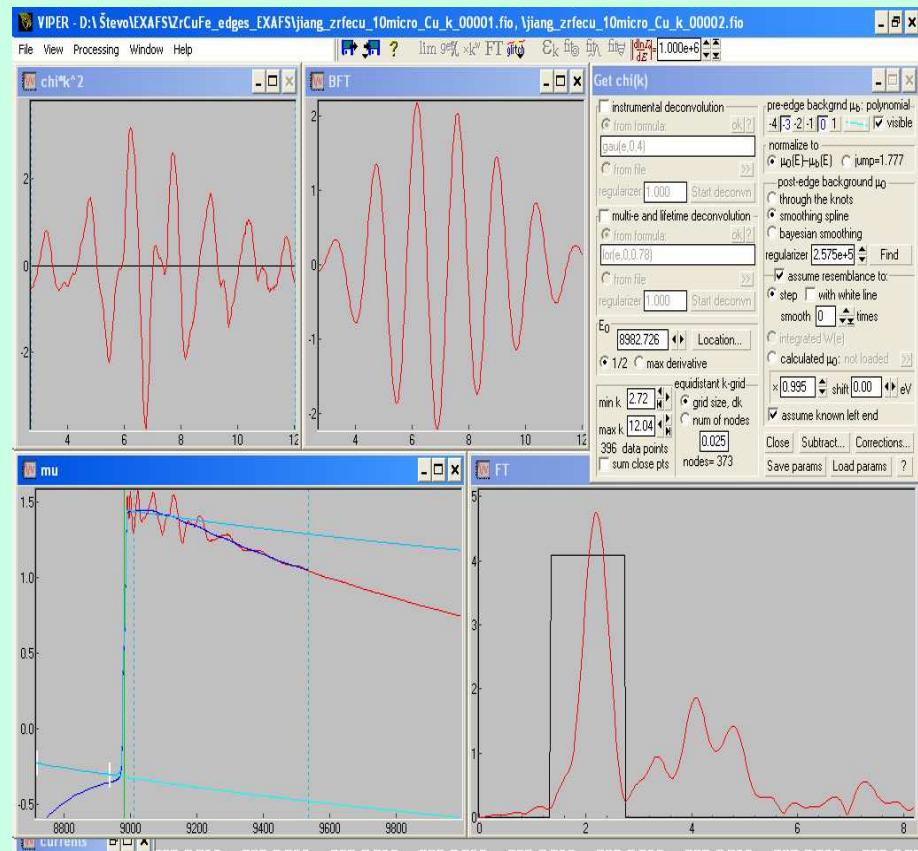
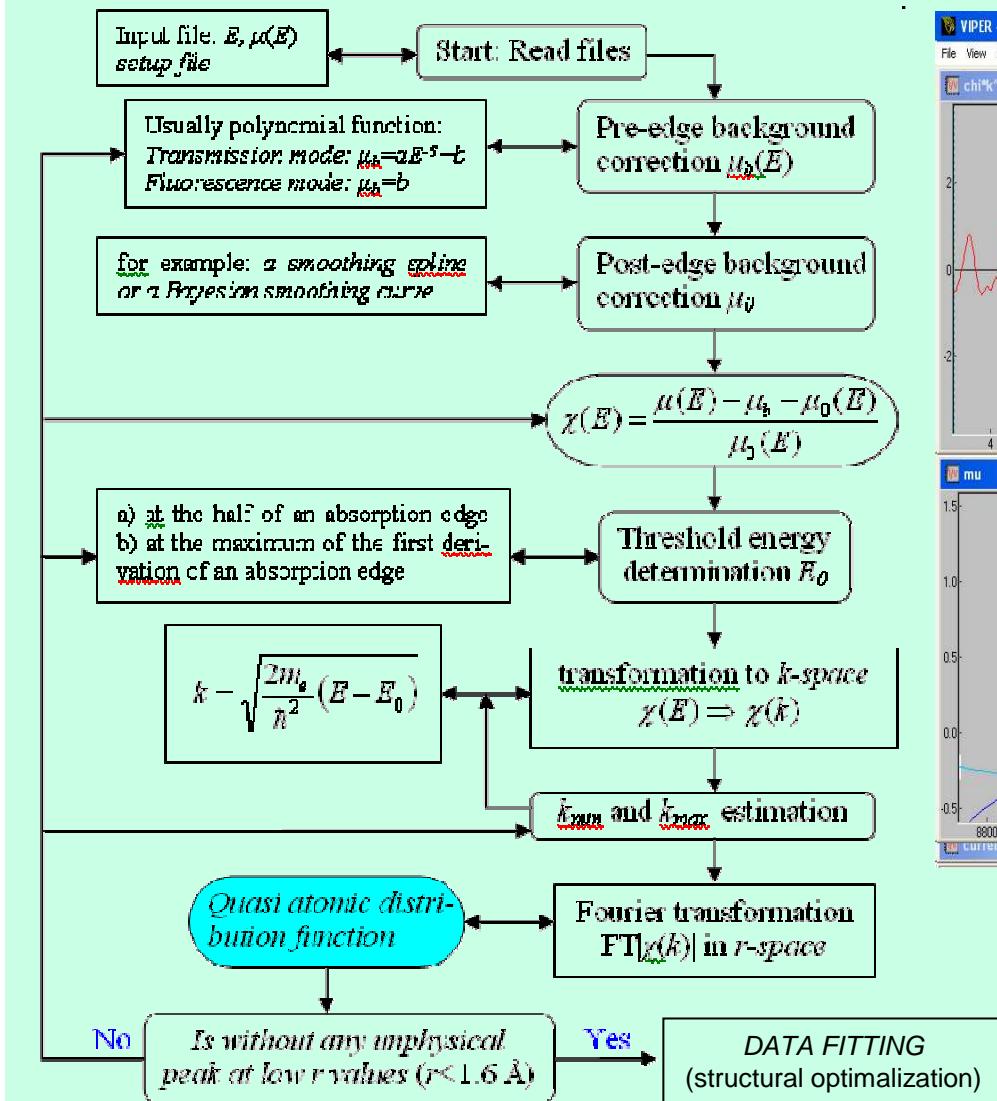
Data processing



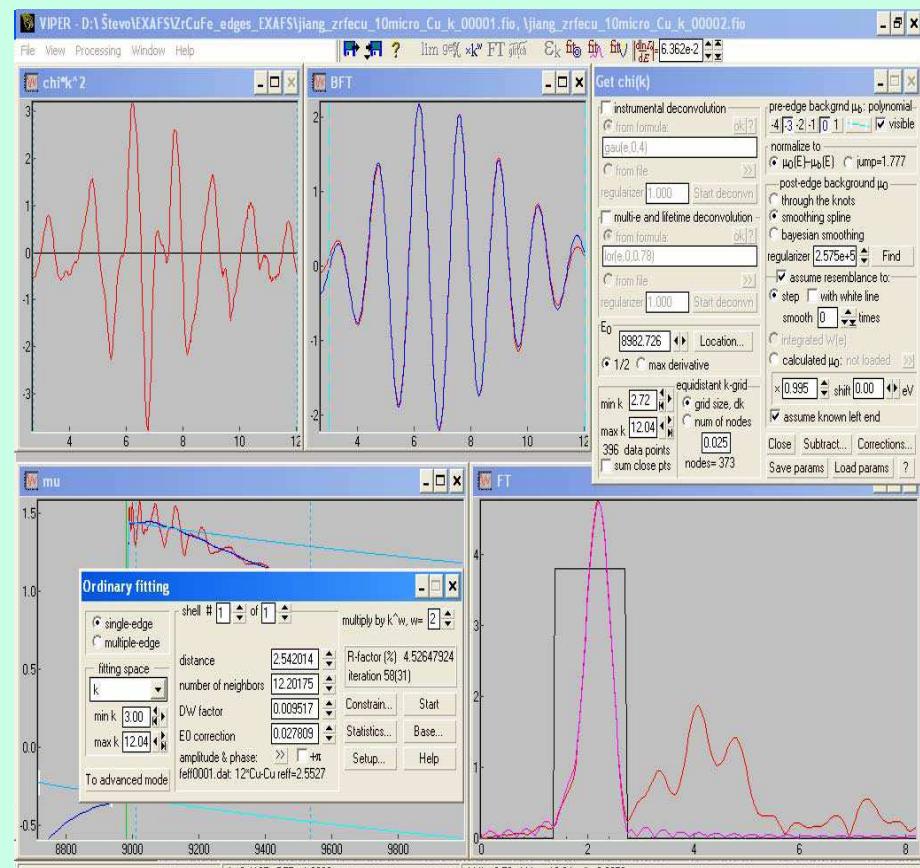
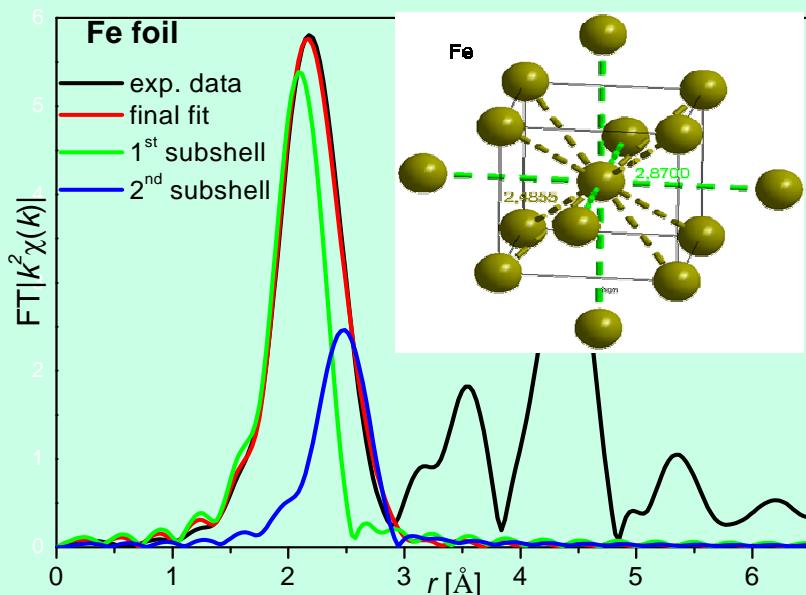
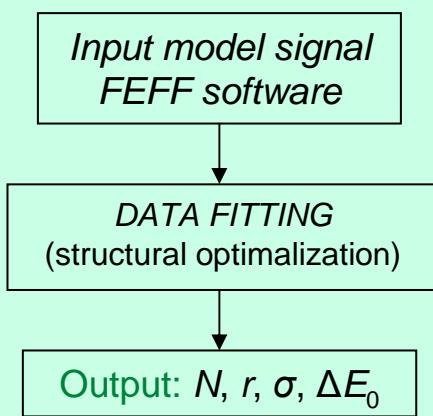
Software VIPER
<http://www.cells.es/Beamlines/CLAES/software/viper.html>



Data processing



Data processing



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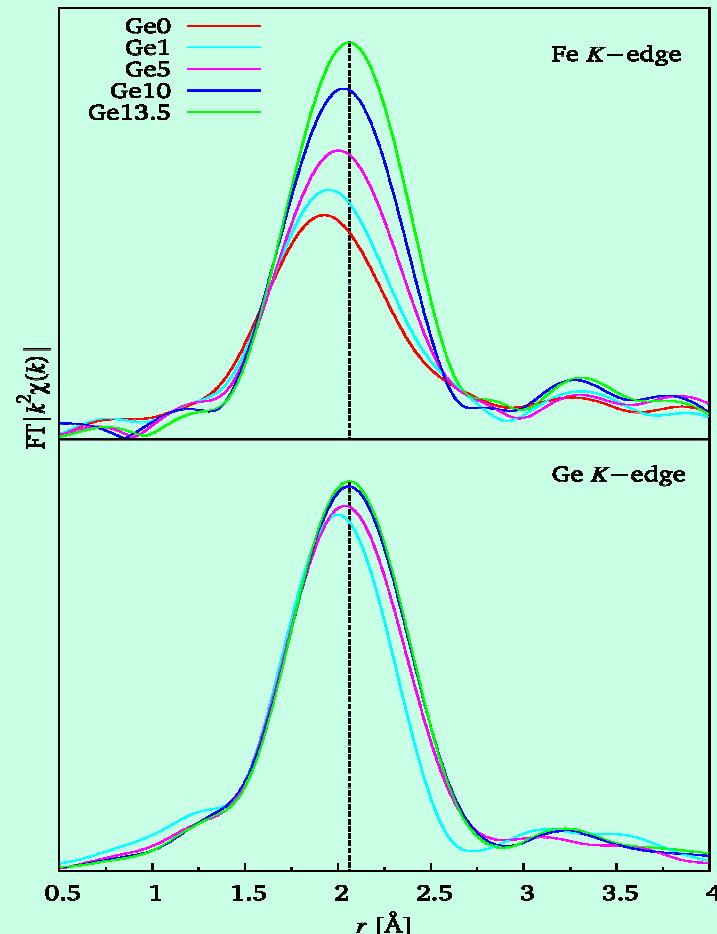
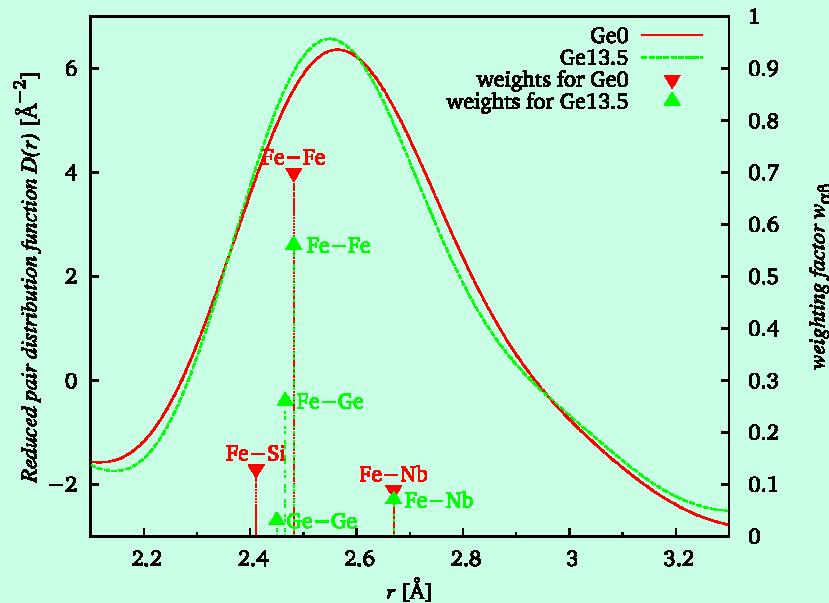


One example



EXAFS – real space

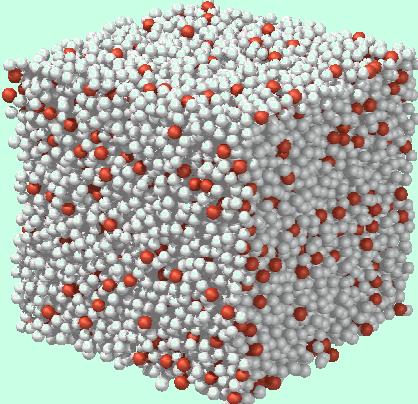
XRD – real space



Principle of RMC modeling

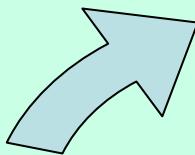
The aim is to minimize difference between functions obtained from model and experiment

Box of ~ 10 000 atoms

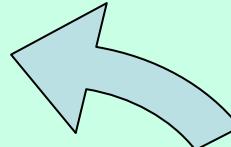


with constraints:

- minimum interatomic distances
- atomic density



$$\psi^2 = \frac{1}{\delta} \sum_{i=1}^m [\zeta^{\text{exp}} - \zeta^{\text{RMC}}]^2$$



How???

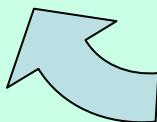
1. Move:

One particle is moved randomly taking into account applied constraints.

2. Acceptance of the move:

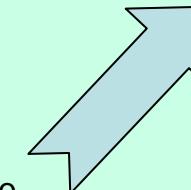
First the experiment-model difference is calculated

If $\psi^2_{n+1} < \psi^2_n$ the move is always accepted.
If $\psi^2_{n+1} > \psi^2_n$ the move is accepted with the probability $\exp[-(\psi^2_{n+1} - \psi^2_n)/2]$



3. Settling:

Everything is repeated until ψ^2 begins to oscillate around a constant value.



Experimental data:

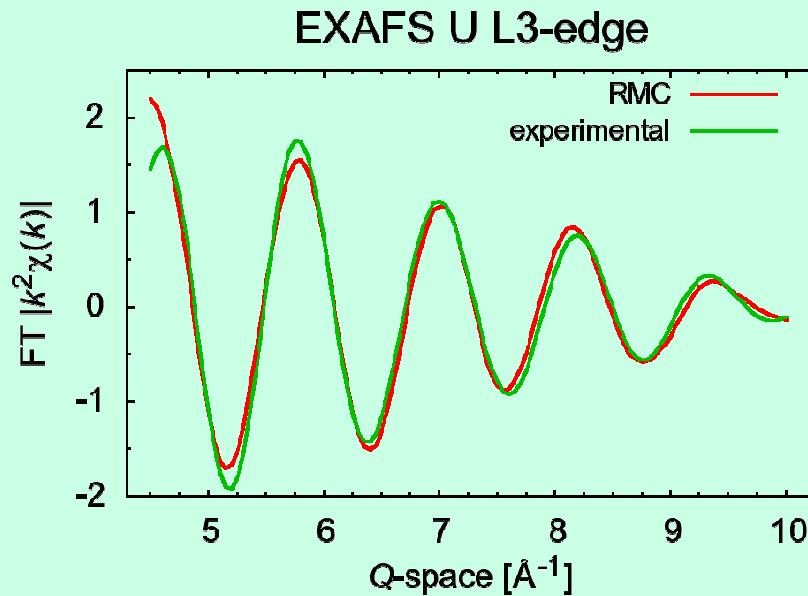
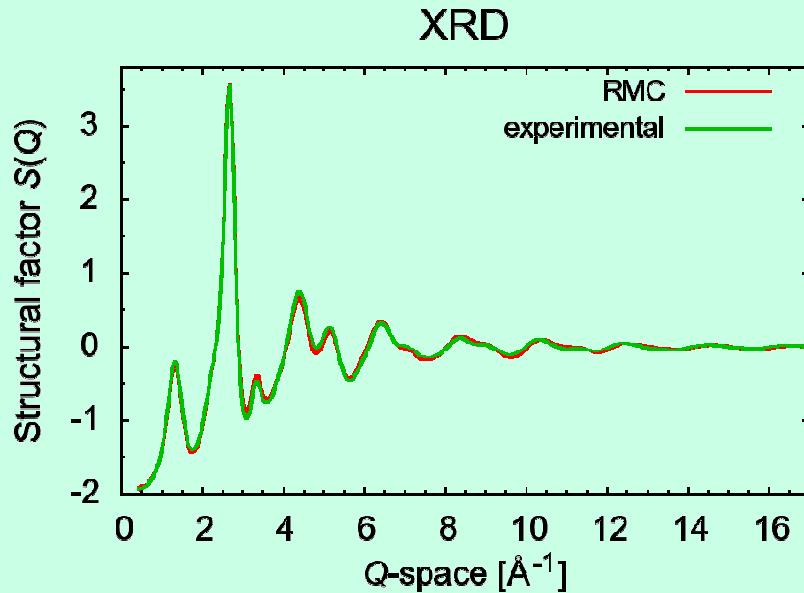
- XRD, ND, ED
- EXAFS
- SAXS, SANX

Result:

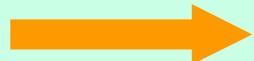
We have an 3D structural model which is consistent with experimental data.



One example: RMC modelling of a binary Al_{92}U_8 glassy alloy



Nice match is obtained between experimental data and model data produced on the base of Reverse Monte Carlo simulation

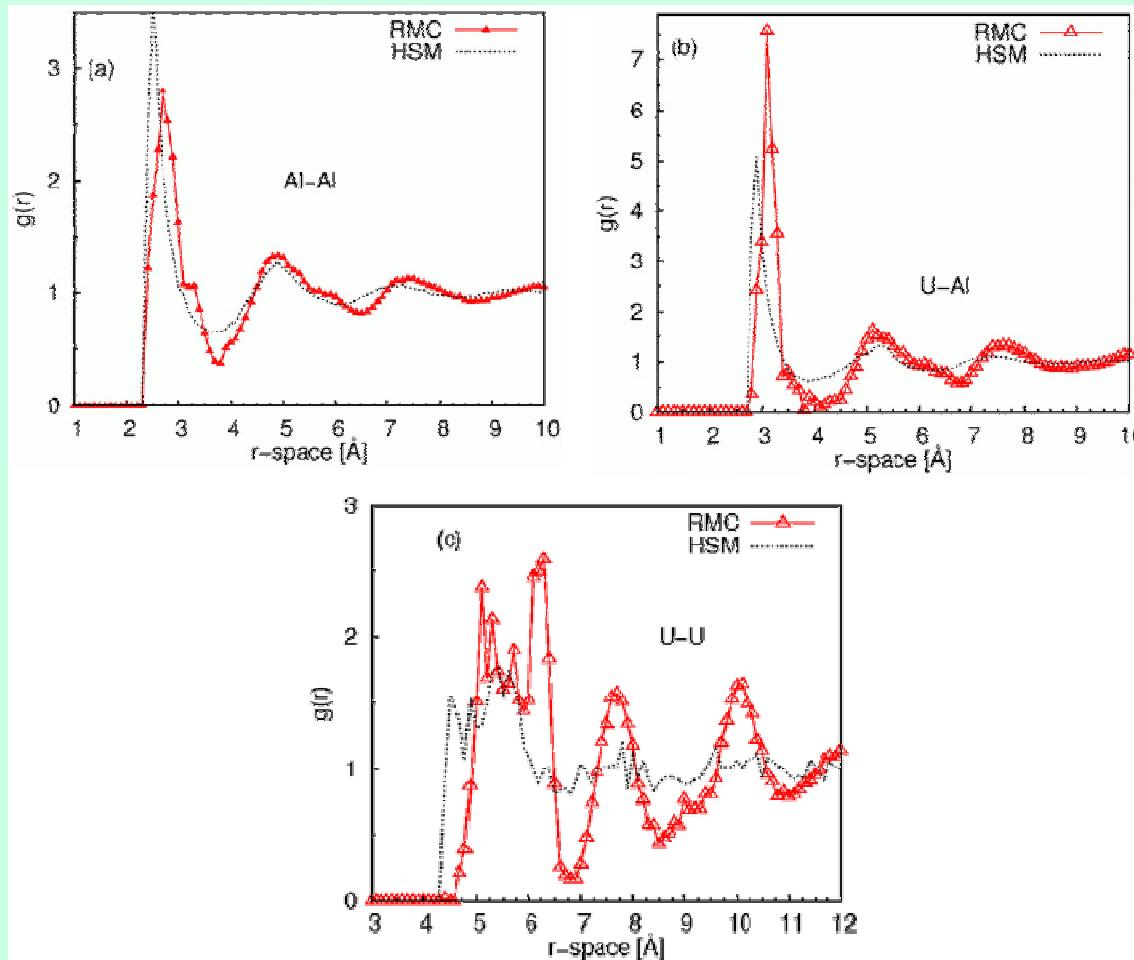


We can perform analysis of the produced 3D structural model.



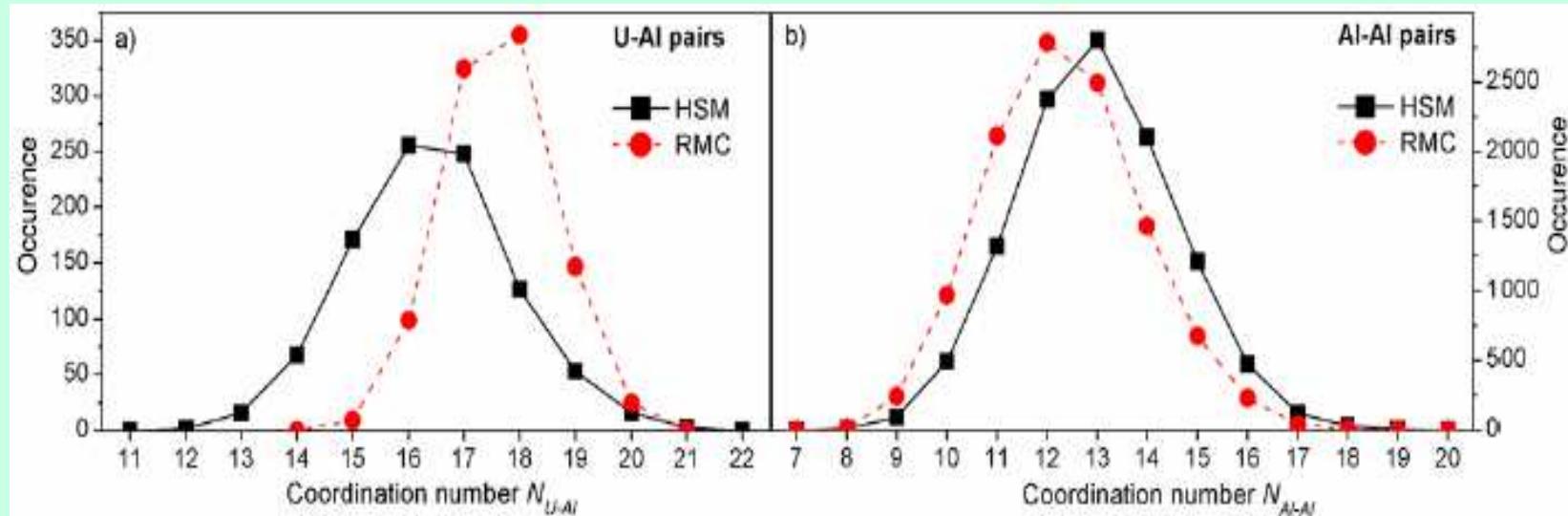
RMC modeling

Partial atomic pair distribution functions can be directly calculated !!!



RMC modeling

Distributions of coordination numbers can be evaluated !!!



Conclusions

- high energy x-ray diffraction: from the structural factor $S(Q)$, the pair distribution function $D(r)$ is calculated → a local atomic arrangement around an average atom is seen and long range correlations are detectable.
- x-ray absorption spectroscopy: an element sensitive method, the nearest neighbouring of a specific type of atoms is observable.
- Combination of XRD and XAFS: compact information about amorphous structure can be obtained.
- Reverse Monte Carlo simulation is a method which can be used for modelling of a 3D disordered structure in a real space.



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Múdrost' minulosti – poznatky prítomnosti – vzdelanie budúcnosti.



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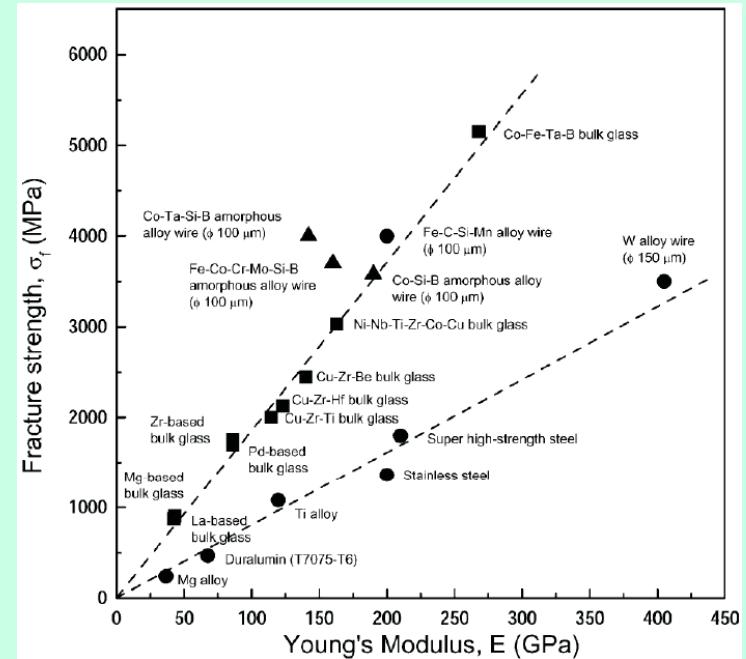
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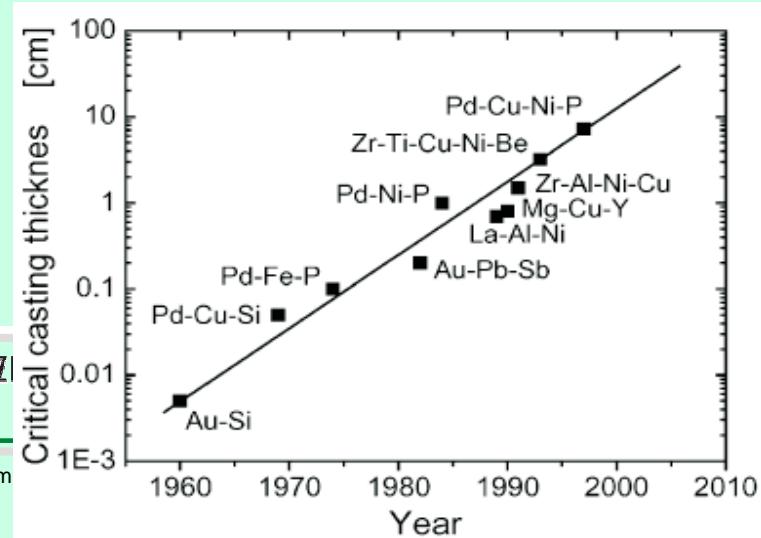
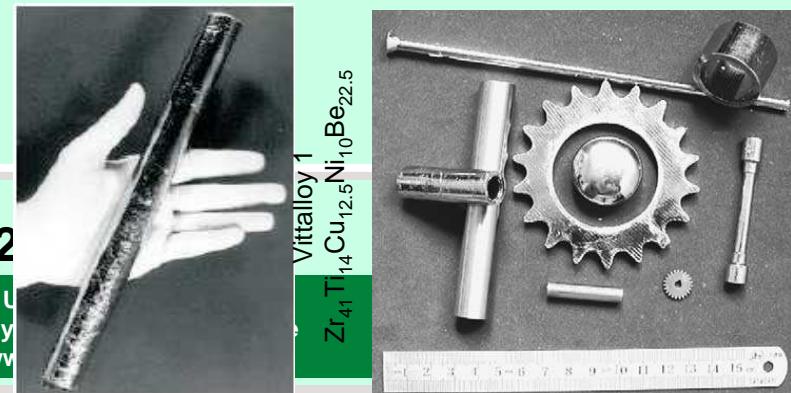
Prečo kovové sklá?

- Výborné mechanické vlastnosti
vysoké hodnoty medzi pevnosťou, lomovej
húževnatosťou, tvrdosťou, výborné elastické
vlastnosti...
- Výborné magnetické vlastnosti
magneticky mäkké materiály
- Výborné antikorózne vlastnosti
- Zaujímavé termodynamické vlastnosti
existencia teploty sklenia T_g , výrazný
pokles viskozity, ľahká tvarovateľnosť
materiálu



Nevýhody

- obmedzená schopnosť sklenia
- neexistencia oblasti plastickej deformácie



Predmet môjho štúdia

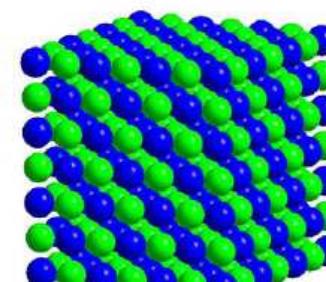
Charakterizácia neusporiadanej štruktúry kovových skiel s využitím vysoko intenzívnych röntgenových zdrojov



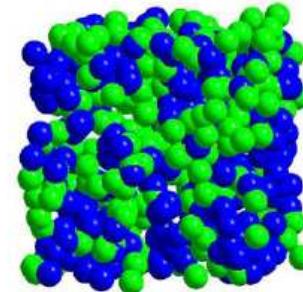
Metallic glasses are a new class of engineering materials having an amorphous structure on atomic level and unlike metals, they can be easily deformed at high temperature.



crystalline



amorphous



RNDr. Štefan Michalik
26.10.2010 v Košiciach

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Múdrost' minulosti – poznatky prítomnosti – vzdelanie budúcnosti.



Zhrnutie: vlastnosti SŽ

- vysoká intenzita
- nízka emitancia, SŽ je takmer úplne kolimované
- vysoký stupeň lineárnej alebo eliptickej polarizácie
- široké a spojité energetické spektrum
- pulzný charakter žiarenia na úrovni nanosekúnd

